

Crystallographic Order and Disorder in
Quasi-One-Dimensional Conductors

Thesis by
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Abstract

Stacked, conducting isocyanide complexes of rhodium(I) were synthesized and characterized by x-ray diffraction and electrical measurements. Good crystals of orthorhombic $[\text{Rh}(\text{CNCHCH}_2)_4]\text{ClO}_4$, with $a_0 = 8.81 \text{ \AA}$, $b_0 = 22.85 \text{ \AA}$, $c_0 = 12.70 \text{ \AA}$, were obtained and a structural refinement was carried out in space group Immm. The rhodium chain is nearly uniform ($\text{Rh-Rh} = 2.94 \text{ \AA}$) and ligands and anions show considerable disorder. The room temperature conductivity is $\sim 2 \Omega^{-1} \text{ cm}^{-1}$ and conductivity is activated. The moderately high conductivity of $[\text{Rh}(\text{CNCHCH}_2)_4]\text{ClO}_4$ is due to the presence of a low-lying conduction band rather than a non-integral rhodium oxidation state.

The disorder, tetragonal phase of $(\text{TTF})\text{Cl}_x$, $a_0 = 11.19 \text{ \AA}$, $c_0 = 3.60 \text{ \AA}$, was studied for compositions $x = 0.67$, $x = 0.70$. A room temperature structural refinement in space group $\text{P4}_2/\text{mm}$ revealed eclipsed stacking of TTF cations and extremely high disorder of chlorides in channels. Low temperature studies revealed ordering of chloride ions for both compositions. $(\text{TTF})\text{Cl}_{0.67}$ undergoes an incomplete structural transition to a monoclinic symmetry phase at $\sim 250^\circ \text{ K}$. Ordering of chloride ions occurs at the same temperature. Fast cooling ($> 1^\circ \text{ K/hr}$) results in peak broadening which is apparently due to the very small size of diffracting domains within the crystal. Both small domain size and the inequivalent environments of TTF

cations following chloride ordering may contribute to the drop in conductivity observed at the phase transition.

Structural refinements of both the subcell (space group $Cmca$, $a_0 = 18.47 \text{ \AA}$, $b_0 = 4.95 \text{ \AA}$, $c_0 = 18.30 \text{ \AA}$) and full cell (space group $Pmc2_1$, $a_0 = 18.47 \text{ \AA}$, $b_0 = 9.90 \text{ \AA}$, $c_0 = 18.30 \text{ \AA}$) of a low disorder crystal of TTT_2I_3 were carried out. The iodine chain is highly disordered and all sites have less than full occupancy. The presence of I_3^- and I_2 species in the chain is likely. The resulting aperiodic potential due to the iodide chain may be expected to be retained at low temperature and inhibit a metal-to-insulator transition.

Electrochemical crystal-growth experiments involving TTF, TMTSF, and HMTSF gave successful results with the first two donors. Crystals of $(TMTSF)Br_{0.8}$ and $(TMTSF)(SCN)_{0.5}$ are isostructural, although the latter exhibits satellite reflections (period = $4.6 \times c_0$) in diffraction patterns. Both structures were refined in space group $Cmcm$, and the satellite data of $(TMTSF)(SCN)_{0.5}$ was modeled in space group $Cmc2_1$. Unit cell parameters are $a_0 = 9.798 \text{ \AA}$, $b_0 = 23.837 \text{ \AA}$, $c_0 = 7.095 \text{ \AA}$, for the bromide and $a_0 = 9.919 \text{ \AA}$, $b_0 = 24.124 \text{ \AA}$, $c_0 = 7.220 \text{ \AA}$, for the thiocyanate. The planes of these cations are perpendicular to the z -axis and consecutive cations slip back and forth by $\sim 1.3 \text{ \AA}$ to reduce methyl group steric repulsion.

All of the systems studied are single carrier conductors with conduction along the cation stack and high disorder of anions. The nature of the disorder and its relationship to phase transitions, as well as interchain coupling and stacking in the cation chain, were evaluated in these compounds. Comparison of $(\text{TTF})\text{Cl}_{0.67}$ and $(\text{TTT})_2\text{I}_3$ were especially useful, as both exhibit comparably very short range order of halide ions at room temperature, but different cation stacking arrangements (eclipsed and slipped, respectively) and hence different interchain coupling and electronic bandwidth. Structural studies at room temperature and low temperature provided an opportunity to understand the important differences in the electrical properties of these two materials.

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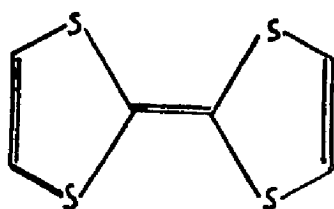
CHAPTER 1

Introduction

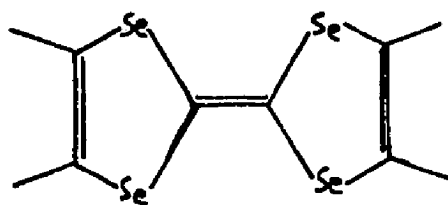
During the past few years there has been considerable interest in highly anisotropic conductors (1-2). From a theoretical viewpoint, one-dimensional metals are much simpler than two- or three-dimensional metal, and they are predicted to have interesting properties such as a Peierl's distortion -- an instability with respect to a metal-to-insulator transition (3-4). Also, a great deal of controversy has originated from theories predicting high temperature superconductivity in highly anisotropic conductors (5-7).

Quasi-one-dimensional conductors, which have high conductivity along one crystal axis and much lower conductivity along the other two axes (8), may approach the behavior of a true one-dimensional metal in some respects, but consideration of the three-dimensional structure and electronic properties of these real materials is necessary for more complete understanding of their properties.

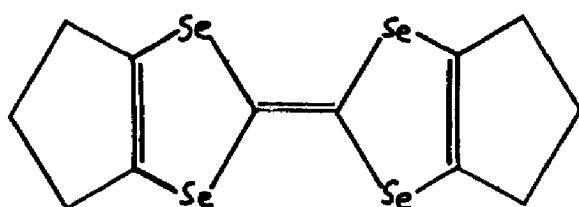
There are two major classes of quasi-one-dimensional conductors. Organic charge-transfer (Figure 1) salts may have two kinds of conducting chains, as in tetrathiafulvalene-tetracyanoquinodimethane (TTF-TCNQ) (9), or only one conducting chain, as in the partially oxidized halide salts of TTF (10-13). These materials generally exhibit metallic conductivity down to a temperature, $T_{MI} \sim 50-200^\circ \text{ K}$,



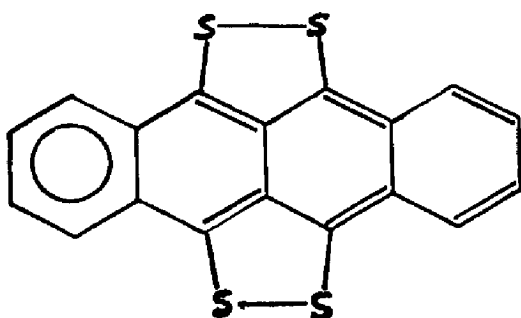
tetrathiafulvalene
TTF



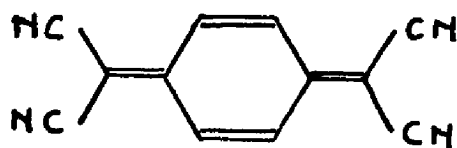
tetramethyltetraselenofulvalene
TMTSF



hexamethylenetetraselenofulvalene
HMTSF



tetrathiatetracene
TTT



tetracyanoquinodimethane
TCNQ

Figure 1. Donors, and the acceptor, TCNQ, in organic conductors.

where a metal-to-insulator transition takes place. The nature of this transition is affected by the presence of disorder, the degree of interchain coupling, and the nature of the molecular stacking along the chain (i.e., eclipsed vs. slipped stacking).

The second major class of quasi-one-dimensional conductors, stacked square planar d^8 transition metal complexes, include various iridium carbonyl halides and partially oxidized platinum cyanide and oxalate salts (14-18). Disorder and interchain coupling are very important in understanding these compounds; stacking may be quite complicated in these materials. The electronic description of the stacked d^8 complexes is somewhat simpler, since metal-metal bonds are the only strong interactions between adjacent monomer units (19-21).

The materials which will be described here include members of both classes. Rhodium(I) isocyanide complexes are electronically similar to the other d^8 metal complexes mentioned above. Since a variety of isocyanide ligands may be employed, systematic alteration of the electrical properties within the series of complexes may be achieved. Conducting salts of organic donors with simple anions were also studied. The donors include TTF, tetramethyltetraseleno-fulvalene (TMTSF), and tetrathiatetracene (TTT) (see Figure 1). Disorder, especially that associated with the anions, is an important structural feature of all of the compounds. The results of detailed structure determinations of tetrakis(vinylisocyanide)rhodium(I)

perchlorate, $\text{TTF Cl}_{.67}$, TTT_2I_3 , $\text{TMTSF}(\text{Br})_{.8}$, and $\text{TMTSF}(\text{SCN})_{.5}$ are reported. Less detailed crystallographic studies of other materials are described.

This work has three specific goals. First, the synthesis and crystallization of new quasi-one-dimensional conductors is important, since the few materials which have received intense study are not sufficient for a good understanding of nearly one-dimensional conductivity. The newer materials have often had more exciting properties, such as retention of high conductivity at very low temperature (22,23).

Second, structural investigations, chemical analysis, and physical measurements carried out concurrently on crystals of the same origin allow extremely useful correlation of subtle structural features with physical properties.

Finally, these materials are of significant crystallographic and chemical interest because of their unique structural characteristics. TTT_2I_3 and $\text{TTF Cl}_{.67}$ have disordered halide lattices with periods two and three times the stacking axis length of the organic sublattice. Tetrakis(vinylisocyanide)rhodium(I) perchlorate has a rhodium atom sublattice with $a_{\text{rh}} = a_0/3$; the ligands and perchlorate anion have extremely high disorder. $\text{TMTSF}(\text{Br})_{.8}$ and $\text{TMTSF}(\text{SCN})_{.5}$ are nearly isostructural. In the latter, however, the thiocyanate positions are modulated with a period $c' \approx 4.5 c_0$. Important parts of the structural work on $\text{TTF Cl}_{.67}$ and tetrakis(vinylisocyanide)rhodium(I) perchlorate are the low temperature investigations. A structural

phase transition of $\text{TTF Cl}_{.67}$ at $\sim 250^\circ \text{ K}$ was studied in detail, and the symmetry of the low temperature phase was determined. A partial structure refinement was carried out on diffraction data of tetrakis (vinylisocyanide)rhodium(I) perchlorate at 22° K .

The structural and electrical characterization of $[\text{Rh}(\text{CNCHCH}_2)_4]\text{ClO}_4$ provides an example of a second way of obtaining high conductivity in d^8 metal complexes, and provides evidence for the importance of interchain coupling in quasi-one-dimensional materials. The importance of disorder in allowing retention of conductivity in single-carrier quasi-one-dimensional materials was revealed in $(\text{TTT})_2\text{I}_3$ and $(\text{TTF})\text{Cl}_x$, which exhibit comparable halide chain disorder at room temperature. Ordering of chloride in $(\text{TTF})\text{Cl}_x$ occurs with a simultaneous drop in conductivity at $\sim 250^\circ \text{ K}$. In contrast, disorder and moderate conductivity are retained at low temperature in $(\text{TTT})_2\text{I}_3$. It appears that the Peierls' transition is incomplete because of the aperiodic potential due to the iodine chain, which produces states within the energy gap. The metal-to-insulator transition is complete in $(\text{TTF})\text{Cl}_x$, although the effect of chloride ordering on the transition is not thoroughly understood. Since the superperiod of the chloride lattice has reciprocal lattice period $\frac{1}{3}c^* = 2k_F$, a favorable interaction between chloride ordering and commensurate charge density formation is expected.

The effect of disorder on conductivity of single-carrier systems was further explored in $(\text{TMTSF})(\text{SCN})_{0.5}$ and $(\text{TMTSF})\text{Br}_{0.8}$. Moderate

disorder associated with the anion lattice is seen in both cases, and an incommensurate superperiod which may result from a $2k_F$ distortion is seen in the thiocyanate.

Structure factor lists are collected in Appendix 1. Electrical properties are described briefly in the text, and plots of the temperature dependence of electrical conductivity and thermoelectric power are collected in Appendix 2. A few remarks concerning structure and refinement for these crystals are given in Appendix 3.

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CHAPTER 2

Conducting Rhodium(I) Isocyanide Complexes

The unusual properties of quasi-one-dimensional conductors have stimulated much interest in the synthesis of new examples of these compounds. While a great variety of organic conductors have been prepared, fewer examples of conducting stacked square planar d^8 metal complexes are known. All of the highly conducting members of the latter group are either iridium carbonyl halides or partially oxidized platinum cyanide and oxalate salts (1-6). The rhodium isocyanide complexes (7-9) reported here are electronically similar to the iridium and platinum compounds.

The intense colors of crystalline rhodium isocyanide complexes indicate that there are significant interactions between monomeric units in the solid state. In addition, the electronic spectra of solutions of these complexes have intense bands in the visible region which show a non-Beers law dependence on concentration (8,10). These bands indicate the presence of oligomers of the tetrakis(isocyanide)rhodium(I) cations. The presence of a mixture of oxidation states is not necessary for oligomerization in solution.

An investigation of several rhodium isocyanide complexes was undertaken in order to determine if highly conducting solids could be prepared. High conductivity could result in these compounds in

two ways. The essential feature of these d^8 compounds is that square-planar complex ions stack along an axis perpendicular to the coordination plane to form linear chains of metal atoms (Figure 1). Electrons are delocalized along this chain in a band formed by overlapping dz^2 metal orbitals, as shown in Figure 2. Since this is a filled band, partial oxidation is required to generate the free carriers needed for metallic conductivity. A simple molecular orbital approach predicts that there will be a low-lying unoccupied band derived from metal Pz and ligand π^* orbitals. If the gap between this band and the filled dz^2 band is sufficiently small, thermally activated conduction will result without partial oxidation (11). The shorter the metal-metal bond, the greater the interaction between monomeric units which give rise to both the dz^2 and Pz- π^* bands will be. The gap will consequently be smaller (Figure 2). Crystalline disorder may also be expected to increase conductivity in this type of material, due to band broadening or creation of states within the gap. In order to understand the electrical behavior of conducting d^8 complexes, it is necessary to evaluate such parameters as the metal-metal bond length, the degree and nature of crystalline disorder, and the oxidation state of the metal.

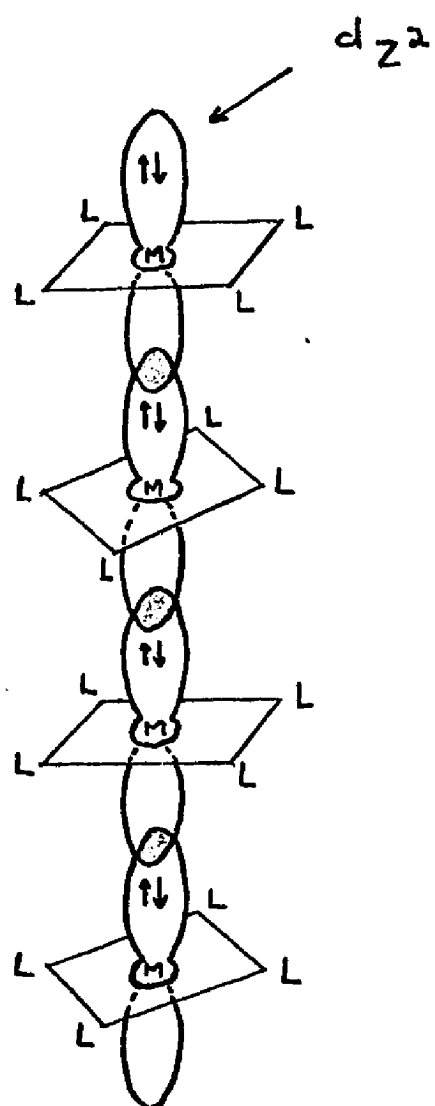


Figure 1. Stacking in square planar d^8 complexes.

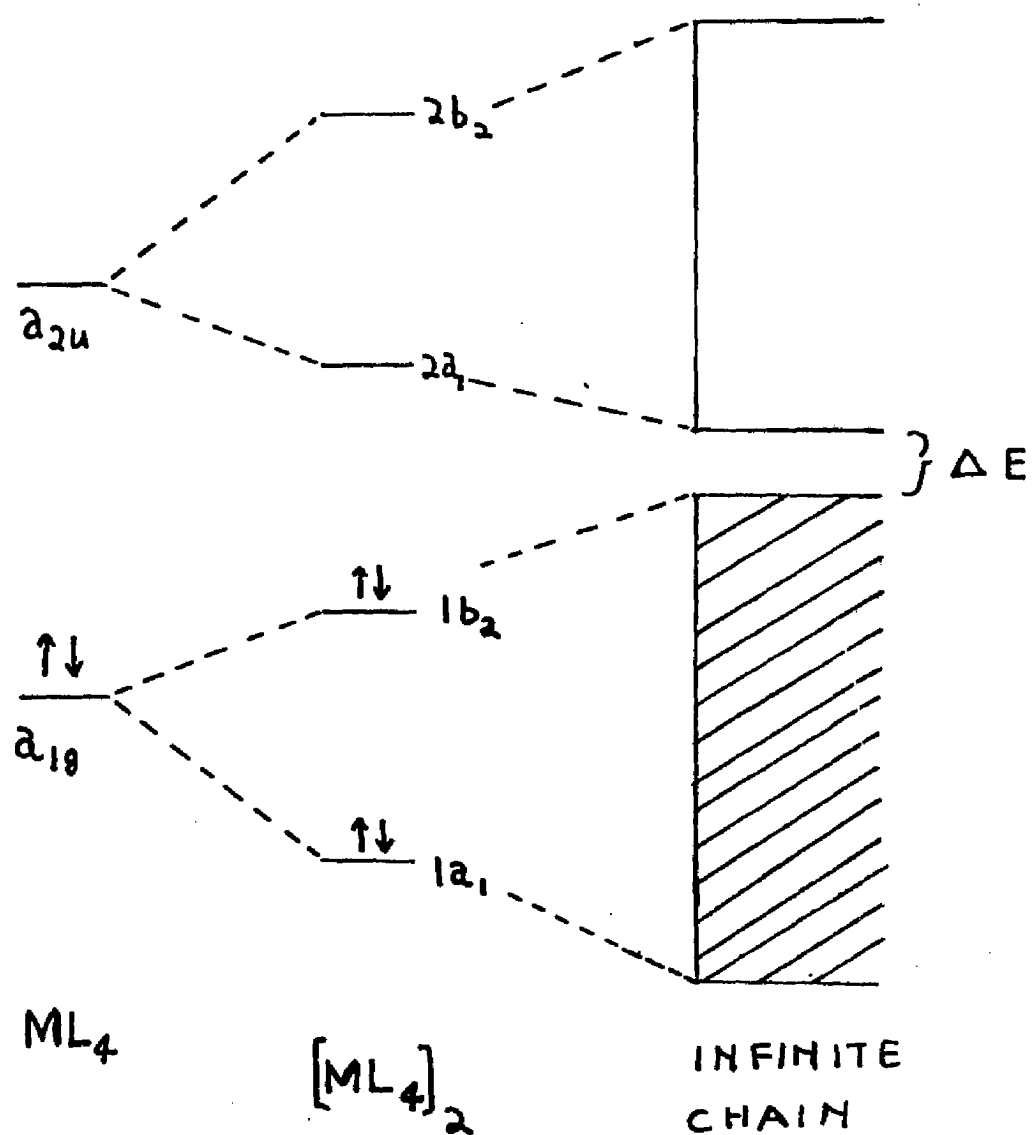


Figure 2. Molecular Orbital diagram of monomer, dimer, and infinite chain of $Rh(CN-R)_4^+$.

Experimental

Synthesis

Bis(1,5-cyclooctadiene)- μ -dichlororhodium, $[\text{Rh}(1,5\text{-C}_8\text{H}_{12})\text{Cl}]_2$, was prepared by the method of Chatt and Venanzi (12), except that recrystallization from acetic acid was omitted.

Bis(1,5-cyclooctadiene)-bis(acetonitrile)rhodium perchlorate and tetrafluoroborate, $[\text{Rh}(1,5\text{-C}_8\text{H}_{12})(\text{CH}_3\text{CN})_2]^+\text{ClO}_4^-$ and $[\text{Rh}(1,5\text{-C}_8\text{H}_{12})(\text{CH}_3\text{CN})_2]^+\text{BF}_4^-$, were prepared by reacting $[\text{Rh}(1,5\text{-C}_8\text{H}_{12})\text{Cl}]$ with AgClO_4 and AgBF_4 respectively, in 1:2 molar ratio in acetonitrile. AgCl was removed by filtration. Addition of diethyl ether yielded the yellow crystalline products. Both $[\text{Rh}(1,5\text{-C}_8\text{H}_{12})(\text{CH}_3\text{CN})_2]^+\text{ClO}_4^-$ and $[\text{Rh}(1,5\text{-C}_8\text{H}_{12})(\text{CH}_3\text{CN})_2]^+\text{BF}_4^-$ were extremely soluble in acetonitrile and other highly polar solvents, and quite insoluble in diethyl ether and hydrocarbons. If not stored in sealed vials, discoloration of the solids occurred, presumably due to loss of acetonitrile. (Caution: All organometallic perchlorates are potentially explosive.)

All isocyanides were prepared by standard methods described in the literature (13-17). Vinylisocyanide was synthesized and used as 30 to 60 mole percent solution in ethanol, rather than in pure form, and was identified by NMR. (Caution: Some isocyanides are explosive. An intermediate in the preparation of vinylisocyanide is reported to be explosive.)

Rhodium(I) isocyanide complexes were obtained by reacting each isocyanide with $[\text{Rh}(1,5\text{-C}_8\text{H}_{12})\text{Cl}]_2$, $[\text{Rh}(1,5\text{-C}_8\text{H}_{12})(\text{CH}_3\text{CN})_2]^+\text{ClO}_4^-$, or $[\text{Rh}(1,5\text{-C}_8\text{H}_{12})(\text{CH}_3\text{CN})_2]^+\text{BF}_4^-$. Since the chloride salts were invariably the most soluble, anion exchange reactions in acetonitrile were also useful in the preparation of the other salts. Growth of crystals was inhibited by the presence of water or cyclooctadiene, among other impurities. Multiple recrystallization of the complexes from acetonitrile solutions containing excess isocyanide was necessary if good single crystals were desired. Acetonitrile was degassed on the vacuum line and distilled from molecular sieves; diethyl ether was distilled from sodium benzophenone ketal. Single crystals of several complexes could be obtained by layering diethyl ether onto an acetonitrile solution of the purified complex; or by slow cooling an acetonitrile-diethyl ether solution of the complex. All syntheses were carried out on the vacuum line. Analytical data are reported in Table 1.

$\text{Rh}(\text{CNCH}_3)_3\text{BF}_4$: 0.33 g $[\text{Rh}(1,5\text{-C}_8\text{H}_{12})(\text{CH}_3\text{CN})_2]^+\text{BF}_4^-$ was dissolved in about 5 ml acetonitrile in a flask with attached Schlenk fritted filters. About 0.5 ml methyl isocyanide was distilled into the reaction flask. The solution became brown, and addition of 10 ml diethyl ether precipitated the complex. The solid was collected on the frit and then recrystallized twice by addition of 1) 5 ml acetonitrile and 0.5 ml methylisocyanide followed by 2) 10 ml diethyl

Table 1. Analytical Data for Rhodium Isocyanide Complexes

		%C	%N	%H	%Cl	%Rh	%O
Rh(CNC ₆ H ₅) ₄ ClO ₄	calc	54.70	9.11	3.27	5.77	16.74	10.41
614.85	found	54.52	9.01	3.54	5.6	15.7	(11.63)
C, H, N average of six analyses							
Rh(CNC ₆ H ₅) _{3.67} (ClO ₄) _{1.33}	calc	50.28	8.38	3.01	7.68	16.77	13.87
613.65	found	50.76	7.77	3.85	7.14	%(Rh+O)=30.64 Δ=30.58	
C, H, N, Cl average of two analyses							
Rh(CNC ₆ H ₅) ₄ Cl	calc	61.05	10.17	3.66	6.44	18.68	
550.86	found	60.83	10.61	4.11	6.17	(18.28)	
Cl average of two analyses							
Rh(CNC ₆ H ₅) ₄ BF ₄	calc	55.85	9.30	3.35	%F=12.62;%R=17.09;%B=1.7		
602.20	found	56.69	9.58	3.80			
C, H, N average of two analyses							
Rh(CNCHCH ₂) ₄ Cl	calc	41.11	15.98	3.45	10.11	29.35	
350.4	found	41.06	16.04	3.74	10.14	29.12	
C, H, N average of two analyses							
Rh(CNCHCH ₂) ₄ ClO ₄	calc	34.76	13.51	2.92	8.55	24.82	15.44
414.6	found	34.96	13.70	3.08	8.80	24.60	(14.86)
C, H, N average of four analyses							
Rh(CNCH ₃) _{3.38} ClO ₄	calc	23.82	13.95	3.00	10.39	30.16	18.75
341.24	found	24.41	13.37	2.92	--	--	--
Rh(CNCHCH ₂) _{2.7} Cl	calc	34.54	13.42	2.90	12.59	36.54	
281.5	found	35.16	12.48	3.18	--	--	

ether. Crystals suitable for x-ray photographic work were obtained by layering 15 ml diethyl ether onto a solution of the complex in 20 ml acetonitrile and 1.0 ml methylisocyanide.

$\text{Rh}(\text{CNCHCH}_2)_4\text{Cl}$: 2.0 g $[\text{Rh}(1,5\text{-C}_8\text{H}_{12})\text{Cl}]_2$ was reacted with ~3.0 g vinyl isocyanide in 8 ml diethyl ether. The black precipitate was collected on a fritted filter and washed several times with additional ether. Yield after drying was 2.7 g. The infrared $\text{C}\equiv\text{N}$ stretching frequency was 2170 cm^{-1} (KCl pellet).

$\text{Rh}(\text{CNCHCH}_2)_4\text{Cl}$: 2.0 g (.008 mole) $[\text{Rh}(1,5\text{-C}_8\text{H}_{12})\text{Cl}]_2$ was reacted with 1.26 g (.024 mole) vinylisocyanide in diethyl ether. The solid product was extracted with acetonitrile and dried.

$\text{Rh}(\text{CNCHCH}_2)_4\text{ClO}_4$: 0.77 g $[\text{Rh}(1,5\text{-C}_8\text{H}_{12})(\text{CH}_3\text{CN})_2]^+\text{ClO}_4^-$ was dissolved in 40 ml acetonitrile. About 1 g of vinylisocyanide was added. Solution was warmed to dissolve all of the complex, then cooled to -15°C and filtered. The red-brown microcrystalline residue was redissolved in 30 ml acetonitrile and ~.5 g vinyl isocyanide. 40 ml diethyl ether was added and product was again collected on the frit. The red-brown mat of hair-like microcrystals was dried by pumping. Failure to recrystallize the initial product apparently results in incorporation of 1,5-cyclooctadiene, which, if present, seems to inhibit formation of good single crystals in later experiments.

Microcrystalline tetrakis(vinylisocyanide)rhodium(I) could also be obtained by Soxhlet extraction of 1.0 g of $\text{Rh}(\text{CNCHCH}_2)_4\cdot\text{Cl}$ with 100 ml dry, degassed acetonitrile containing 10.0 g tetraethylammonium perchlorate under a pressure of ~100 torr dry, oxygen-free nitrogen. After completion of the extraction, the flask was allowed to cool slowly and was left undisturbed for about a day. The yield after filtration was 0.5 g. (Caution: The dry complex will detonate on heating to about 270° C.)

$\text{Rh}(\text{CNCHCH}_2)_4\text{BF}_4$ could be prepared in an analogous manner, but was very hygroscopic.

$\text{Rh}(\text{CNC}_6\text{H}_5)_4\text{Cl}$: 4.0 g $[\text{Rh}(1,5\text{-C}_8\text{H}_{12})\text{Cl}]_2$ was dissolved in a 1:1 mixture of acetonitrile and diethyl ether. 12 ml (~9.6 g) phenyl isocyanide was added to the reaction flask. Solid $\text{Rh}(\text{CNC}_6\text{H}_5)_4\text{Cl}$ was collected in two fractions totalling 8.25 g. The first fraction consisted of brown microcrystals.

$\text{Rh}(\text{CNC}_6\text{H}_5)_4\text{BF}_4$: 3.0 g $[\text{Rh}(1,5\text{-C}_8\text{H}_{12})\text{Cl}]_2$ and 1.5 g NaBF_4 were dissolved in 40 ml acetonitrile. After filtration to remove NaCl , 8.0 g phenylisocyanide was added. Addition of ether gave 6.15 g coppery-brown microcrystals. The infrared $\text{C}\equiv\text{N}$ stretching frequency of a sample in dichloromethane was 2160 cm^{-1} .

$\text{Rh}(\text{CNC}_6\text{H}_5)_4\text{ClO}_4$: 0.5 g $[\text{Rh}(1,5\text{-C}_8\text{H}_{12})(\text{CH}_3\text{CN})_2]^+\text{ClO}_4^-$ was dissolved in 13 ml acetonitrile. About 1 ml phenylisocyanide was distilled into the reaction flask, giving a blue solution. About 20 ml diethyl ether was added to precipitate product, which was collected on a fritted filter and washed with an additional 10 ml of ether. Layering of the additional ether over the filtrate in the receiving flask resulted in crystals up to 0.05 x 0.05 x 0.5 ml.

$\text{Rh}(\text{CNC}_6\text{H}_5)_{3.67}(\text{ClO}_4)_{1.33}$: 0.5 g $\text{Rh}(\text{CNC}_6\text{H}_5)_4\text{Cl}$, somewhat oxidized by several months exposure to air, was extracted with about 10 ml acetonitrile. 0.11 g NaClO_4 was added and after stirring, the solution was filtered to remove the fine precipitate of NaCl. Addition of diethyl ether gave the complex as fine hair-like crystals.

$\text{Rh}(\text{CNC}_6\text{H}_5)_{3.67}(\text{ClO}_4)_{1.33}$ crystals: 0.05 g $\text{Rh}(\text{CNC}_6\text{H}_5)_{3.67}(\text{ClO}_4)_{1.33}$ was placed in a solvent diffusion cell (Figure 3). About 3 ml acetonitrile were distilled onto the complex and about 5 ml diethyl ether were distilled into the other leg of the cell. After sealing and allowing the solvent to reach room temperature, the diethyl ether was slowly distilled onto the acetonitrile solution of the complex. Crystals ~1 mm long appeared after about 5 days.

$\text{Rh}(\text{CNCHCH}_2)_4\text{ClO}_4$ single crystals: 0.40 g $\text{Rh}(\text{CNCHCH}_2)_4\text{ClO}_4$ was placed in a solvent diffusion cell and degassed. 4 ml acetonitrile

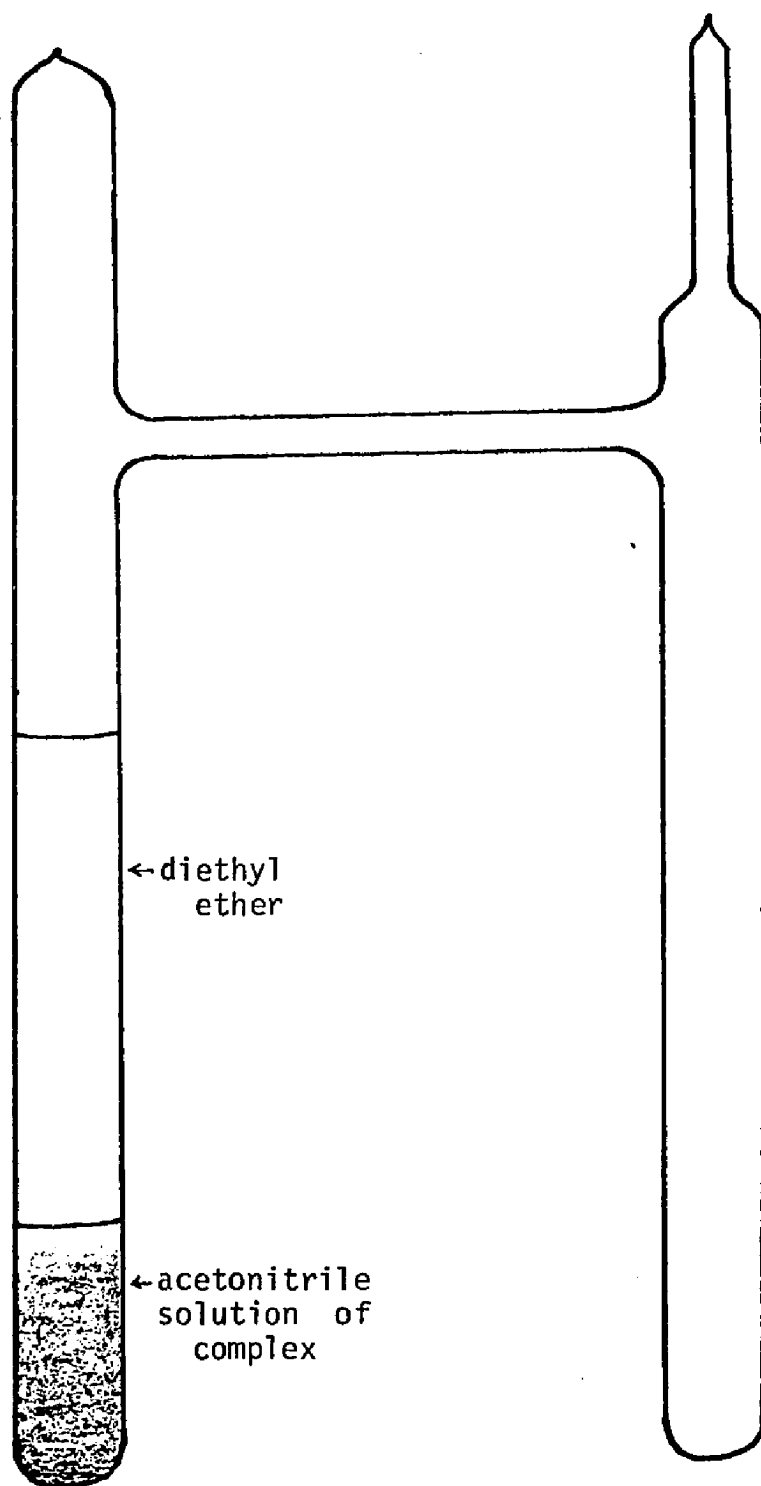


Figure 3. Solvent diffusion crystallization apparatus.

and 15 ml diethyl ether were distilled onto the complex and into the second leg of the cell, respectively. After warming to room temperature, the ether was distilled onto the acetonitrile solution. Coppery crystals, ~10 mm x 0.08 mm x 0.04 mm, were harvested after four days.

Electrical measurements

The electrical conductivity was measured on each complex with the use of two probes applied to a pressed powder of thickness 0.25-0.6 mm. Single crystals of tetrakis(vinylisocyanide)rhodium(I) perchlorate were large enough for four probe conductivity measurements with silver paint contacts. Both kinds of conductivity data obtained at room temperature are given in Table 2. The temperature dependence of the conductivity of tetrakis(vinylisocyanide)rhodium(I) perchlorate is given in Appendix 2.

X-ray and density measurements

The sharp extinction of transmitted polarized light was used as the criterion for selecting crystals for x-ray diffraction work. Most crystals were fibrous and twinned; however, this resulted in misalignment of the diffracting domains (within the crystal) only in the directions perpendicular to the stacking axis. Axis lengths and strong repeats along the stacking (needle) axis could be identified, although the other unit cell parameters could not be determined for most crystals. Tetrakis(vinylisocyanide)rhodium(I) perchlorate formed good single crystals and oscillation and Weissenberg

Table 2. Room Temperature Conductivities of Stacked Rhodium
Isocyanide Complexes

Complex	Source	Conductivity $\Omega^{-1} \text{ cm}^{-1}$
$\text{Rh}(\text{CNCHCH}_2)_4(\text{C}_{10})_4$	single crystal powder	5.6×10^{-3}
$\text{Rh}(\text{CNCHCH}_2)_4\text{Cl}$	powder	2.9×10^{-4}
$\text{Rh}(\text{CNCHCH}_2)_{2.7}\text{Cl}$	powder	2×10^{-4}
$\text{Rh}(\text{CNC}_6\text{H}_5)_4\text{Cl}$	powder	6×10^{-7}
$\text{Rh}(\text{CNC}_6\text{H}_5)_4\text{BF}_4$	powder	2×10^{-5}
$\text{Rh}(\text{CNC}_6\text{H}_5)_{3.67}(\text{C}_{10})_{1.33}$	powder	1.7×10^{-4}
$\text{Rh}(\text{CNCH}_3)_{3.4-x}\text{BF}_4$	powder	2.2×10^{-2}

photographs were used to determine its space group and unit cell dimensions. Table 3 gives stacking axis lengths and strong repeats for the complexes studied. The axis length of tetrakis(phenyl isocyanide)rhodium(I) chloride was determined from a tentative indexing of a Guinier-Hägg powder photograph.

More accurate unit cell parameters (Table 3) were determined for tetrakis(vinylisocyanide)rhodium(I) perchlorate by least-squares refinement based on 2θ , ϕ , and χ of 12 accurately centered reflections, measured on the quarter circle General Electric diffractometer. The crystal used measured 0.05 x 0.1 x 0.5 mm. The intensities of 1199 unique reflections were measured using 2θ - θ at rates of $1^\circ/\text{min}$ or $2^\circ/\text{min}$. Intensities were corrected for Lorentz, polarization, background, absorption and decay; atomic scattering factors of rhodium and chloride were corrected for anomalous dispersion (real part) (18). Initial atom positions were assigned on the basis of Patterson maps. Oxygen atoms and vinyl carbon atoms (as well as hydrogen atoms) could not be seen on Pattersons and were added at calculated 2- or 4-fold disordered positions. After refinement reached $R \approx 0.09$, the vinyl carbon atoms were removed from the model and their positions were reassigned on the basis of difference Fourier maps. Refinement in space group Immm proceeded to $R = 0.084$, $wR = 0.02567$ for 1199 reflections; $R = 0.058$, $wR = 0.02462$ for 676 reflections with $F^2 > 3\sigma$. The real goodness-of-fit reached 4.37 for 1199 reflections. Difference maps were essentially flat after

Table 3. Unit Cell Parameters of Rhodium Isocyanide Complexes

a) Stacking Repeats

Complex	\bar{a}_0 (Å)	n	\bar{a}_0/n (Å)	Source
$[\text{Rh}(\text{CNCHCH}_2)_4]\text{ClO}_4$	8.78	3	2.93	oscillation
$[\text{Rh}(\text{CNCH}_3)_{3.38}]\text{BF}_4$	5.87	2	2.93	oscillation
$[\text{Rh}(\text{CNC}_6\text{H}_5)_4]\text{BF}_4$			3.03	oscillation
$[\text{Rh}(\text{CNC}_6\text{H}_5)_4]\text{Cl}$	23.54	8	2.94	powder
$[\text{Rh}(\text{CNC}_6\text{H}_5)_{3.67}](\text{ClO}_4)_{1.33}$	23.79	8	2.98	oscillation
$[\text{Rh}(\text{CNC}_6\text{H}_5)_4]\text{ClO}_4$			3.20	oscillation

b) Unit Cell Parameters of $[\text{Rh}(\text{CNCHCH}_2)_4]\text{ClO}_4$

	a (Å)	b (Å)	c (Å)	v (Å ³)
Room Temperature	8.810	22.851	12.696	2556
22° K	8.671	22.114	12.391	2376

Space Group, Immm

Absorption Coefficient 99.6 cm⁻¹ CuKα; 11.56 cm⁻¹ MoKαDensity, calc. 1.623 g/cm³ (6 Formula Units/cell)

Major Crystal Faces [010], [011], [001]

c) Indexing of $[\text{Rh}(\text{CNC}_6\text{H}_5)_4]\text{Cl}$ Powder Patterns

θ (obs)	hkl
5.63	003
7.25	222
8.625	400
11.33	006
11.685	520
13.61	620
16.96	650
17.39	652
18.67	654
19.10	0010
31.36	0016

$$\bar{a}_0 = \bar{b}_0 = 20.64 \text{ Å}$$

$$\bar{c}_0 = 23.54 \text{ Å}$$

refinement, except for small residual peaks ($\sim 1\text{e}^-/\text{\AA}^3$) at rhodium and chloride positions.

The intensities of 1506 unique reflections were measured at 21.4 - 22.4° K using 2θ - ω scans on the locally designed low temperature diffractometer (19). Graphite monochromatized $\text{MoK}\alpha$ radiation was used. The crystal used measured 0.036 x 0.081 x 0.36 mm³. The scan rate used was 1°(2 θ)/minute, and backgrounds were measured at both ends of the scan range for a total of 1 minute. The same kind of corrections applied to room temperature data were applied (18). Because of the small size of the crystal, and the low intensity of diffraction with $\text{MoK}\alpha$, only 1032 reflections had observed intensities greater than zero after correction for background. Only 282 reflections had $F^2 > 3\sigma$. Moreover, the rapid drop-off in intensity with increasing 2θ is nearly unchanged between 300° K and 22° K.

The low temperature structure of $[\text{Rh}(\text{CNCHCH}_2)_4]\text{ClO}_4$ was partially refined by least-squares. Only the coordinates and temperature factors of the rhodium atoms and the atoms of one of the three independent vinylisocyanide ligands could be refined, due to the poor quality of the data. The coordinates of the atoms of the other two ligands were determined from Fourier maps. Spurious (noise) peaks on the Fourier map had a magnitude of up to $\pm 3\text{e}^-/\text{\AA}^3$, while the isocyanide carbon and nitrogen peaks had a magnitude about twice as great. Vinyl carbon peaks were of about the same intensity as the larger noise peaks, and were assigned on the basis of both

Fourier maps and geometrical considerations. The partly refined model gave $R = 0.342$ for 1032 reflections with $F > 0$, and $R = 0.132$ for 282 reflections with $F^2 > 3\sigma$. Weighted residuals, wR were 0.0729 and 0.0613 for all 1506 reflections, and 282 reflections having $F^2 > 3\sigma$, respectively. The weighted goodness-of-fit was 2.58 for the full data set.

Electronic Spectra

Electronic absorption spectra of tetrakis(vinylisocyanide) rhodium(I) chloride and tetrafluoroborate in a variety of polar solvents were obtained from 10,000 to 50,000 cm^{-1} . The observed bands and intensities are tabulated in Table 4. The absorption spectra of the chloride salt at different concentrations are shown in Figure 4. Photoacoustic spectra of $[\text{Rh}(\text{CNCHCH}_2)_4]\text{ClO}_4$ and the partially oxidized $[\text{Rh}(\text{CNC}_6\text{H}_5)_3.67](\text{ClO}_4)_{1.33}$ were obtained using powder samples, and are shown in Figure 5 (9).

Results

Characterization and composition

The complexes are black or coppery-brown, hygroscopic (especially with smaller ligands and anions), and difficult to crystallize. The chloride salts are soluble in water and polar organic solvents, while salts of larger anions are nearly insoluble in water. Complexes

Table 4. Solution Spectroscopic Data - $[\text{Rh}(\text{CNCHCH}_2)_4]\text{Cl}$ and $[\text{Rh}(\text{CHCHCH}_2)_4]\text{BF}_4$

Monomer Spectra

Absorbtion band	$\text{Rh}(\text{CNCHCH}_2)_4\text{BF}_4^+$		$\text{Rh}(\text{CNCHCH}_2)_4\text{Cl}^\ddagger$	
	λ	ϵ	λ	ϵ
intraligand absorbtion	233	(34,000)	not measured	
$^1\text{A}_{1g} \rightarrow ^1\text{E}_u$	329	(19,000)	324	(17,000)
$^1\text{A}_{1g} \rightarrow ^1\text{A}_{2u}$	409	(5,000)	395	(3,000)
$^1\text{A}_{1g} \rightarrow ^3\text{A}_{2u}$	457	(700)	450	(550)

$^+$
in CH_3CN

5×10^{-5} and 3×10^{-4} M

‡
in H_2O

9×10^{-5} M

Positions of lowest intense band in the spectra of $[\text{Rh}_n(\text{CNCHCH}_2)_{4n}]^{n+}$

	$\text{BF}_4^-^\ddagger$	Cl^-^\ddagger
$n = 1$	403	395
$n = 2$	555	550
$n = 3$	715	715
$n = 4$		962

‡
in H_2O

9×10^{-4} M

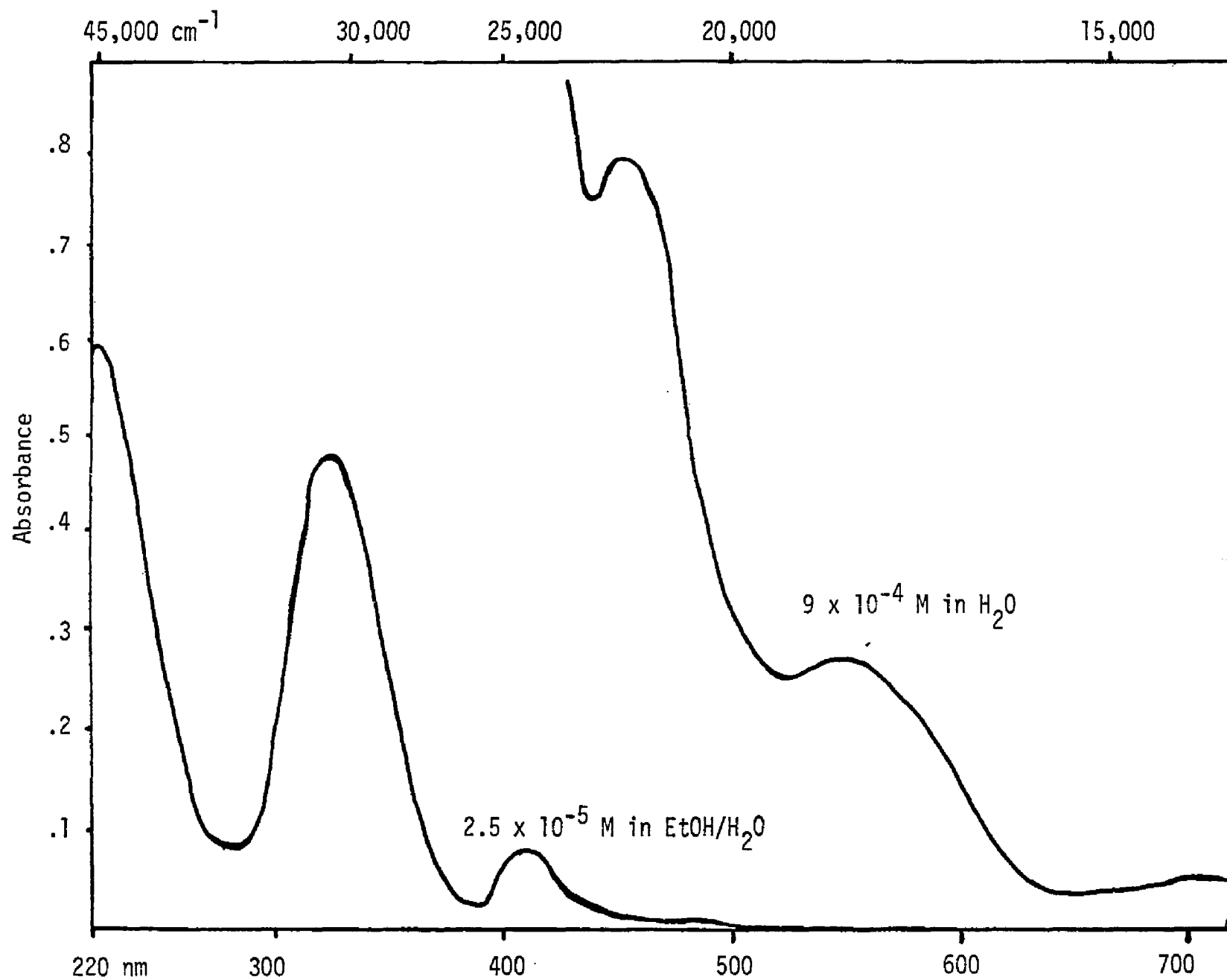


Figure 4. Absorption spectra of $\text{Rh}(\text{CNCHCH}_2)_4\text{Cl}$

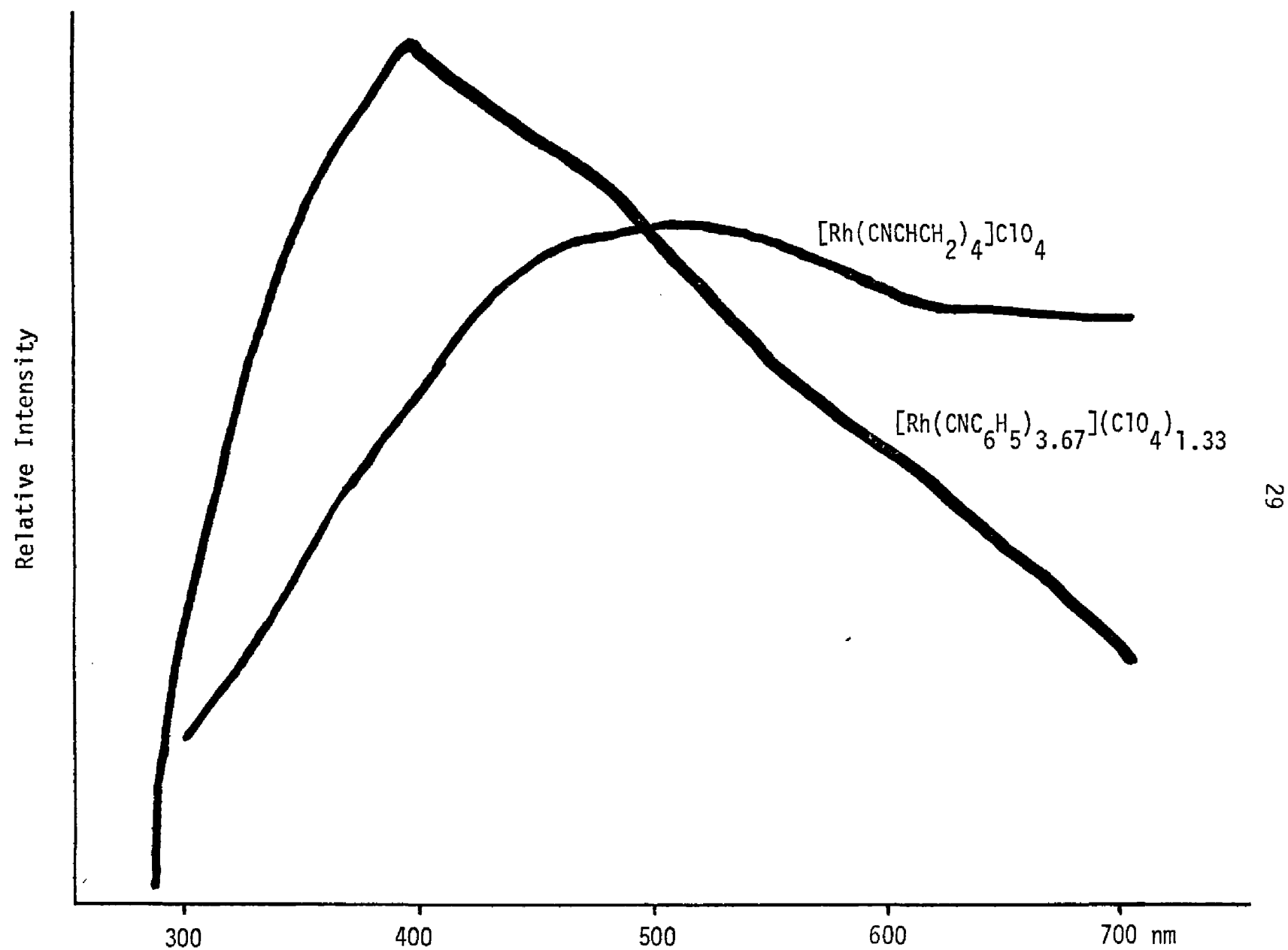


Figure 5. Photoacoustic spectra of solid $[\text{Rh}(\text{CNCHCH}_2)_4]\text{ClO}_4$ and $[\text{Rh}(\text{CNC}_6\text{H}_5)_{3.67}](\text{ClO}_4)_{1.33}$

decrease in solubility with decrease in size of ligand, reflecting stronger crystal-binding forces.

The analytical data in Table 1 indicate that deviations from the expected stoichiometry of tetrakis(isocyanide)rhodium(I) salts may occur. $\text{Rh}(\text{CNC}_6\text{H}_5)_3.67(\text{ClO}_4)_1.33$ clearly demonstrates both non-integral ligand-to-rhodium and anion-to-rhodium ratios, although the C, H, N analyses of " $\text{Rh}(\text{CNCH}_3)_3.38\text{BF}_4$ " and " $\text{Rh}(\text{CNCHCH}_2)_2.7\text{Cl}$ " suggest similar non-integral stoichiometry. Careful preparation of tetrakis(phenylisocyanide)rhodium(I) perchlorate in the absence of oxygen results in a material without partial oxidation. Tetrakis(phenylisocyanide)rhodium(I) chloride, tetrakis(vinylisocyanide)rhodium(I) chloride and perchlorate all appear to have little or no partial oxidation. Deliberate attempts to produce high degrees of partial oxidation by addition of Rh(III) complexes to the rhodium(I) vinylisocyanide complexes have not been successful.

The stoichiometry of $\text{Rh}(\text{CNC}_6\text{H}_5)_3.67(\text{ClO}_4)_1.33$ may reflect its structural details. Extra perchlorate anions may occupy ligand positions within the structure. Partial oxidation may account for the increase in conductivity of this complex with respect to tetrakis(phenylisocyanide)rhodium(I) chloride (Table 2).

Optical properties and crystal structures

Well formed needle-shaped crystals of $[\text{Rh}(\text{CNCHCH}_2)_4]\text{ClO}_4$, $[\text{Rh}(\text{CNC}_6\text{H}_5)_3.67(\text{ClO}_4)_1.33]$, $[\text{Rh}(\text{CNC}_6\text{H}_5)_4]\text{ClO}_4$, $[\text{Rh}(\text{CNC}_6\text{H}_5)_4]\text{BF}_4$, and $[\text{Rh}(\text{CNCH}_3)_3.38]\text{BF}_4$ exhibit strong optical dichroism in plane

polarized light. Maximum extinction occurs when the electric vector is parallel to the needle axis, which in all cases coincides with a unique crystallographic direction (\bar{a}_0). All crystals transmit when the electric vector is perpendicular to \bar{a}_0 , if the crystal is not extremely thick. Crystals of $[\text{Rh}(\text{CNCCHCH}_2)_4]\text{ClO}_4$ transmit red light in the b_0 direction.

The rotation photographs of each of the crystals (rotation axis = \bar{a}_0) showed the same typical set of intense layer lines corresponding to a d spacing of about 2.95 Å, except that the extremely fibrous crystal of $[\text{Rh}(\text{CNC}_6\text{H}_5)_4]\text{ClO}_4$ showed a significantly larger "strong repeat" of 3.20 Å. In contrast, the partially oxidized $[\text{Rh}(\text{CNC}_6\text{H}_5)_{3.67}](\text{ClO}_4)_{1.33}$ showed a well defined pattern of seven pairs of weak layer lines and one pair of strong layer lines corresponding to $d = a_0/8 = 2.98$ Å.

Gunier powder photographs of $[\text{Rh}(\text{CNC}_6\text{H}_5)_4]\text{Cl}$ showed a simple pattern of 11 lines which could be indexed to a tetragonal cell having $c_0/8 = d(\text{Rh-Rh})$ as in $[\text{Rh}(\text{CNC}_6\text{H}_5)_{3.67}](\text{ClO}_4)_{1.33}$. All of the lines could be indexed to a tetragonal cell with $\bar{a}_0 = 20.64$; $c_0 = 23.54$ Å (Table 3c). Since the cell is large, the indexing is questionable, but the four strong 00ℓ reflections (especially, very strong, broad 00^{16} at $2\theta = 62.72$) lend strong support to a Rh-Rh repeat of 2.94 Å.

The Rh-Rh bond lengths in these crystals are considerably shorter than in dimeric rhodium isocyanide complexes (3.19 Å); longer than in rhodium metal (2.69 Å) (20) or in rhodium(0) complexes such as

$[\text{Rh}(\text{CO})(\text{PPh}_3)_2]_2$ (2.63 Å) (21); and similar to the bond lengths in rhodium(II) complexes as $\text{Rh}_2(\text{DMG})_2(\text{PPh}_3)_2 \cdot \text{H}_2\text{O} \cdot \text{C}_3\text{H}_7\text{OH}$ (2.936 Å) (22). Only $[\text{Rh}(\text{CNCHCH}_2)_4]\text{ClO}_4$ crystals showed the diffraction patterns, on oscillation and Weissenberg photographs, of non-fibrous single crystals. The strong layer lines of this crystal were interspaced with two additional, very weak layer lines, indicating that $\bar{a}_0 = 3 \times 2.93 \text{ Å}$ for this compound.

Weissenberg photographs indicate that $[\text{Rh}(\text{CNCHCH}_2)_4]\text{ClO}_4$ forms body centered orthorhombic crystals. All reflections are of the kind $h + k + l = 2n$, and there are no other systematic absences. Thus, the probable space group was determined to be Immm , $\text{Imm}2$, $\text{I}2_12_12_1$, or $\text{I}222$.

Crystal structure of $[\text{Rh}(\text{CNCHCH}_2)_4]\text{ClO}_4$

A three-dimensional Patterson map, calculated with all of the available room temperature diffractometer data, revealed that six rhodium atoms occupy two point sets: 2 Rh in $(000, \frac{111}{222})$ and 4 Rh in $(000, \frac{111}{222}) + (x00, \bar{x}00)$, where $x = 0.33272(27)$. Six perchlorate chlorines also occupy two point sets: 2 Cl in $(0\frac{1}{2}0, \frac{1}{2}0\frac{1}{2})$ and 4 Cl in $(00\frac{1}{2}, \frac{11}{22}0) + (0x0, 0\bar{x}0)$, where $x = \frac{1}{4}$. There are twenty-four ligands of three independent kinds, four of each of two kinds bonded to Rh at $(000, \frac{111}{222})$ and sixteen of one kind bonded to the other set of Rh atoms. The former two ligands have vinyl carbons exhibiting high thermal motion and disorder. Ligand 1 lies along the $(0, y, 0)$

axis and both vinyl carbons have very high thermal motion, indicating that the electron density maximum along the (0y0) axis is the result of an average of multi-fold disorder of atoms about this axis. Since the root-mean-squared amplitude of vibration of these atoms is so high, $\sim 0.4 \text{ \AA}$, the required geometry of the ligand (a C = C-N bond angle of $\sim 120^\circ$) is not contradicted. Ligand 2 lies along the (00z) axis. Its end vinyl carbon atom is disordered across the mirror plane at $y = 0$, and has extremely high thermal motion in the x direction, perhaps reflecting torsional motion of the ligand. This atom was revealed in Fourier maps (Figure 6) and its position was not refined due to proximity to the $z = 0$ mirror plane. Ligand 3 is not disordered, but has fairly high thermal motion. It is bent significantly out of the $x = 1/3$ plane, allowing mirror related ligands on Rh atoms at $(\pm 0.33272, 0.0, 0.0)$ to minimize their steric interactions. The only close ligand-ligand interatomic distances involve isocyanide carbon atoms. Atom positions and temperature factors are collected in Table 5. Bond lengths and angles, as well as some non-bonding contact distances, are shown in Table 6. An ORTEP diagram of the structure is shown in Figure 7. Note that all light atom bond lengths have fairly high standard deviations, due to the dominance of rhodium scattering. Bond lengths and angles associated with the vinyl carbon atoms have especially high standard deviations.

The low intensity of the low temperature data set impeded refinement of the structure. A rapid drop-off in intensity with

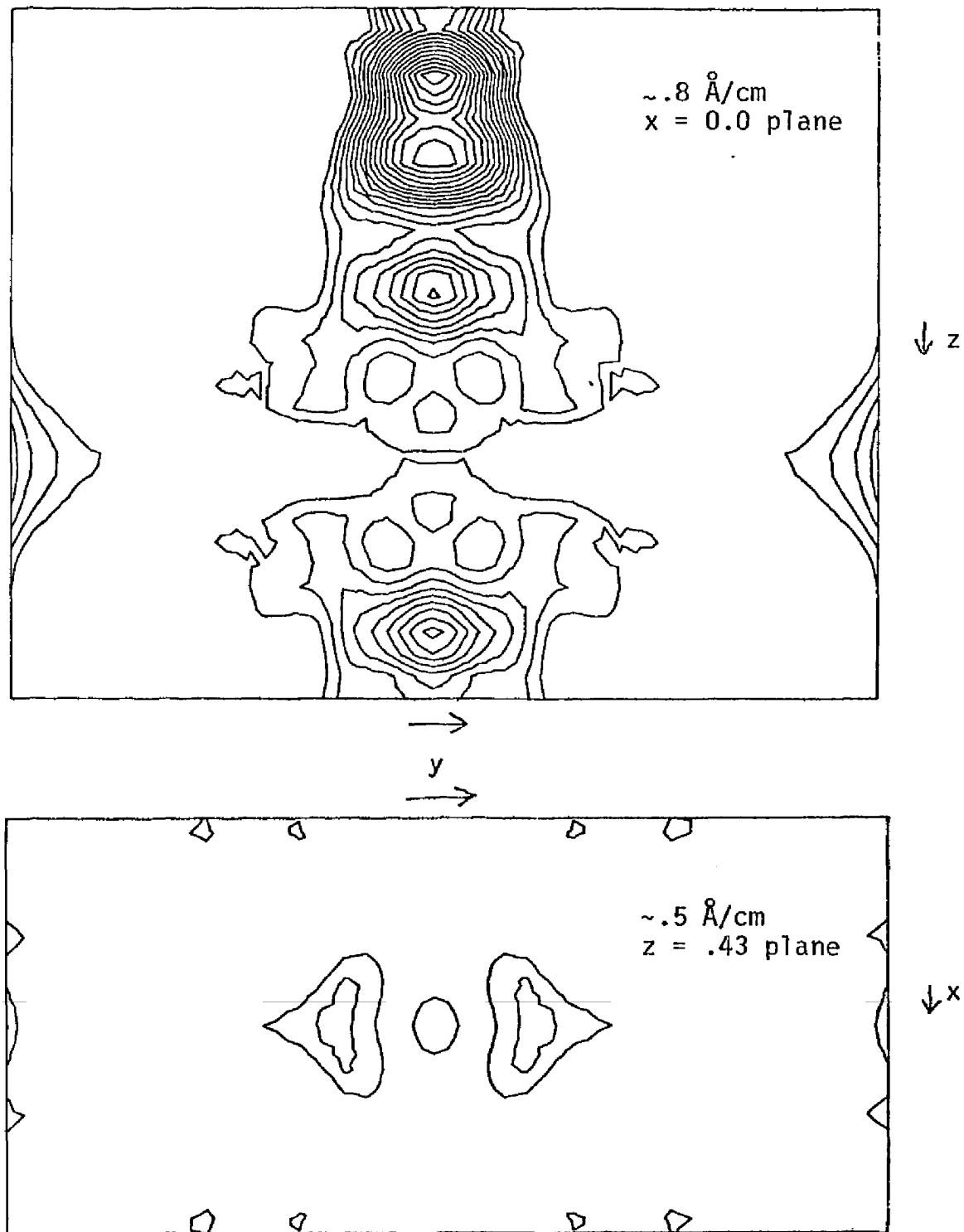


Figure 6. Fourier map of ligand 2, $[\text{Rh}(\text{CNCHCH}_2)_4]\text{ClO}_4$.
Contours at $.2e^-/\text{\AA}^3$ intervals.

Table 5a. Refined Atomic Coordinates for $\text{Rh}(\text{CNCHCH}_2)_4\text{ClO}_4$ at Room Temperature

	x	y	z
Rh 1	0	0	0
Rh 2	33272 (27)	0	0
Cl 1	0	50000	0
Cl 2	0	25000	50000
C 1	0	8954 (134)	0
N 1	0	13490 (113)	0
C 2	0	20425 (286)	0
C 3	0	25026 (354)	0
C 4	0	0	16582 (373)
N 2	0	0	25189 (326)
C 5	0	0	36326 (282)
C 6	0	5000	41651 (635)
C 7	32687 (199)	5936 (67)	11363 (107)
N 3	31173 (176)	9442 (60)	18026 (110)
C 8	28888 (360)	13990 (160)	25897 (242)
C 9	27718 (537)	13591 (267)	34075 (413)

	x	y	z
O1	0	5639	0
O2	0	4787	1084
O3	1348	4788	-539
O4	0	3139	5000
O5	0	2287	6084
O6	1348	2288	4460
O7	0	1861	5000
O8	0	2713	3916
O9	-1347	2712	5540

Oxygen coordinates have been multiplied by 10^4 . All others have been multiplied by 10^5 .

Table 5b. Refined Temperature Factors for $\text{Rh}(\text{CNCHCH}_2)_4\text{ClO}_4$ at Room Temperature.

	U11	U22	U33	U12	U13	U23
Rh 1	592 (22)	962 (33)	974 (40)	0	0	0
Rh 2	568 (11)	798 (14)	859 (18)	0	0	0
C1 1	2590 (298)	4070 (363)	4126 (394)	0	0	0
C1 2	1480 (102)	1736 (103)	4853 (246)	0	0	0
C 1	943 (215)	899 (190)	1767 (349)	0	0	0
N 1	1105 (217)	837 (163)	2865 (434)	0	0	0
C 2	4082	1738 (539)	4203	0	0	0
C 3	3063 (938)	4837	1901 (563)	0	0	0
C 4	312 (118)	1175 (241)	2456 (500)	0	0	0
N 2	1468 (314)	3076 (529)	1886 (427)	0	0	0
C 5	3814 (879)	4256 (937)	508 (188)	0	0	0
C 6	6184	3085 (960)	3048	0	0	-1887 (751)
C 7	653 (68)	1530 (114)	1196 (95)	238 (132)	-48 (134)	-49 (90)
N 3	844 (98)	1795 (117)	1705 (115)	-57 (93)	-35 (95)	-770 (103)
C 8	1726 (270)	3598 (355)	2630 (296)	-273 (237)	-184 (230)	-2235 (290)
C 9	4301 (554)	5815 (780)	5313 (783)	-599 (457)	1714 (533)	-4321 (746)

B

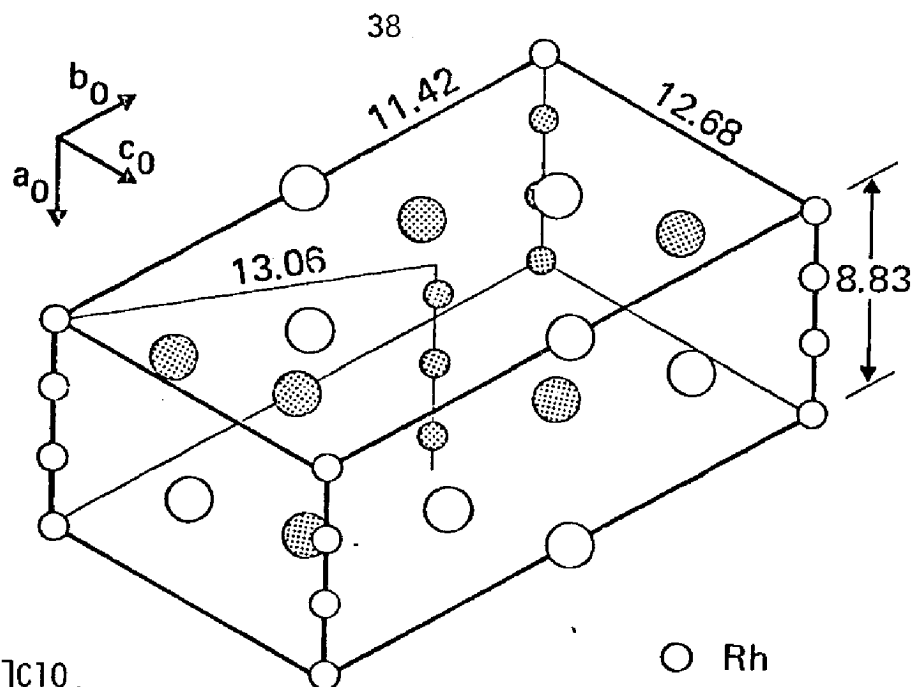
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02	16.00
03	16.00
04	16.00
05	16.00
06	16.00
07	16.00
08	16.00
09	16.00

All U_{ij} 's have been multiplied by 10^4 .

Table 6

$\text{Rh}(\text{CNCHCH}_2)_4\text{ClO}_4$ Room Temperature Structure, Bond Lengths, Angles,
and Contact Distances

Rh 1 - Rh 2	2.9314 (24) Å
Rh 2 - Rh 2'	2.9475 (48) Å
Rh 1 - C 1	2.05 (3) Å
Rh 1 - C 4	2.11 (4) Å
Rh 2 - C 7	1.93 (2) Å
C 1 - N 1	1.04 (4) Å
C 4 - N 2	1.09 (5) Å
C 7 - N 3	1.17 (3) Å
N 1 - C 2	1.58 (6) Å
N 2 - C 5	1.41 (4) Å
N 3 - C 8	1.46 (6) Å
C 2 - C 3	1.05 (12) Å
C 5 - C 6	1.33 (10) Å
C 8 - C 9	1.05 (12) Å
< N 2 - C 5 - C 6	120 (7)°
< Rh2 - C 7 - N 3	175 (2)°
< C 7 - N 3 - C 8	177 (4)°
< N 3 - C 8 - C 9	129 (8)°
C 7 ---- C 7'	3.05 (4) Å
C 7 ---- C 7''	2.71 (2) Å
C 1 ---- C 4	2.94 (3) Å



ORTEP diagram of the carbon chain asymmetric unit. Thermal ellipsoids at 30% probability level.

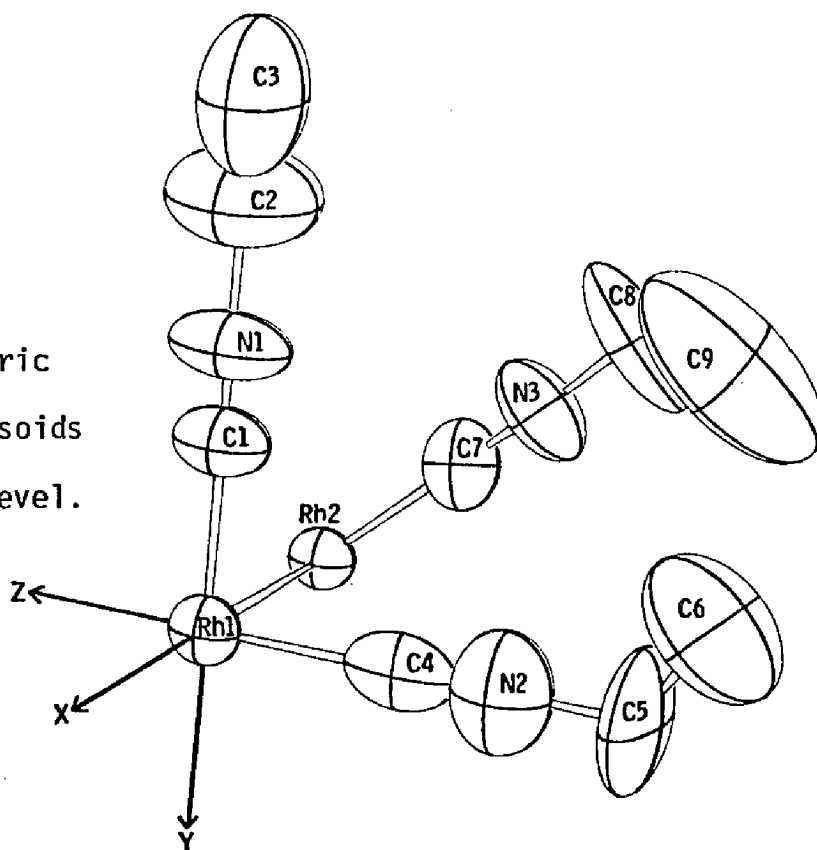


Figure 7. $[\text{Rh}(\text{CNCHCH}_2)_4]\text{ClO}_4$ structure.

increasing k and l remained at low temperature, indicating retention of disorder in y and z parameters. However, $h = 9$ and 12 reflections were observed, reflecting a precisely defined Rh-Rh distance in the x direction. The "high thermal motion" of several atoms at room temperature is resolved into two-fold disorder at 22° K. However, only the rhodium atom coordinates, anisotropic temperature factors, and coordinates of the atoms of ligand 3, may be refined by least-squares. Other atom positions were obtained from Fourier maps. Atom positions and temperature factors are shown in Table 7; bond lengths, and bond angles are collected in Table 8. There is no structural phase transition between 22° K and 300° K in $[\text{Rh}(\text{CNCHCH}_2)_4]\text{ClO}_4$. There is some reduction in thermal motion, but disorder is retained. There is no crystallographic evidence for partial oxidation in $\text{Rh}(\text{CNCHCH}_2)_4\text{ClO}_4$.

Electronic Structure

The electronic properties of crystalline $[\text{Rh}(\text{CNR})_4]^+\text{X}^-$ may be rationalized from simple molecular orbital arguments and a knowledge of the properties of oligomeric species in the solid state and solution (1,2,10). Figure 2 shows a molecular orbital diagram for $[\text{Rh}(\text{CNR})_4]_n^{n+}$ where $n = 1, 2$, and 3, as well as a simple band diagram for the polymeric solid. The dimer and trimer are assumed to have staggered configurations, resulting in D_{4d} and D_{4h} symmetry, respectively. The highest occupied orbital of the monomer is $a_{1g}(dz^2)$; the lowest empty orbital is an a_{2u} orbital with Rh P_z and ligand π^* character.

Table 7a. Atomic coordinates of $\text{Rh}(\text{CNCHCH})_2\text{ClO}_4$ at 22° K.

	x	y	z
Rh 1	0	0	0
Rh 2	33333	0	0
C 1	0	9700	0
C 2	4000	21000	2000
C 3	0	24000	9000
C 4	0	1000	14300
C 5	0	1000	33000
C 6	6000	5000	37000
C 7	32761	6109	12042
C 8	29552	14164	25196
C 9	25326	12772	34295
N 1	0	15700	0
N 2	0	1000	22500
N 3	31422	9248	18062
Cl 1	0	50000	0
Cl 2	0	25000	46000
O1	0	56390	0
O2	0	47870	10840
O3	13480	47880	5400
O4	0	31790	46000
O5	0	23270	56840
O6	13480	23280	40600

All coordinates have been multiplied by 10^5 .

Table 7b. Temperature Factors of $\text{Rh}(\text{CNCHCH}_2)_4\text{ClO}_4$ at 22° K.

	U11	U22	U33	U12	U13	U23
Rh 1	743	1179	746	0	0	0
Rh 2	632	616	1189	0	0	0
C 1	464	791	3047	0	0	0
C 2	B = 10.0					
C 3	B = 10.0					
C 4	1063	1090	1786	0	0	0
C 5	B = 10.0					
C 6	B = 10.0					
C 7	793	2005	1620	584	564	136
C 8	1266	5911	3337	-423	61	-3084
C 9	4468	4744	4748	-801	2386	-3794
N 1	746	1409	3202	0	0	0
N 2	712	3126	1638	0	0	0
N 3	563	2294	1920	-225	-14	-685
Cl 1	1746	8654	8334	0	0	0
Cl 2	1287	1666	6200	0	0	0
O1	B = 16.0					
O2	B = 16.0					
O3	B = 16.0					
O4	B = 16.0					
O5	B = 16.0					
O6	B = 16.0					

All Uij's have been multiplied by 10^4 .

Table 8

$[\text{Rh}(\text{CNCHCH}_2)_4]\text{ClO}_4$ Low Temperature Structure, Bond Lengths, and Angles

Rh 1 - C 1	2.14*	Rh 1 - Rh 2	2.890 (10)
Rh 1 - C 4	1.78*	Rh 2 - Rh 2'	2.890 (10)
Rh 2 - C 7	2.01 (3)		
C 1 - N 1	1.26*		
C 4 - N 2	1.02*		
C 7 - N 3	1.03 (4)		
N 1 - C 2	1.29*		
N 2 - C 5	1.30*		
N 3 - C 8	1.41 (5)		
C 2 - C 3	1.05*		
C 5 - C 6	1.14*		
C 8 - C 9	1.22 (6)		
< C 1 - N 1 - C 2	164° *		
< N 1 - C 2 - C 3	145° *		
< N 2 - C 5 - C 6	116° *		
< Rh2 - C 7 - N 3	175 (3)°		
< C 7 - N 3 - C 8	172 (4)°		
< N 3 - C 8 - C 9	115 (5)°		

*At least one atom position estimated from Fourier map, and not refined by least-squares.

On dimerization, in symmetry D_{4d} , two sets of a and b_2 orbitals result, each of which is primarily associated with either the $a_{1g}(dz^2)$ or $a_{2u}(Pz\pi^*)$ orbitals of the monomer. The low energy transition at 550 nm in water or acetonitrile solutions of $[Rh(CNCHCH_2)_4]ClO_4$ is assigned to the $1b_2 \rightarrow 2a_1$ transition of $[Rh(CNCHCH_2)_4]_2^{2+}$. Likewise, the 715 nm band is assigned to the $2a_{1g} \rightarrow 2a_{2u}$ transition of $[Rh(CNCHCH_2)_4]_3^{3+}$. Oligomerization is allowed because of significant mixing of the $a_{1g}(dz^2)$ and $a_{2u}(Pz\pi^*)$ derived levels of the monomer, stabilizing the lower, filled orbitals of the oligomer, and destabilizing the higher, empty orbitals. If the Rh-Rh interaction in the stacked crystalline modifications is similar in strength to that in the oligomers, a band gap between the dz^2 and $Pz-\pi^*$ bands might be expected to be significantly smaller than lowest energy band of the tetramer at 962 nm ($10,400\text{ cm}^{-1} = 1.3\text{ eV}$). Furthermore, the Rh-Rh bond length in the polymer is apparently significantly shorter than in the oligomers. Photoacoustic spectroscopy of $[Rh_2(CNC_6H_5)_8][B(C_6H_5)_4]_2$ in the solid state reveals absorption at about the same wavelength as the $1b_2 \rightarrow 2a_1$ band in solution. The Rh-Rh bond length in $[Rh_2(CNC_6H_5)_8][B(C_6H_5)_4]_2$, from refinement of the crystal structure, is 3.19 Å. The significant shortening of the Rh-Rh bond in $[Rh(CNCHCH_2)_4]ClO_4$ must lead to a much stronger overlap of dz^2 and $Pz\pi^*$ orbitals in the adjacent monomeric units, and a much lower band gap than would be predicted from the oligomer spectra.

The photoacoustic spectra of $[Rh(CNC_6H_5)_{3.67}](ClO_4)_{1.33}$ and $[Rh(CNCHCH_2)_4]ClO_4$ have strong absorption throughout the visible

region. This is consistent with strong interactions resulting in a band structure, with activation energy <1 eV for excitation into the conduction band.

The room temperature conductivity of $[\text{Rh}(\text{CNCHCH}_2)_4]\text{ClO}_4$ is $\sim 2 \Omega^{-1} \text{ cm}^{-1}$ as measured on single crystals. The plot of $\ln \sigma$ vs. $1/T$ (Appendix 2) reveals that the conductivity is activated, $\Delta E \sim 0.10$ eV in the linear low temperature region. This electrical behavior is consistent with a small band gap between the dz^2 and $\text{Pz}-\pi^*$ bands, and conduction via thermal population of $\text{Pz}\pi^*$ band. Disorder may play a role in increasing the conductivity.

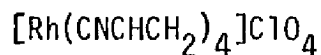
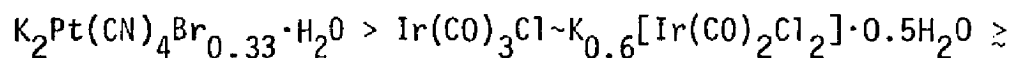
The transmission of polarized light, with electric vector perpendicular to \bar{a}_0 , through single crystals of $[\text{Rh}(\text{CNCHCH}_2)_4]\text{ClO}_4$ in the \bar{b}_0 direction, indicates that the activation energy for hopping between chains is very high (≥ 1.5 eV). Although crystals are far too thin for conductivity measurements with the Montgomery configuration, it is certain that these complexes are among the most anisotropic of known quasi-one-dimensional conductors.

Powder conductivities increase with decrease in ligand size. This is presumably due to greater interchain coupling when rhodium chains are close together.

Conclusions

Tetrakis(isocyanide)rhodium(I) salts crystallize in conducting, stacked modifications if ligands and anions are reasonably small (CN-R: R = methyl, vinyl or phenyl; x^- : Cl^- , BF_4^- , ClO_4^-). Partial oxidation of the rhodium chain is likely in some of the complexes, for example $[Rh(CNC_6H_5)_3(CNC_6H_5)](ClO_4)_4$. However, a non-integral average oxidation state does not seem to be a necessary condition for moderately high conductivity (11). Chemical analysis and refinement of the room temperature crystal structure of $[Rh(CNCHCH_2)_4]ClO_4$ indicate that this complex has one perchlorate anion per rhodium atom. If a very low level of partial oxidation exists, it may effect the magnitude of the conductivity, but would be present as a crystallographic impurity and no metallic state would exist. A minor difference in two Rh-Rh bond lengths occurs at room temperature and probably at low temperature. This inequality, along with the differences in the chemical environments of the two crystallographically inequivalent rhodium atoms, seems to be due to steric interactions (i.e., steric repulsion, between the eclipsed sets of ligands on the equivalent Rh atoms at $x \approx 1/3$ and $x \approx 2/3$, results in displacement from the ideal positions and lengthening of the Rh2-Rh2' bond). Both integral oxidation state and crystallographic inequivalence of metal atoms have been observed separately in conducting, stacked d^8 metal complexes (11,23). The conditions for conductivity in these materials seem to be less restrictive than previously thought.

Non-integral oxidation has been described as a necessary condition for high conductivity in d^8 metal complexes (5). The conductivities of complexes exhibiting both integral and non-integral oxidation states have similar dependence on temperature. Magnitudes of the room temperature conductivities of the rhodium isocyanide complexes are ~100 times less than the partially oxidized platinum complexes, while the activation energies are similar. The decrease of room temperature conductivities in the sequence:



follows the increase in interchain distances.

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CHAPTER 3

Structural Phase Transition and Disorder in $(\text{TTF})(\text{Cl})_x$

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Abstract

(TTF)Cl_x, $x = 0.67$ and 0.70 , is a quasi-one-dimensional organic conductor with a room temperature conductivity of $\sim 150 \Omega^{-1} \text{ cm}^{-1}$. At room temperature the structure is tetragonal and consists of chains of uniformly spaced, eclipsed TTF molecules surrounding channels occupied by chloride ions, which form a disordered lattice. The chloride sublattice becomes ordered and the TTF sublattice undergoes a structural phase transition from tetragonal to monoclinic symmetry at $\sim 250^\circ \text{ K}$. The angle β of the monoclinic crystal increases continuously as the temperature is decreased from 245° K to 19° K . The electrical conductivity shows a sharp decrease at the phase transition which is suggestive of the formation of commensurate charge density waves in the monoclinic crystal.

Introduction

The stabilization of the metallic state in quasi-one-dimensional organic metals has been a goal of researchers for many years. These materials exhibit metallic behavior down to a temperature, $T_{\text{MI}} \sim 50 - 200^\circ \text{ K}$, where a metal-to-nonmetal (M-NM) transition takes place, resulting in non-metallic behavior at low temperature. It is important to understand the nature of the forces that drive this transition in order that it may be controlled, and eventually suppressed, so as to stabilize the metallic state. Some of the experimentally controllable parameters which should influence the phase transition are the interchain coupling, the degree of disorder, and the nature of the molecular stacking along the chain (i.e., slipped versus eclipsed stacking).

There have been numerous studies of the M-NM phase transition in organic metals in which the cation and anion radicals stack in a slipped fashion.¹ However, very little work has been done on systems such as the halides and pseudohalides, (i.e., thiocyanate, SCN and selenocyanate SeCN) of tetrathiafulvalene (TTF)^{2,3,4}, in which the TTF molecules in the chain are eclipsed. In these materials the metallic state appears to be less stable (i.e., higher T_{MI}). Recent X-ray diffuse scattering studies of $(TTF)_{12}(SCN)_7$ reveal a tetragonal-to-monoclinic structural phase transition at $\sim 340^\circ$ K and the presence of one-dimensional incommensurate charge density waves (CDW) below this temperature.⁵ In this paper, we report the results of our single-crystal X-ray study of the crystal structure of $(TTF)Cl_x$ ($x = 0.67$ and 0.70) at various temperatures down to 19° K. Here, the TTF molecules are eclipsed and uniformly stacked and the chloride lattice exhibits considerable structural disorder at room temperature.⁶ The electrical conductivity and thermoelectric power undergo changes at 200 - 250° K, which are suggestive of a M-NM transition. Similar changes are observed in all of the halides and pseudohalides of TTF.

Experimental

Crystals of $(TTF)Cl_x$ were grown by co-diffusion of solutions of $(TTF)(ClO_4)_{0.7}$, and tetraethylammonium chloride in acetonitrile. The crystal structure was studied by a variety of X-ray diffraction techniques. Oscillation and Weissenberg photographs were obtained at room temperature. In addition, oscillation photographs were obtained at $120 \pm 10^\circ$ K by cooling with a stream of cold N_2 gas.

The room-temperature structure of $(TTF)Cl_{0.67}$ was determined and refined with the use of a three-dimensional data set that was obtained

with nickel-filtered $\text{CuK}\alpha$ radiation ($\lambda = 1.54178 \text{ \AA}$) and a Datex automated, locally modified General Electric quarter-circle diffractometer. The crystal used was a tetragonal prism of the size $0.11 \times 0.11 \times 0.63 \text{ mm}^3$, the prism axis being \underline{c}_0 . A total of about 800 reflections were measured up to $2\theta = 155^\circ$, using $2\theta - \theta$ scans. These were corrected for Lorentz, polarization, and absorption ($\mu = 104 \text{ cm}^{-1}$), and merged to a set of 303 unique reflections.

For the study of structural changes at temperatures ranging from 300° K to 19° K we used a locally designed and built low-temperature diffractometer⁷ consisting of a CTI model 21 closed-cycle "Cryocooler", an E & A full-circle and base goniometer, a Syntex $\text{P}\bar{1}$ interface (and software package), a graphite monochromator of our own design, and a molybdenum-target X-ray tube. The temperature can be varied in steps of 0.1° K and kept constant for long periods of time. The crystal used for this study $[(\text{TTF})\text{Cl}_{0.67}]$ was from the same batch as that used for the room-temperature work and had the dimensions $0.17 \times 0.20 \times 0.27 \text{ mm}^3$. It was mounted with the \underline{c}_0 -axis approximately parallel to the rotation axis.

The electrical conductivities were measured by the standard four-probe technique using aquadag contacts.

Results

General Features of the Layer Lines and Stoichiometry

A number of batches of good crystals were obtained by the co-diffusion method. The two kinds of crystals studied in detail represent the extremes of the short range of stoichiometry exhibited by disordered tetragonal $(\text{TTF})\text{Cl}_x$.

Figure 1 shows an oscillation photograph taken at room temperature of a crystal from one of the batches, which we label $(\text{TTF})\text{Cl}_{0.70}$. The rotation axis is \underline{c}_0 . It is seen that there are two sets of diffuse layer lines

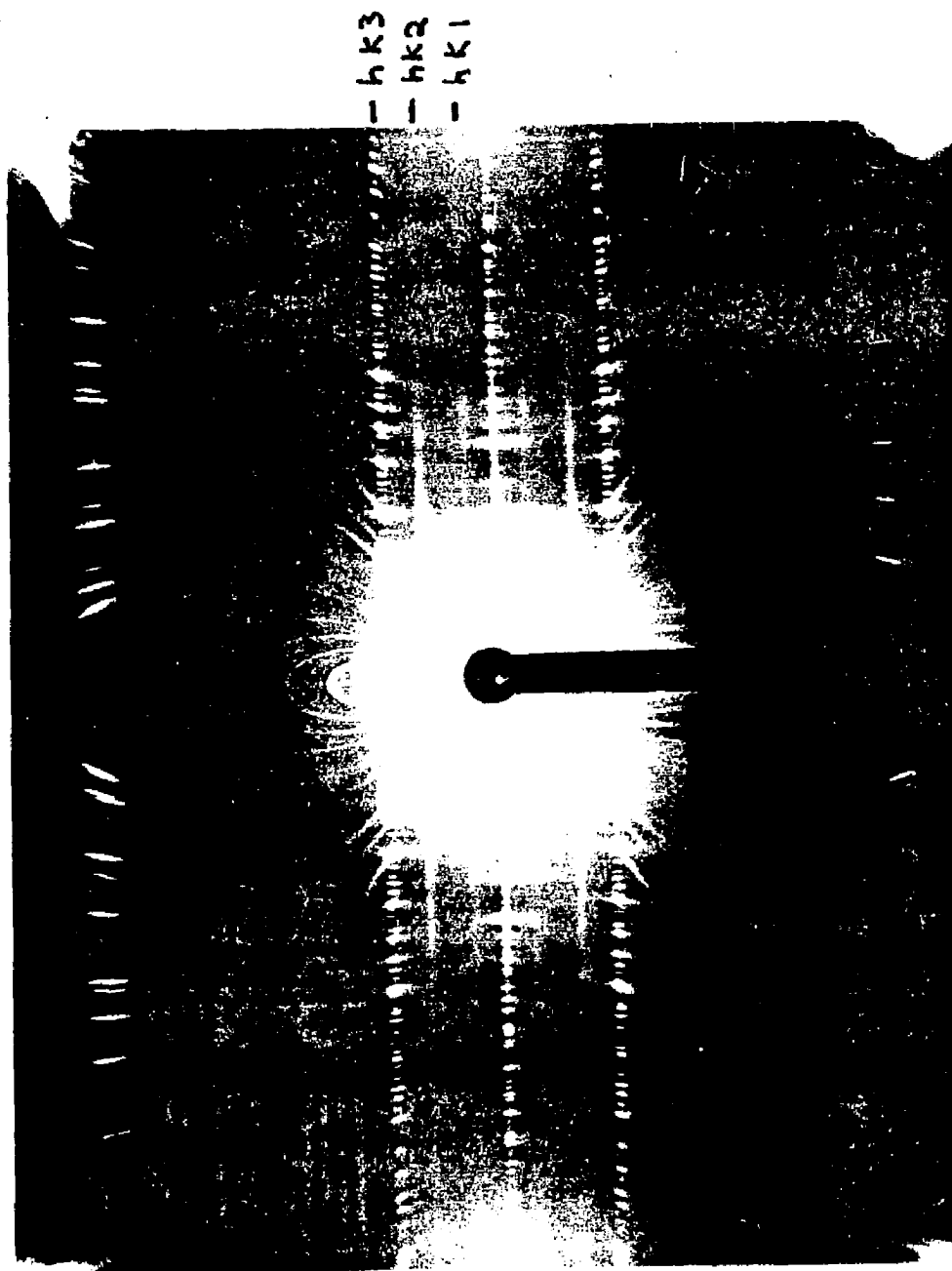


Figure 1. Oscillation photograph of (TTF)Cl_{0.67}.

marked hkl and hk2. Single crystals from the other batch which we label (TTF)Cl_{0.67} (again, rotation axis is \underline{c}_0) gave rise to two similar sets of diffuse lines, which, however, differ somewhat in the spacings. In each case the hkl set corresponds to the d-spacing $d_{001} \sim 3.0 \times \underline{c}_0$ and the hk2 set to $d_{002} \sim 1.5 \times \underline{c}_0$, where \underline{c}_0 is the length of the cell edge that corresponds to d_{003} or the (non-diffuse) hk3 set (TTF sublattice). Each hk2 set is due solely to the chloride ions in the channels (Cl-sublattice) and the weaker hkl set results from both the Cl-sublattice and the TTF-sublattice. If the assumption is made that d_{002} is the same as the average Cl-Cl distance along z and that d_{003} represents the average TTF-TTF distance, the stoichiometry should correspond to the ratio d_{003}/d_{002} . In the absence of chemical analyses of sufficient accuracy to determine the minute differences in composition of the two batches, each label used above was assigned so as to represent the d_{003}/d_{002} ratio as determined from the corresponding oscillation photograph; see Table 1. The nominal camera radius was used for the determination of each d-spacing without applying a correction for film shrinkage. Thus, each d-spacing by itself may lack accuracy, but for the practical purpose considered here errors due to film shrinkage cancel out in the ratios. The stoichiometries 1:0.70 and 1:0.67 are consistent with the composition range previously reported⁶ for the disordered tetragonal (TTF)Cl_x.

In (TTF)Cl_{0.67} the chloride sublattice is commensurate with the TTF lattice whereas in (TTF)Cl_{0.70} the two lattices are incommensurate with each other. The diffuseness of the hkl and hk2 layer lines indicates that there is a considerable degree of disorder in the chloride

Table 1. Repeat distances in the TTF and Cl Stacks

Phase	Temperature (°K)	d_{TTF} (Å)	d_{Cl} (Å)	$d_{\text{TTF}}/d_{\text{Cl}}$
TTF Cl _{0.67}	298	3.57	5.32	0.671
TTF Cl _{0.67}	120	3.54	5.36	0.660
TTF Cl _{0.70}	298	3.58	5.08	0.705
TTF Cl _{0.70}	120	3.57	5.36;	0.666;
			4.97	0.718

sublattices of each compound at room temperature. The diffuse nature of the spots which make up the $hk1$ and $hk2$ layer lines, indicates that only very short-range order exists in the chloride sublattice.

The features of the oscillation photographs just discussed are in general similar to those that we have observed on photographs of $(TTF)Br_{0.74}$, $(TTF)_{12}(SCN)_7$, and $(TTF)_{12}(SeCN)_7$, except that in the case of $(TTF)Cl_x$, the diffuseness of the anion-sublattice reflections is considerably enhanced.

Refinement of the Room-Temperature Structure of $(TTF)Cl_{0.67}$

Weissenberg photographs of $(TTF)Cl_{0.67}$ showed that the structure is tetragonal with Laue symmetry $4/\underline{mmm}$. All $Ok\ell$ reflections were of the type $k + \ell = 2n$ and no other conditions for reflection were detectable. Thus, the probable space groups are $\underline{P4}_2/\underline{nm}$, $\underline{P4n}2$, and $\underline{P4}_2 \underline{mm}$, of which the last one has all the necessary properties to incorporate the flat TTF molecules in columns parallel to the c -axis and leaving channels parallel to these for the chlorides. The lengths of the edges of the smallest unit cell (TTF -lattice) are $\underline{a}_0 = \underline{b}_0 = 11.1931(7) \text{ \AA}$ and $\underline{c}_0 = 3.6002(2) \text{ \AA}$, as determined by a least-squares fit of 2θ values for 14 reflections measured with a diffractometer which was carefully calibrated for $2\theta_0$.

The trial structure consisted of two TTF molecules, with mmm symmetry, which are centered at $(0,0,0; \text{etc.})$, with \bar{c}_0 perpendicular to the molecular plane. The chloride ion and hydrogen atom were left out at this stage. A nearly uniform column of electron density at $x = 1/2, y = 0$ was observed on Fourier maps, and was ascribed to the highly disordered chloride ion. A chloride was included in the model structure; it was placed corresponding to the maximum in electron density of the column, at $(1/2, 0, 1/4)$.

This trial structure was refined first isotropically and then anisotropically by full-matrix least-squares minimization of $\sum w(F_o^2 - F_c^2)$ and weights equal to $1/\sigma^2(F_o)^2 + t$, where t is a term accounting for errors other than counting statistics (CRYM program). The scattering factors of sulfur and chloride were corrected for the real part of anomalous dispersion; the imaginary component was ignored for this centrosymmetric space group.

Inclusion of a population factor for chloride as refinable parameter in the full matrix in some of the refinement cycles led to inconclusive results because of the high standard deviation of that parameter. Also, the chloride anisotropic temperature factor in the z direction, U_{33} , could not be refined. The Fourier section through the chloride channels ranging from $z = 0$ to $z = 1/2$ showed a variation in electron density of merely about 20% over that range, the minima being at $z = 0$ and $z = 1/2$ and the maximum at $z = 1/4$; see Fig. 3. Thus, the refinement was continued with 1.34 chlorides distributed over the two point sets $4d(1/2, 0, 1/4; \text{etc.})$ and $4c(1/2, 0, 0; \text{etc.})$ and with the inclusion of population factors P_d and P_c as refinable parameters that were constrained to $P_c = 1 - P_d$. These population parameters could be refined adequately. The anisotropic temperature factor of the chloride ions still showed high standard deviations and oscillated after partial refinement. The chloride temperature factors were held constant in the final stages of refinement, when the hydrogen atom, positioned by geometrical considerations, and an additional parameter to account for secondary extinction and counting losses, were included in the least-squares calculation. The final

agreement index obtained was $R = 0.062$ for all 299 reflections with $F_o > 0$ and $R = 0.057$ for 271 reflections with $F_o \geq 3\sigma$. The goodness-of-fit $[\sum w(F_o^2 - F_c^2)/n-p]^{1/2}$ for $n = 299$ observations and $p = 30$ refinable parameters (extinction included) was 4.39. This agreement is very satisfactory in view of the complications caused by the disorder of the chloride ions.

Room-Temperature Structure of (TTF)Cl_{0.67}

The refined atomic coordinates and the occupancies of the two point sets 4c and 4d by the 1.34 chloride ions are given in Table 2. The anisotropic temperature factors are listed in Table 3 together with the isotropic one for hydrogen.

Table 4 gives the molecular dimensions of the TTF species found in four refined structures including that of (TTF)Cl_{0.67}. All distances and angles (column headings a, b...etc. and α , β ,... etc) as identified in the drawing at the bottom of that table are averaged over the mmm molecular symmetry observed for (TTF)Cl_{0.67}. Our estimated standard deviations for (TTF)Cl_{0.67} are in the range of 0.009 - 0.011 Å (C-C), 0.005 - 0.006 Å (C-S) and 0.4 - 0.8 degrees (bond angles).

An interesting trend exhibited in Table 4 is the gradual elongation of the C=C bond labeled a and the shortening of the C-S bonds labeled b as the anticipated charge transfer (column e) increases. It is seen that the central C=C bond length (column a) is most sensitive to charge transfer. Some reservation seems appropriate as regards the degree of charge transfer in (TTF)HgCl₃ because HgCl₃⁻ species are not clearly identifiable in

Table 2. The refined positional parameters

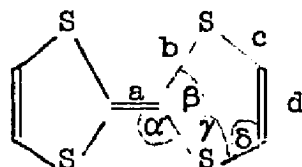
Kind of Atom	Occu- pancy	Point Set	x	y	z
C(1)	4	4f	0.04367(42)	0.04367(42)	0
C(2)	8	8i	0.24007(48)	0.15649(49)	0
S	8	8i	0.19364(12)	0.01008(11)	0
H	8	8i	0.3147(44)	0.1759(45)	0
Cl(1)	0.73(2)	4d	1/2	0	1/4
Cl(2)	0.61(2)	4c	1/2	0	0

Table 3. Refined anisotropic temperature factors for (TTF)Cl_{0.67}

Kind of Atom	U ₁₁	U ₂₂	U ₃₃	U ₁₂
C(1)	0.0653(10)	0.0653(10)	0.0468(32)	0.0012
C(2)	0.0781(31)	0.0987(36)	0.0736(35)	-0.0224(30)
S	0.0644(8)	0.0824(9)	0.0657(9)	0.0015(5)
Cl(1)	0.1051	0.1012	0.1069	-0.0051
Cl(2)	0.1153	0.1183	0.0678	0.0079
H	B(isotropic) = 6.00 Å ²			

Table 4. Dimensions of TTF cations in four refined structures

Type of Compound	Ref.	a	b	c	d	α	β	γ	δ	e
(TTF) ^o	10	1.349	1.757	1.726	1.314	122.7	114.5	94.3	118.6	0.
(TTF)(TCNQ)	11,24	1.369	1.743	1.736	1.323	122.6	114.7	94.9	117.7	~0.
(TTF)Cl _{0.67}	present paper	1.383	1.720	1.719	1.323	122.4	115.3	95.0	117.4	~0.
(TTF)HgCl ₃	11*	$\left\{ \begin{array}{l} 1.40 \\ 1.41 \\ 1.40 \end{array} \right.$	$\left\{ \begin{array}{l} 1.71 \\ 1.72 \\ 1.72 \end{array} \right.$	$\left\{ \begin{array}{l} 1.72 \\ 1.72 \\ 1.70 \end{array} \right.$	$\left\{ \begin{array}{l} 1.33 \\ 1.30 \\ 1.28 \end{array} \right.$	$\left\{ \begin{array}{l} 122.3 \\ 122.2 \\ 122.5 \end{array} \right.$	$\left\{ \begin{array}{l} 115.5 \\ 115.7 \\ 115.0 \end{array} \right.$	$\left\{ \begin{array}{l} 94.7 \\ 94.4 \\ 94.1 \end{array} \right.$	$\left\{ \begin{array}{l} 117.5 \\ 117.8 \\ 118.4 \end{array} \right.$	~1.



*Bond lengths and angles from Kistenmacher, private communication.

that structure. Nevertheless, the overall trend in the data is indicative of a charge transfer of nearly one electron.

Figure 2 shows a projection of the structure onto the (001) plane. The TTF cations stack in an eclipsed fashion with a uniform spacing of 3.6002 Å and form segregated columns along [00z] and [1/2,1/2,z]. The chloride ions reside in channels along [0,1/2,z] and [1/2,0,z] between the TTF chains. The electron-density map, through such a channel parallel to the channel axis as shown in Fig. 3, clearly exhibits the high disorder of the chloride ions as do the diffuse layer lines shown in Fig. 1.

Effect of Lowered Temperature on the Chloride Sublattice

1. $(\text{TTF})\text{Cl}_{0.67}$: Oscillation photographs taken at ~120° K revealed considerable sharpening of the diffuse layer lines and the emergence of weak but sharp Bragg reflections superimposed on these. A more detailed study at various temperatures and a cooling rate of 1° K/hour with the use of our low-temperature diffractometer⁷ revealed distinct effects close to 250° K. Here, the reflections with $\ell = 1, 2$, and 4, associated with the chloride sublattice, increased substantially in sharpness and integrated intensity as is shown in the graph, Fig. 5a; the increase occurs over a relatively narrow temperature range ($\Delta T \sim 20^\circ \text{ K}$). All of these reflections remain weak in comparison to the TTF-sublattice reflections ($\ell = 3n$). Throughout this temperature range the chloride sublattice remains commensurate with the TTF sublattice. Below about 250° K the changes in the integrated intensities of the chloride-sublattice reflections ($\ell = 1, 2$, and 4) are less distinct (and

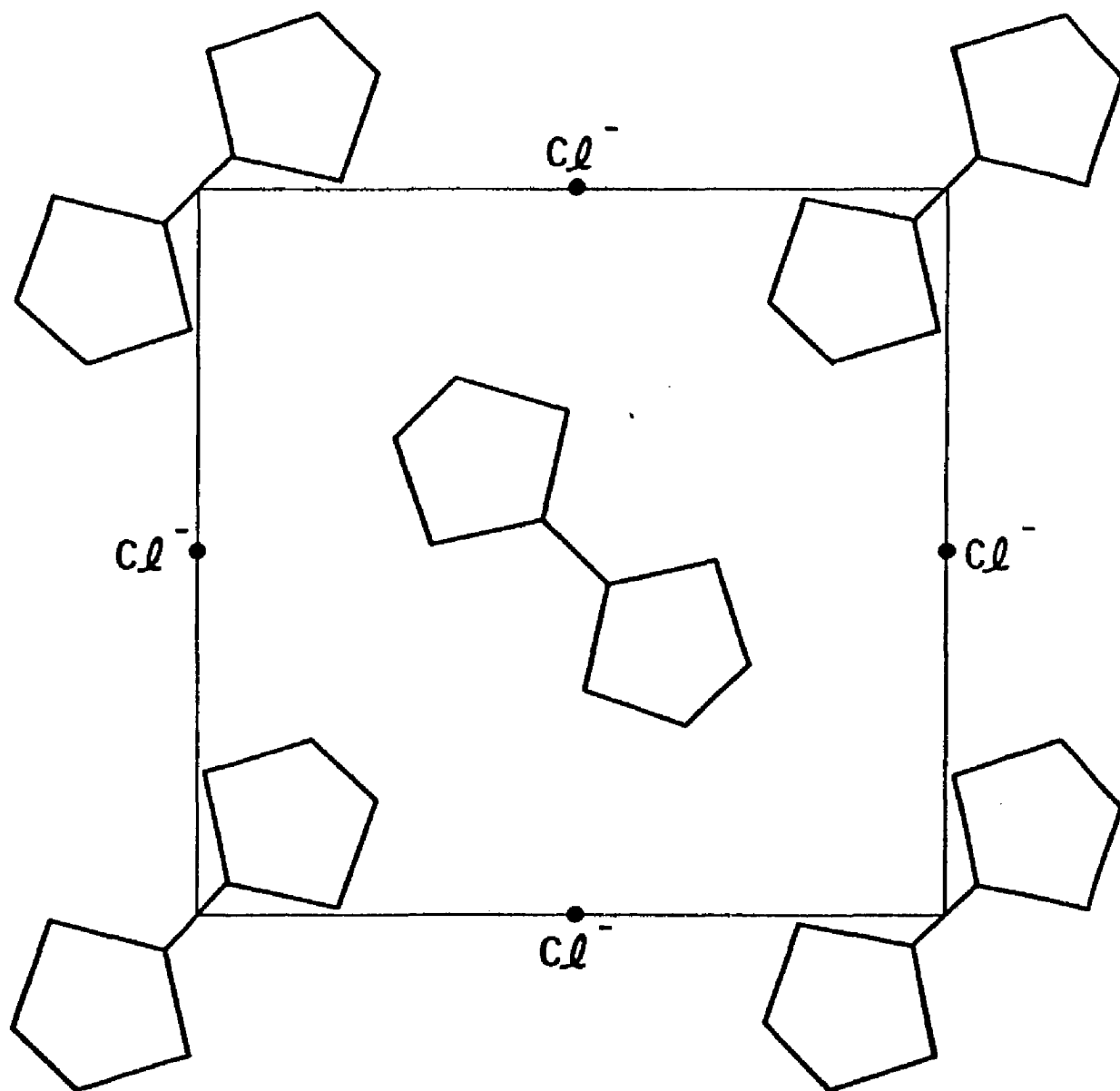


Figure 2. Projection of the $(\text{TTF})\text{Cl}_{0.67}$ room temperature structure onto the (001) plane.

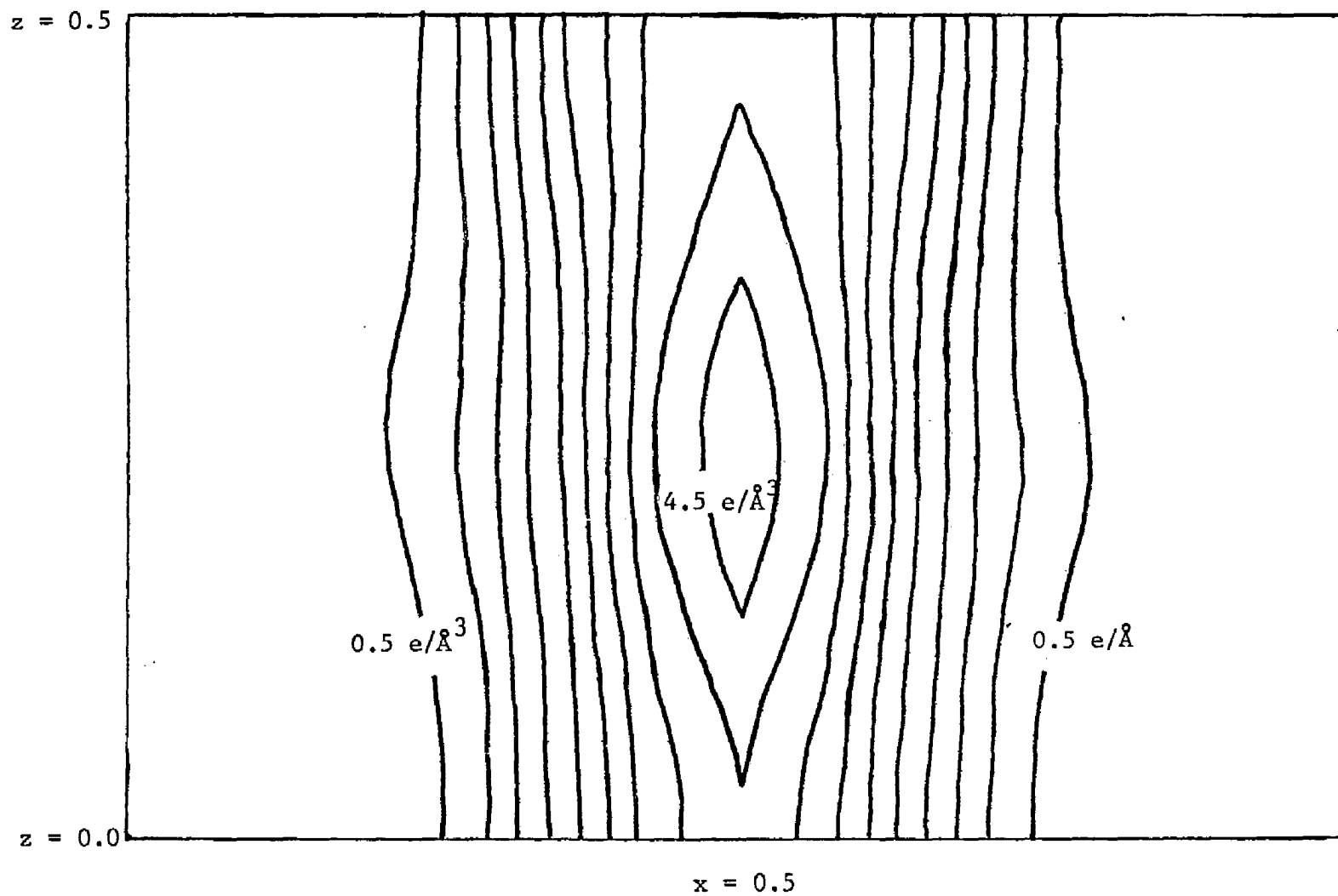


Figure 3. Electron density map of chloride ion channel in $(\text{TTF})\text{Cl}_{0.67}$. Scale is about $0.15 \text{ \AA}/\text{cm}$.

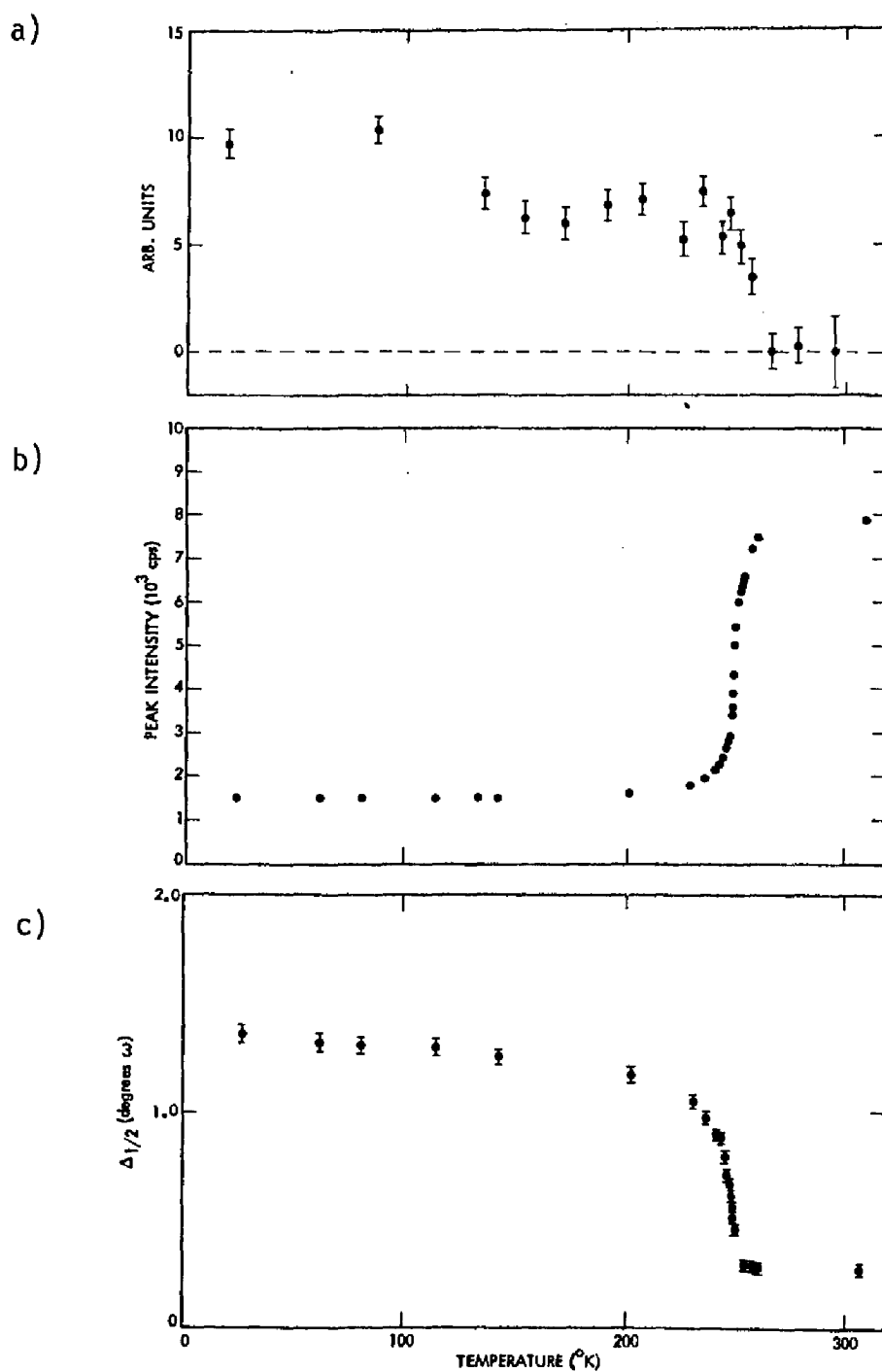


Figure 4. Temperature dependence of: a) Integrated intensity of 114 reflection; b) Peak intensity of 330 reflection; c) Full width at half-height of 330 reflection of TTF Cl_{0.67}. a) at 1° K/hr, b) and c) at 10° K/hr.

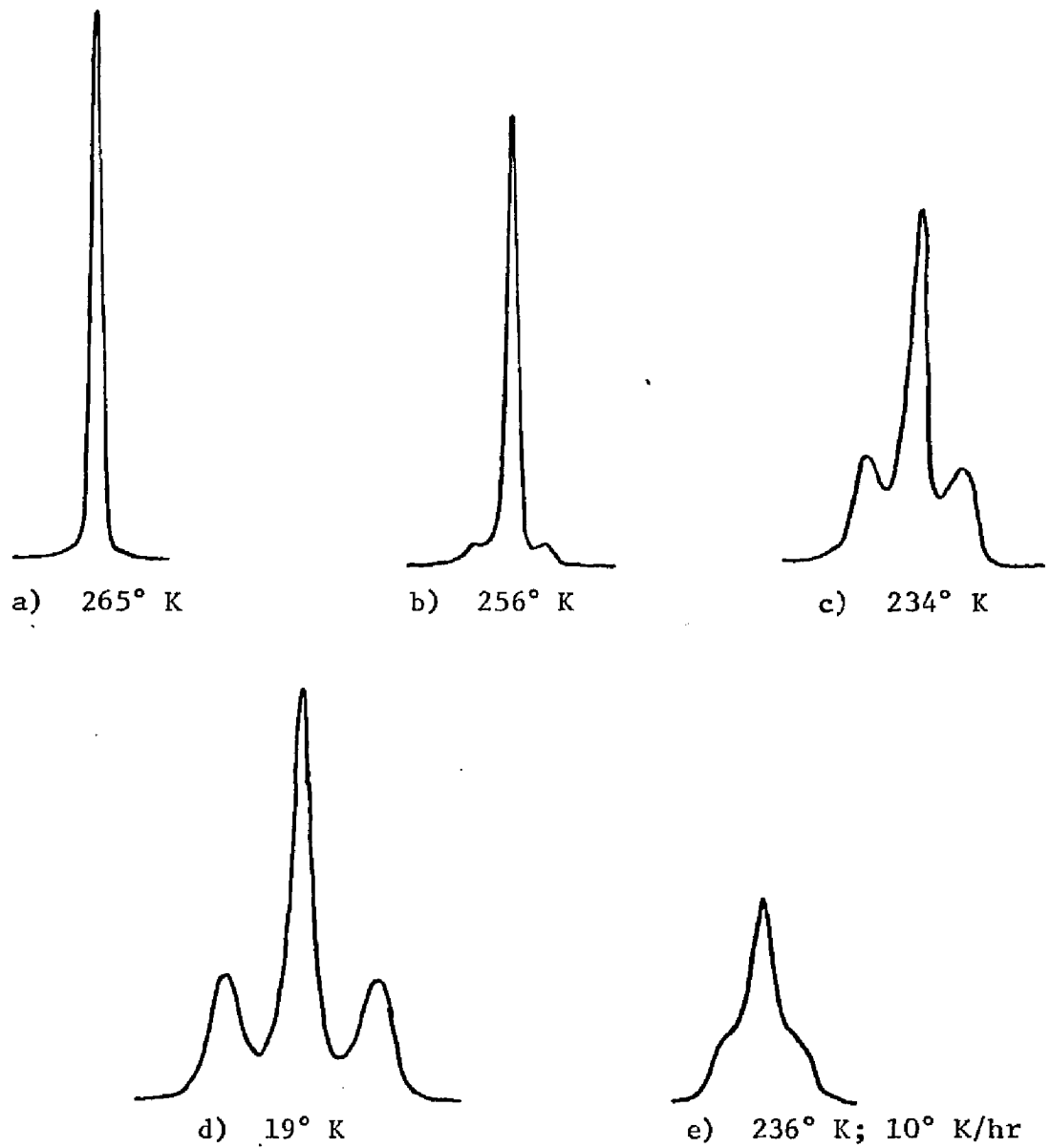


Figure 5. Peak profiles of 330 reflections (w scan) at various temperatures. Profiles a) - d) taken with cooling rate of 1° K/hour.

there is a smooth increase in the region between about 90° K and 150° K; Fig. 4a).

The relatively sharp increase in integrated intensity close to 250° K reflects ordering in the chloride sublattice, which occurs over a relatively narrow temperature range. The extremely high disorder of the chloride lattice at room temperature, together with low-temperature ordering, suggests that dynamic disorder is present at room temperature, and disappears at the phase transition.

2. (TTF)Cl_{0.70}: These crystals behave in a different manner on cooling. Each of the diffuse layer lines corresponding to hkl and hk2 on the room-temperature photograph splits at low temperature into a pair of closely-spaced sharper layer lines on which weak diffuse spots are superimposed. The temperature at which this splitting occurs is somewhere above 120° K. (The low-temperature diffractometer was not used for this study.) Thus, there are now two chloride sublattices, one corresponding to (TTF)₃Cl₂ [or (TTF)Cl_{0.667}] and the other to (TTF)₇Cl₅ [or (TTF)Cl_{0.714}], and each one is commensurate with the same TTF sublattice. The stoichiometries as determined from the oscillation photographs are (TTF)Cl_{0.666} and (TTF)Cl_{0.718} respectively. Thus, upon cooling through 250° K, the highly disordered (TTF)Cl_{0.70} crystal incorporates two kinds of domains, one representing (TTF)₃Cl₂ and the other (TTF)₇Cl₅. Within each domain the chloride lattice is commensurate with the TTF lattice. A phase of stoichiometry 7/5 is also observed in the TTF-iodide system, (TTF)₇I₅.^{2,13}

Effect of Lowered Temperature on the TTF Sublattice of $(\text{TTF})\text{Cl}_{0.67}$

All studies of $(\text{TTF})\text{Cl}_{0.67}$ at low temperature were carried out using the same crystal. Annealing was rapid and complete at room temperature, and there was no permanent damage to the room-temperature single crystal, even after repeated thermal cycling.

Changes in peak intensity and profile of the TTF sublattice reflections were monitored at various temperatures at cooling rates of $\sim 1^\circ \text{ K/hour}$, $\sim 10^\circ \text{ K/hour}$, and $\sim 100^\circ \text{ K/hour}$. In the temperature range $\sim 295^\circ \text{ K}$ to $\sim 250^\circ \text{ K}$ all changes were independent of cooling rate.

Figure 4b shows the temperature dependence of the peak intensity of the 330 reflections and Fig. 4c the change in full width at half maximum observed in omega scans of the same reflection. The cooling rate was 10° K/hour in both cases. Faster cooling ($\sim 100^\circ \text{ K/hour}$) results in a greater decrease in peak intensity, and a greater increase in the width of the profile. Slower cooling ($\sim 1^\circ \text{ K/hour}$) results in resolution of several components of the peak profile. Figure 5 shows omega-scan profiles of the same reflection at: (a) 265° K , (b) 256° K , (c) 134° K , and (d) 19° K . Figure 5e shows the effect of faster cooling (10° K/hour) on the scan profile [compare with Fig. 5c].

It is seen (Fig. 5) that peak splitting occurs as the temperature is lowered, the onset being at about 256° K (Fig. 5b). From here on the two smaller peaks in each profile increase in intensity at the expense of the larger central one. Also, the separation between the individual peaks of each profile increases as the temperature is lowered. At 153° K each component was resolved sufficiently to be measured individually with the diffractometer.

A least-squares fit of the 2θ values of the central peak of 14 profiles gave the unit-cell parameters of the tetragonal phase at 153° K; $a_o = 11.126(2) \text{ \AA}$ and $c_o = 3.5768(4) \text{ \AA}$. An analysis and least-squares fit of the 2θ , ω , ϕ , and χ values of 13 sufficiently strong and well resolved "side peaks" revealed that they originate from a monoclinic structure with unit-cell parameters $a_o = 15.742(56) \text{ \AA}$, $b_o = 15.721(38) \text{ \AA}$, $c_o = 10.744(21) \text{ \AA}$, and $\beta = 92.98(23)$ degrees. Seven of the 13 side peaks could be ascribed to one of four possible orientations of the cell of the monoclinic phase with respect to the tetragonal cell; the remaining six corresponded to a second orientation. Within experimental error, the cells of the two phases are related to each other by $a_{\text{mono}} = a_{\text{tetr}} \cdot \sqrt{2}$ and $c_{\text{mono}} = 3 c_{\text{tetr}}$.

The four directions $[110]$, $[\bar{1}10]$, $[1\bar{1}0]$, and $[\bar{1}\bar{1}0]$ of the tetragonal cell are equivalent. Increasing the angle, between one of these directions and $c_{o\text{tetr}}$, from 90° to β , results in a monoclinic cell. Since the four directions are equivalent, four orientations of the monoclinic cell with respect to the tetragonal cell (and crystal) can result with equal probability. Because part of the crystal retains the tetragonal structure, the result of the phase transition is a quintuple twin containing domains of the tetragonal phase and domains of the monoclinic phase having four possible orientations.

The relationship between the monoclinic a, b plane and the tetragonal (001) plane is shown in Fig. 6. The monoclinic angle β increases as the temperature is lowered and reaches 93.6 degrees at 19° K. At 247° K, β is 91.8 degrees. Figure 7 shows a plot of β versus temperature.

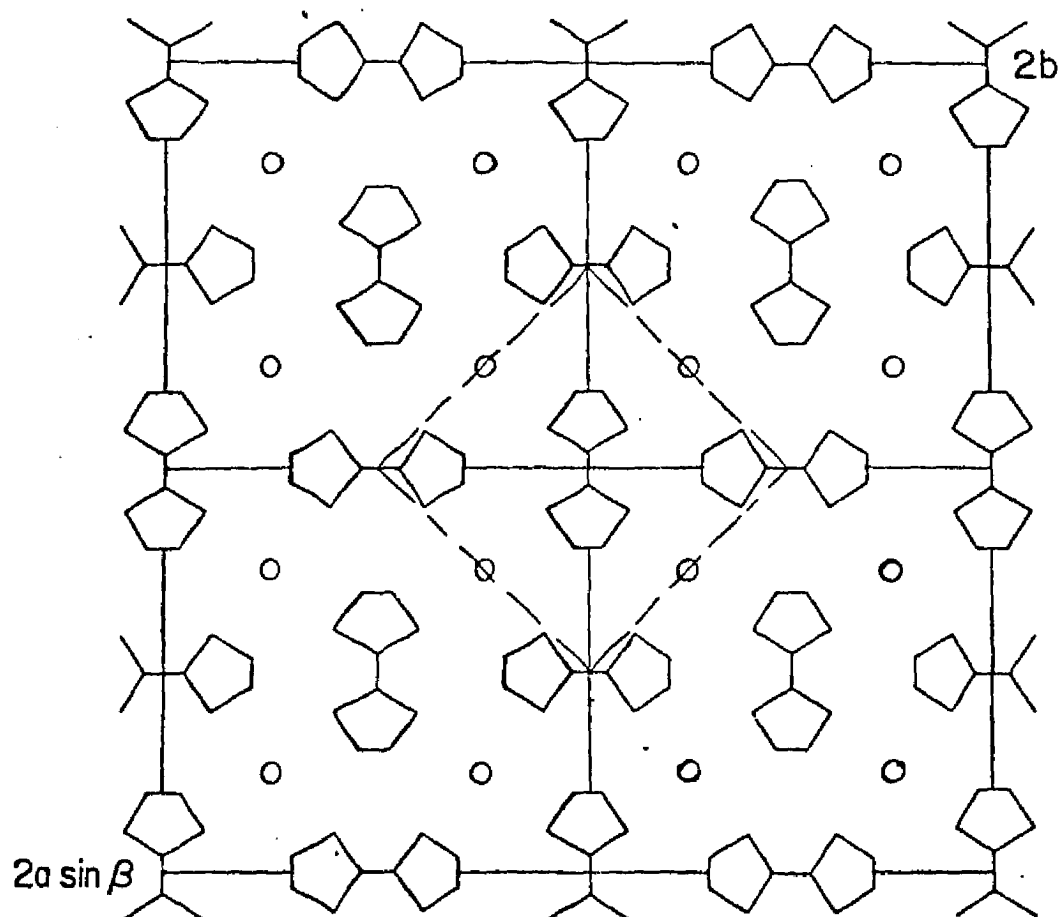


Figure 6. Relative orientations of monoclinic and tetragonal a , b planes. The dashed line shows the tetragonal cell.

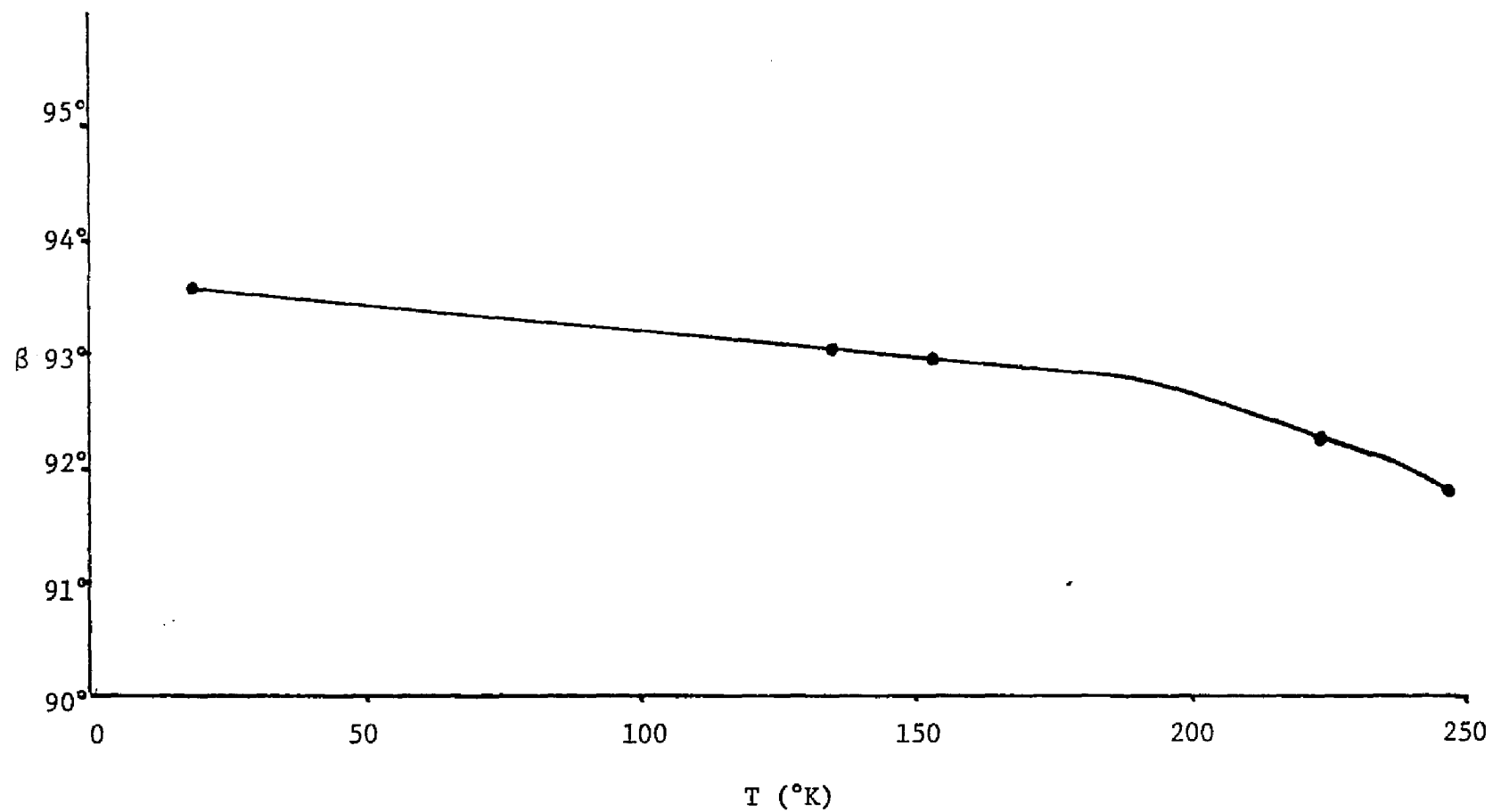


Figure 7. Monoclinic angle β vs. T for the low temperature phase of $(\text{TTF})\text{Cl}_{0.67}$.

The arrangement of the TTF cations in the monoclinic cell cannot be determined with certainty because of the considerable overlap of reflections caused by the quintuple twinning. One interpretation of the monoclinic distortion is presented in Fig. 8, which assumes that the TTF molecules retain $2/m$ symmetry in space group $C2/m$; the eclipsed stacking is retained and the monoclinic c -axis remains perpendicular to the TTF molecular planes. This interpretation implies that the individual TTF columns are translated slightly relative to each other in the $[001]$ direction. This slipping of parallel TTF columns is seen in the structures of several monoclinic partially oxidized TTF halides and pseudohalides.^{5,6}

The location of the chloride ions in this structure cannot be determined from the present diffraction data. Nevertheless, it is tempting to attribute the tripling of the monoclinic c -axis (TTF-stacking period) to ordering of the chloride ions in each channel inasmuch as $3d_{\text{TTF}} = 2d_{\text{Cl}^-}$ (or $d_{003}/d_{002} = 0.67$ as was discussed earlier in this paper; see also Table 1). In the absence of refinable structure data (because of quintuple twinning) we cannot, of course, exclude the possibility of a very slight Peierls distortion that would tend to drive the TTF entities toward dimerization or trimerization. However, a large Peierls distortion seems unlikely because the reflections with $l \neq 3n$ remain weak.

The phenomenon of increasing β -angle with decreasing temperature was also observed in X-ray studies by Thomas *et al.*⁵ of $(\text{TTF})_5(\text{SCN})_7$, where it is believed to reflect coupling of charge-density waves with lattice strains. We cannot rule out the possibility that the TTF stacks are undergoing a dynamic transition such as was measured in that compound and in $(\text{TTF})(\text{TCNQ})$ by X-ray diffuse scattering.^{1,5}

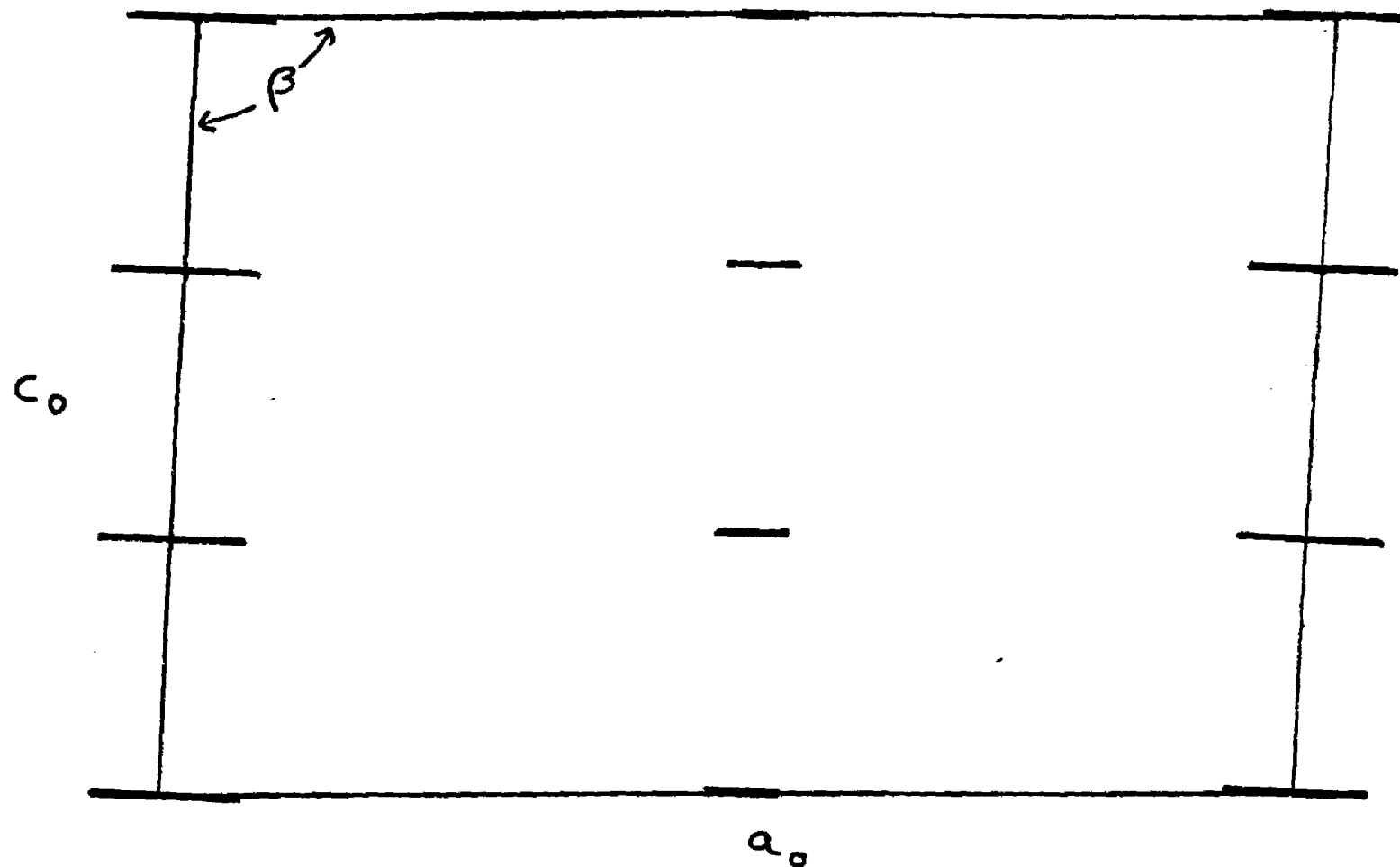


Figure 8. Interpretation of monoclinic distortion. Long and short thick lines represent side- and end-on TTF cations, respectively.

The peak broadening occurring on fast cooling of $(\text{TTF})\text{Cl}_{0.67}$ completely obscures peak splitting when a cooling rate of $\sim 100^\circ \text{ K/hour}$ is employed. Furthermore, while there is little peak broadening and no observable peak splitting in 2θ scans after slow cooling ($\sim 1^\circ \text{ K/hour}$), the full width at half maximum, $W_{1/2}$, of these profiles increases dramatically with fast cooling. For example, $W_{1/2}$ corresponds to $\sim 0.6^\circ (2\theta)$ at 300° K and $\sim 3.0^\circ (2\theta)$ at 19° K after cooling at a rate of $\sim 100^\circ \text{ K/hour}$. After cooling this quickly, a very slow increase in peak intensity may occur.¹⁴

This peak broadening is most likely due to formation of extremely small diffracting domains within the crystal. Since there are five kinds of diffracting domains below the phase transition, it is likely that individual domains of any one kind are very small. The dependence of peak profiles on the size of diffracting domains is well understood.¹⁵ We have not investigated this phenomenon in great detail. A calculation based on Wilson's treatment indicates that domains in $(\text{TTF})\text{Cl}_{0.67}$ may be $\lesssim 300 \text{ \AA}$ in diameter after cooling to 19° K at a rate of $\sim 100^\circ \text{ K/hour}$.

Electrical Properties

The electrical properties of $(\text{TTF})\text{Cl}_{0.67}$ are identical to those of $(\text{TTF})\text{Cl}_{0.70}$ as determined by measurements on single crystals from the same batches that were used in the crystallographic work. However, the properties differ from those of the other halides of TTF.^{2,3,4} The room-temperature conductivity is $\sigma \sim 150 \Omega^{-1}\text{cm}^{-1}$, the lowest in this series of compounds. As the temperature is lowered the conductivity decreases and

drops off sharply at the phase transition ($\sim 250^\circ \text{ K}$), and then continues to decrease less rapidly as shown in Fig. 9. The plot of $\log \sigma$ versus $1/T$ is linear only over the narrow temperature range from 125° K to 200° K and yields an activation energy, Δ , of $\sim 0.13 \text{ eV}$. Assumption of a disordered model,¹⁶ where $\ln \sigma \sim 1/T^\gamma$ and $\gamma = 1/3$ or $1/4$, did not lead to a significant improvement in the fit of the data. This was expected because the results obtained from the crystallographic work provided evidence for ordering in the chloride lattice.

The thermoelectric power (TEP) is small and positive at room temperature, and is relatively constant down to 260° K . At 250° K there occurs a sharp upturn and then the TEP continues to increase with decreasing temperature.

Thus, both the electrical conductivity and the TEP change dramatically at the phase transition. Below the transition nonmetallic behavior is observed. In view of the disorder in the chloride lattice one might expect $(\text{TTF})\text{Cl}_x$ to behave as a disordered 1-D metal in which all the states of the conduction electrons are localized. The finite conductivity would then be due to the strong electron-phonon interaction such that electron transport takes place by phonon-assisted hopping.¹⁸ Thus, the conductivity would be diffusive in nature and would decrease as the temperature is lowered, as was observed experimentally. However, the mere presence of disorder is not sufficient to guarantee such behavior because of the competing factors such as the disorder potential, δ , the interchain coupling, and the electronic bandwidth. For example, in an anisotropic conductor with nearest-neighbor transfer integrals, the critical value of

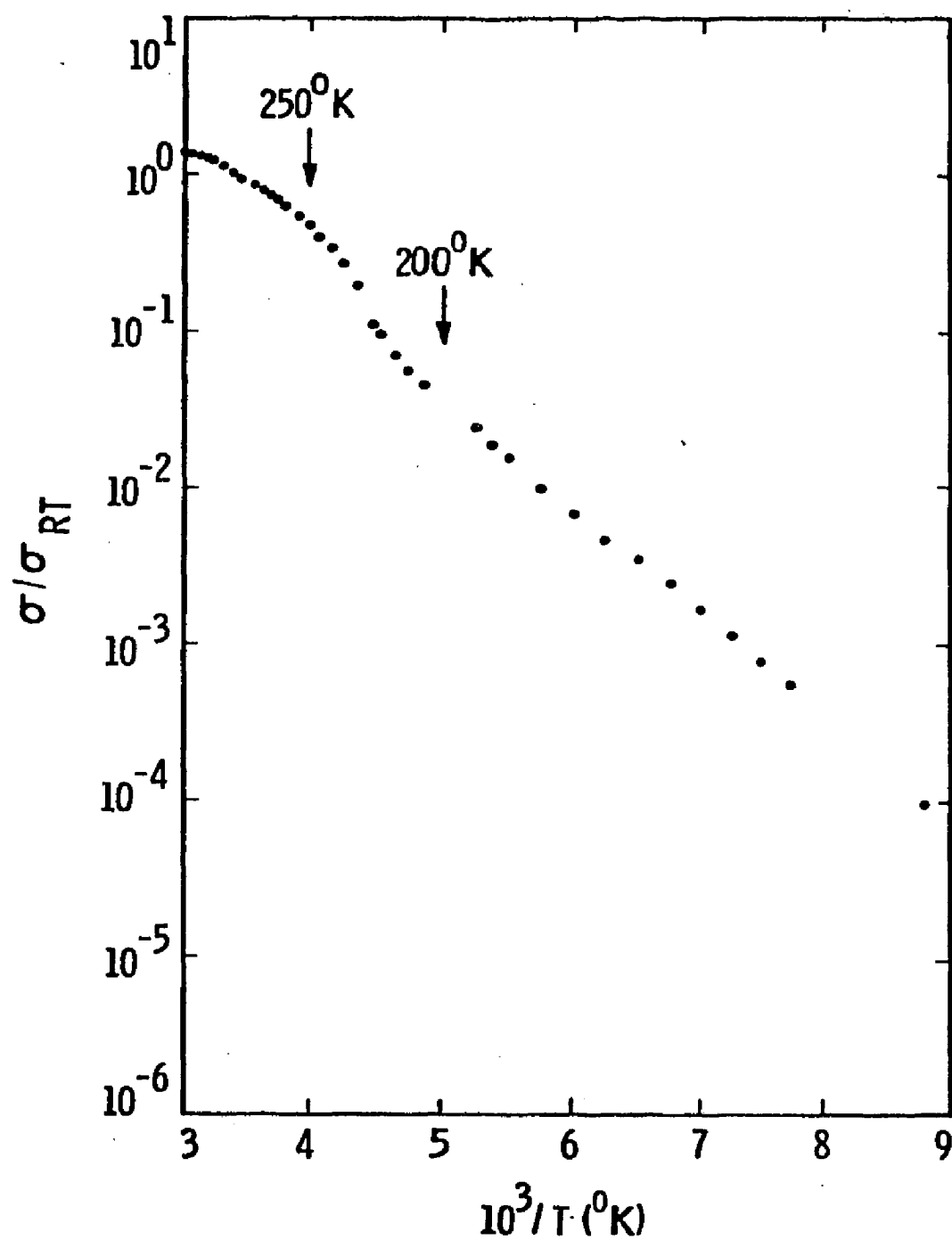


Figure 9. Temperature dependence of the conductivity of $(\text{TTF})\text{Cl}_{0.67}$.

$$\sigma_{RT} \sim 100 \, \Omega^{-1} \text{ cm}^{-1}.$$

the disorder potential is $\delta_c \sim (t_{\parallel} t_{\perp})^{1/2}$, where t_{\parallel} is the longitudinal transfer integral (proportional to the electronic bandwidth) and t_{\perp} is transverse transfer integral (interchain coupling).¹⁹ All of the conduction-electron states will be localized only if $\delta > \delta_c$. The electronic bandwidth is expected to be large in $(\text{TTF})\text{Cl}_x$ because of the eclipsed stacking of TTF molecules.^{2,20} Similarly, the occurrence of ordering and a structural phase transition near 250° K indicates that interchain forces are not weak. Therefore, we believe that the conductivity in the high temperature state is not dominated by disorder, but reflects the presence of fluctuations, or precursors, of the phase transition.

Another example of this interaction between disorder and interchain coupling is (tetrathiatetracene)₂(iodide)₃, $(\text{TTT})_2\text{I}_3$. This is a single-carrier, quasi-one-dimensional organic metal in which the TTT molecules stack uniformly in a slipped fashion, in contrast to the eclipsed stacking of the TTF molecules in $(\text{TTF})_3\text{Cl}_2$.¹⁴ The iodine chains exhibit considerable disorder. Unlike $(\text{TTF})_3\text{Cl}_2$, the temperature dependence of the conductivity of $(\text{TTT})_2\text{I}_3$ is metal-like, even in the presence of observable disorder. The conductivity exhibits a broad maximum near 100° K and slowly decreases as the temperature is lowered. A possible phase transition occurs at 20-25° K. X-ray diffraction studies do not reveal any significant ordering of the iodine ions, nor is there any evidence of a structural transition in the TTT lattice upon cooling to 27° K.¹⁶ We believe that this is another example of structural disorder disrupting interchain correlations such that the phase transition is smeared and suppressed to low temperatures. Two other interesting points emerge from the comparison of $(\text{TTF})_3\text{Cl}_2$ and

(TTT)₂I₃: (a) the temperature dependence of the structural disorder may vary significantly among different quasi-1D organic metals; and (b) the disorder observed at room temperature may be quite different at low temperatures.

Below the transition, the conductivity exhibits a nonmetallic temperature dependence even though the chloride sublattice is now ordered. In view of the recent work on (TTF)₁₂(SCN)₇,⁵ which is electronically and structurally analogous to (TTF)Cl_x, we feel that a dynamic distortion occurs along the chains and that charge density waves are present in the monoclinic phase. However, an important difference between the monoclinic phases of the two materials should be noted. (TTF)₁₂(SCN)₇ exhibits structural disorder corresponding to the two possible orientations of the SCN anions in the chains,¹⁶ while the chloride lattice is ordered in the monoclinic phase. We feel that the effect of the disorder in the SCN lattice is to favor the formation of incommensurate CDW. The influence is absent in (TTF)₃Cl₂ and should lead to commensurate CDW and semiconducting behavior. A similar situation is found in TaS₃ where a Peierls transition occurs along with the formation of commensurate CDW.¹⁷ The temperature dependence of the conductivity of (TTF)₃Cl₂ and TaS₃ are quite similar. Obviously, (TTF)Cl_x needs be studied by X-ray diffuse scattering experiments to verify or disprove our predictions.

Conclusions

The chloride sublattice in (TTF)Cl_x, $x = 0.67$ and 0.70 , is disordered at room temperature. The disorder diminishes over a narrow temperature range at about 250° K. For (TTF)Cl_{0.70}, in which the disordered chloride

sublattice is incommensurate with the TTF sublattice at room temperature, the chloride chains order into two different configurations, resulting in the formation of the commensurate structures $(\text{TTF})_3\text{Cl}_2$ and $(\text{TTF})_7\text{Cl}_5$. In $(\text{TTF})\text{Cl}_{0.67}$, the chloride sublattice is ordered and remains commensurate below the transition temperature. The TTF sublattice undergoes a static structural phase transition from tetragonal to monoclinic symmetry at $\sim 250^\circ \text{ K}$. This transition is incomplete in the sense that domains of both tetragonal and monoclinic symmetry are present. No evidence of a static Peierls transition is observed. The ordering of the chloride sublattice is important for the development of the structural phase transition of the TTF sublattice. Comparison of the electrical conductivity of $(\text{TTF})\text{Cl}_x$ with other quasi-1D conductors [i.e., $(\text{TTF})_{12}(\text{SCN})_7$ and TaS_3] suggests that a dynamic distortion occurs along the TTF chains, resulting in the formation of commensurate CDW and semiconducting behavior. The monoclinic cell of $(\text{TTF})\text{Cl}_{0.67}$ is similar to the room temperature cells of $(\text{TTF})\text{Br}_{0.74-0.78}$ and $(\text{TTF})\text{I}_{0.71}$ except that the c_0 -axis periods are different. We would expect to observe a high temperature tetragonal phase for the two latter materials.

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CHAPTER 4

The Structure of Low-Disorder
Bis-Tetrathiatetracene Triiodide

Introduction

The tetrathiatetracene (TTT) iodide system is of great current interest because it contains two quasi-one-dimensionally conducting phases, $(\text{TTT})_2\text{I}_3$ and $(\text{TTT})\text{I}$, both of which have unusual structural and electrical properties (1,2). Furthermore, the highly conducting phase, $(\text{TTT})_2\text{I}_3$, exists in several modifications which show a systematic variation in iodide-associated disorder and a concurrent variation in temperature dependent electrical properties (3). A room temperature x-ray diffraction study of low-disorder (l.d.) $(\text{TTT})_2\text{I}_3$ was undertaken in order to learn more about the nature of the iodide chain (2,4).

Experimental

Long, flat, needle-like crystals of l.d. $(\text{TTT})_2\text{I}_3$, grown by slow cooling of a nitrobenzene solution of TTT and I_2 from 95° C to room temperature, were supplied by Dr. R. B. Somoano. Crystals were six-sided prisms with [001] and [101] faces predominating and a metallic-gold luster. Crystals were initially characterized by

x-ray diffraction photography using oscillation and Weissenberg photographs. Unit cell parameters were obtained from a least-squares fit to the 2θ , ϕ , and χ values of 9 reflections manually centered on the diffractometer with the use of beam splitter and peak profiles. Crystal data are listed in Table 1. The crystals were all very small; therefore, intensity data were collected using Ni-filtered $\text{CuK}\alpha$ radiation. A crystal measuring $0.383 \times 0.043 \times 0.008$ mm was used for intensity data collection at room temperature on a quarter-circle G.E. diffractometer. A total of 1276 unique reflections, 915 due to the sublattice and 361 from the diffuse third layer, were measured using 2θ - θ scans and scan rates of $1^\circ/\text{min}$ and $1/2^\circ/\text{min}$, and background intensities were measured at the beginning and end of each scan for periods totaling 1 min or 1 min, 20 sec. Three standard reflections, 224, 002, and 606 were measured every 40 reflections for purposes of scaling. They did not show ($>1\%$) significant crystal decay, although oscillation photographs taken after data collection revealed that sharpness and intensity of the diffuse $h3l$ reflections of the supercell had decreased substantially. The intensities and their standard deviations were scaled (including minor decay correction), and corrected for absorption ($\mu = 272.9 \text{ cm}^{-1}$), Lorentz, and polarization effects (5). The atomic scattering factors of S and I were corrected for the real contribution to anomalous dispersion (6). 2θ - θ scans of several reflections of the subcell and third diffuse layer were recorded with a strip-chart recorder in order to compare profiles.

Table 1. Crystal Data (l.d.) TTT_2I_3

$$a = 18.468 \text{ (2)}$$

$$b = 9.851 \text{ (1)}$$

$$b \text{ subcell} = b/2$$

$$c = 18.299 \text{ (2)}$$

$$\text{Volume} = 3329.0 \text{ \AA}^3$$

$$\text{Space Group} = \text{Cmca} \text{ (subcell)}$$

$$\text{Pmc2}_1 \text{ (full cell)}$$

$$\text{F.W. (C}_{36}\text{H}_{16}\text{S}_8\text{I}_3) = 1085.75$$

$$z = 4 \text{ formula units per cell}$$

$$d_{\text{calc}} = 2.166 \text{ g/cm}^3$$

$$\mu \text{ (absorption coefficient)} = 272.87 \text{ cm}^{-1} \quad \text{CuK}\alpha \text{ rad.}$$

Electrical measurements of TTT_2I_3 crystals were performed using standard 2 and 4 probe techniques and aquadag contacts. Conductivity vs. temperature and thermoelectric power vs. temperature are shown as plots in Appendix 2.

Photographic results

Oscillation photographs of l.d. $(\text{TTT})_2\text{I}_3$ revealed a set of sharp diffraction lines corresponding to a spacing of 4.92 \AA . Diffuse layer lines corresponding to a spacing of 9.86 \AA were also seen, with spots superimposed on the third diffuse line. These diffuse spots correspond to a doubled unit cell, within the limits of the measurement. The photograph is shown in Figure 1. Weissenberg photographs of subcell layers indicated conditions for reflections, $h0l$: $h = 2n$, $l = 2n$ and hkl : $h + l = 2n$, consistent with Cmca symmetry. A Weissenberg photograph of the third diffuse line revealed spots with slight extension in ω and not significantly broadened in the 2θ direction. No systematic absences were seen in this Weissenberg photograph, implying that the iodide lattice and TTT lattice have different symmetry.

Structure determination and refinement

Atomic coordinates for C, S, and I atoms were obtained from previous work (7), and provided sufficient phasing to make refinement of C and S in the subcell possible. Positions of I atoms could

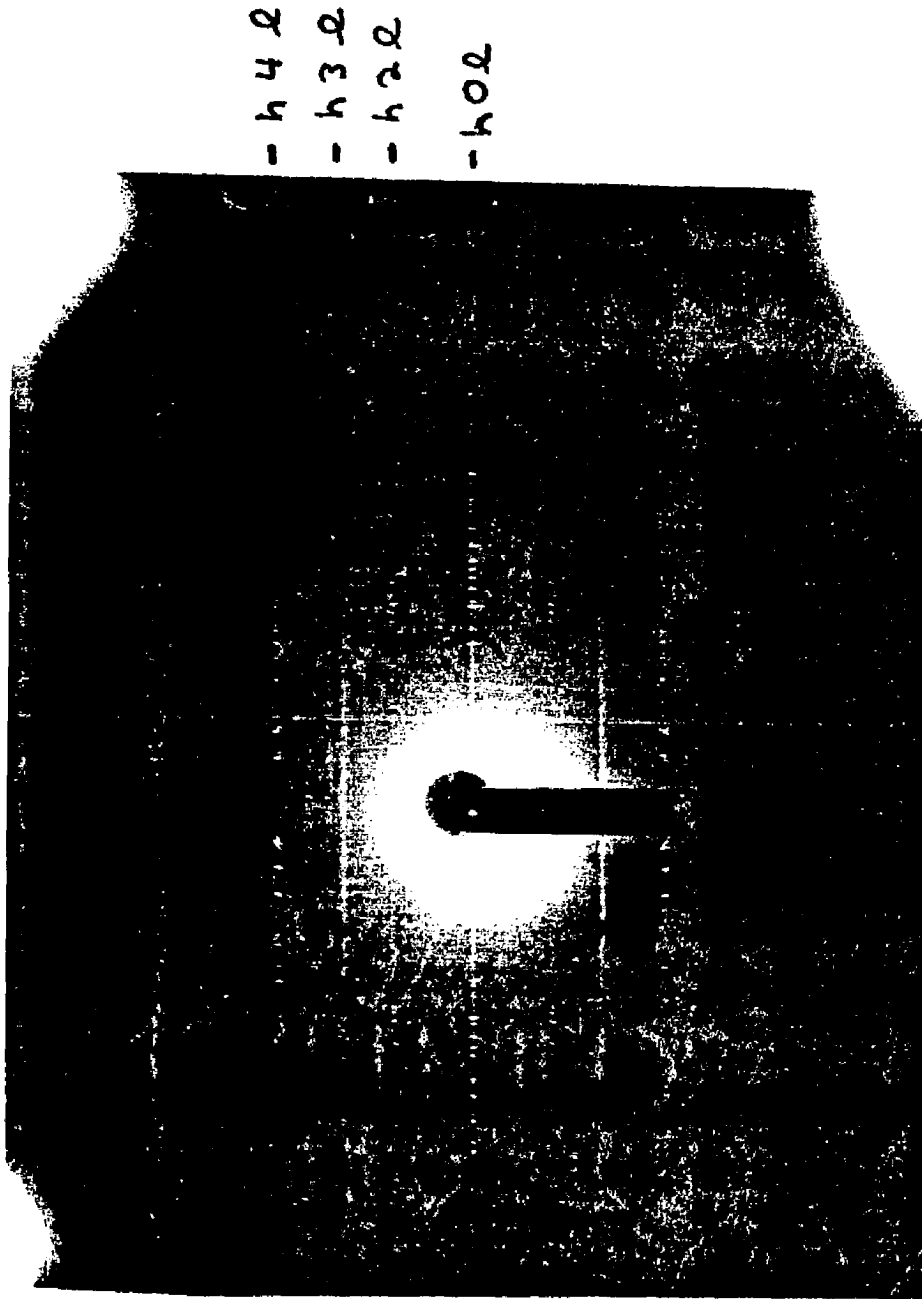


Figure 1. Oscillation photograph of (1.d.) TTT_2I_3 .

not be refined. After C and S atoms were refined, I atom coordinates, populations, and anisotropic temperature factors were adjusted in order to obtain a flat difference Fourier map. H atom positions were calculated, and the final stage of refinement consisted of refining H atom coordinates and isotropic temperature factors, and I atom populations and anisotropic temperature factors. C and S atom parameters were fully refined at this level. The converged structure gave $R = 0.111$, $wR = .037$ for all 915 subcell reflections; $R = 0.085$, $wR = .034$ for 548 reflections with $F^2 > 3\sigma$. The distribution of iodine obtained from subcell data is a smoothly varying column of electron density along $(\frac{1}{4}, y, \frac{1}{4})$ with a minimum of 5.15 and a maximum of 26.2 $e/\text{\AA}^3$. This is similar to the results obtained by Smith and Luss (4), except that they see a variation from 16 to 37 $e/\text{\AA}^3$. Comparison of relative maxima and minima imply a greater degree of registry between the I chain and TTT lattice in (1.d.) TTT_2I_3 , as is also suggested by the differences in diffuse layer line spacings observed photographically. The differences in absolute value of electron density maxima is not significant, or grid size on Fourier maps may have been different. As there is only one I-chain maximum in electron density in the asymmetric unit, interpretation is subject to question. For this reason, a structural refinement was carried out on 1276 unique data, including 915 subcell data, and 361 $h3l$ data. Since there are no systematic absences in the $h3l$ data set, the symmetry of the

full cell is lower than $Cmca$. Space group $Pmc2_1$ is consistent with the arrangement of the TTT molecules, and with the lack of systematic absences for $h3l$ reflections. The most important effect of adding $k = 3$ data is to add electron density to two of the four now independent electron density maxima along $(\frac{1}{4}, y, \frac{1}{4})$ and to subtract electron density from, and smear out, the other two maxima. The presence of weak $h3l$, h odd, reflections requires a small displacement from $x = \frac{1}{4}$ for some of the iodine atoms. The phase of $h3l$, h even, reflections is determined by the arbitrary selection of two of the four equivalent electron density maxima for reinforcement. The phasing of the $h3l$, h odd, reflections cannot be determined as they are all very weak, and do not interact with other data. These reflections would have zero intensity if the iodine atoms were located precisely at $x = \frac{1}{4}$. The magnitude of the displacement can be determined from the $h3l$, h odd, reflections, but the direction of the displacement cannot be determined. Refinement of the model was performed by adjustment of populations, temperature factors, and coordinates of iodine atoms until the difference Fourier map was nearly flat. The R factor reached 0.137 for all 1276 data, and 0.091 for 623 data with $F^2 > 3\sigma$. The weighted residuals were 0.055 and 0.050, respectively. The residual, R , for third level reflections alone was 0.297. Fourier maps of $(\frac{1}{4}, y, z)$ and $(x, y, \frac{1}{4})$ reflections are shown in Figure 2. Note that there is significant electron density at three of the four minima; that all maxima

represent less than full iodide occupancy, and that the distance between the strongest maxima is 2.68 Å, just 0.02 Å longer than the I₂ bond length. Although minor diffracted intensity in diffuse lines, corresponding to one-dimensional order, could not be included in the structural refinement, this additional diffracted intensity is fairly weak. The diffraction pattern, with third level data included, is clearly inconsistent with discrete triiodide ions. The electron density map is consistent with the presence of species such as I⁻, I₂, and I₃⁻, which may be in dynamic equilibrium.

The atomic coordinates and temperature factors of the TTT cation were not refined in the lower symmetry space group of the complete cell. Patterson maps of the h3ℓ reflections show significant intensity only at (0,y,0); (0.5,y,0); (0.5,y,0.5); and (0,y,0.5). Thus only iodine-iodine vectors are necessary for interpretation of the Patterson map, and contributions from C and S atoms of TTT molecules are negligible.

The atomic coordinates and temperature factors of C, S, and H atoms in the subcell are given in Table 2. Figure 2 shows the molecule and its bond lengths and angles. Coordinates and temperature factors of the iodine atoms used to model the I chain structure in the subcell and full cell are given in Table 3. It should be understood that these parameters model a continuous distribution of iodide species and, individually, have little significance.

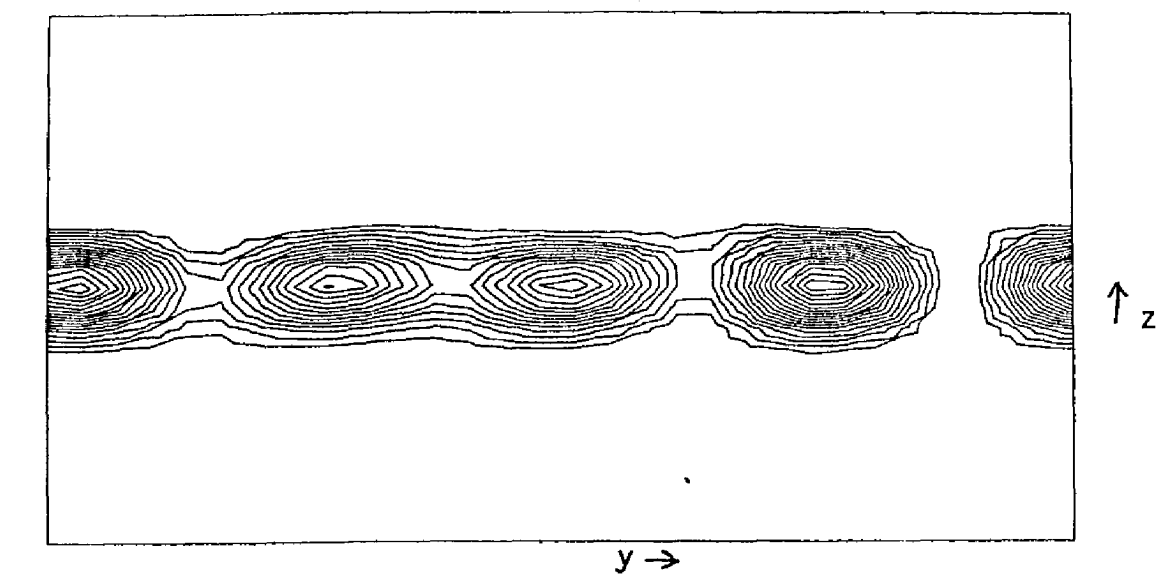
Table 2. Refined Atomic Coordinates and Temperature Factors of the TTT Cation in (TTT)₂I₃ (1.d.)

	x	y	z	U11	U22	U33	U12	U13	U23
C 1	96094(52)	0	0	419(55)	315(66)	373(48)	0	0	-58(57)
C 2	92395(37)	17964(176)	4441(46)	370(36)	424(59)	481(42)	11(41)	11(33)	25(47)
C 3	96052(37)	36659(178)	8951(38)	484(43)	337(45)	356(34)	-20(42)	60(34)	27(39)
C 4	92279(48)	55312(184)	13493(46)	728(57)	396(60)	420(41)	25(50)	80(41)	44(46)
C 5	96146(46)	73247(202)	17678(46)	798(58)	488(59)	416(38)	-20(50)	26(43)	-17(48)
S 1	82910(11)	15187(55)	3861(14)	362(9)	640(15)	724(14)	45(12)	39(10)	-100(15)

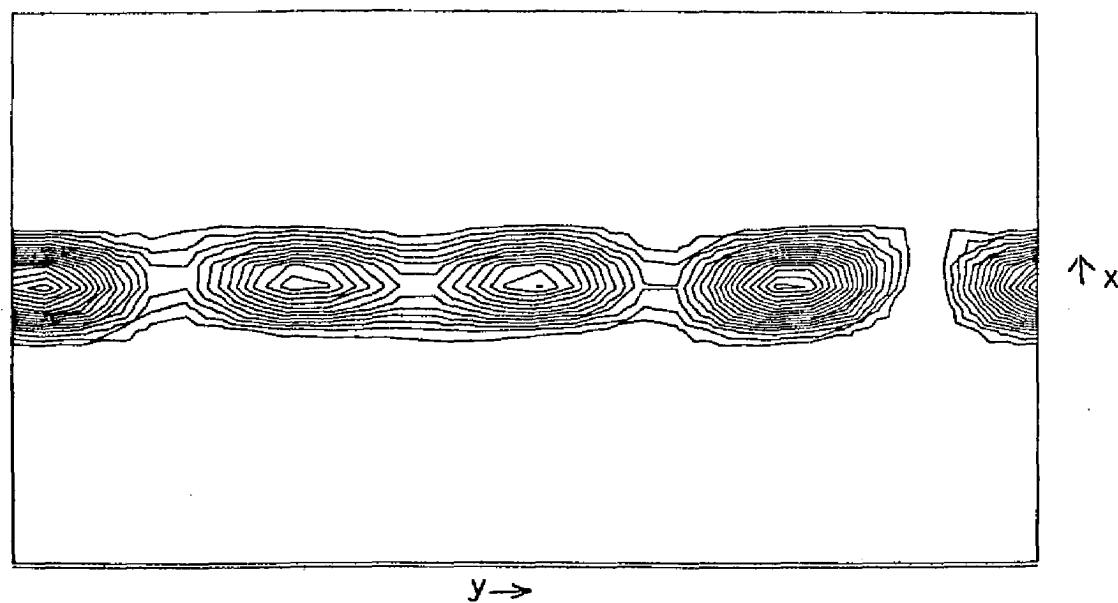
26

	x	y	z	
H 1	8473	5561	1161	B = 7.48
H 2	9135	8661	2109	B = 8.25

Atomic coordinates of C and S are multiplied by 10⁵. Uij's and atomic coordinates of H atoms are multiplied by 10⁴.



SECTION X = .250003



SECTION Z = .250003

Figure 3. Electron density maps through iodine chain in (l.d.) $(\text{TTT})_2\text{I}_3$. Scale is about $0.8 \text{ \AA}/\text{cm}$; contours at $2e/\text{\AA}^3$ intervals.

Table 3. Coordinates and Temperature Factors used to Model I-Chain in TTT_2I_3

Subcell - Cmca

	x	y	z	U11	U22	U33	U12	U13	U23	Population
I 1	250	125	250	852	2306	1667	0	772	0	0.224
I 2	250	285	250	414	2624	505	0	-14	0	1.014
I 3	250	405	250	2233	435	4125	0	-683	0	0.121

Full Cell - P

I 1	254	25	246	380	2483	560	93	171	9	0.657
I 2	253	420	248	380	12910	560	0	171	0	0.153
I 3	253	758	254	380	2483	560	93	171	9	0.693
I 4	247	278	250	380	2483	560	93	-171	92	0.351
I 5	245	499	254	380	2483	560	93	171	92	0.369
I 6	251	92	250	518	12910	679	0	171	0	0.050
I 7	248	580	250	518	12910	679	0	171	0	0.020
I 8	253	680	250	518	12910	679	0	171	0	0.060
I 9	242	194	250	518	12910	679	0	171	0	0.065
I 10	255	340	250	B=6.0						0.030
I 11	240	840	250	B=6.0						0.040
I 12	256	940	250	B=6.0						0.040

64

All coordinates have been multiplied by 10^3 . All U_{ij} 's have been multiplied by 10^4 .

Discussion

Figure 4 shows a model which adequately explains the observed electron density of the iodine chain in (l.d.) TTT_2I_3 . It is likely that I_3^- is the dominant species. Two-fold disorder between the crystallographically distinct configurations shown in 4a and 4b accounts for the fractional occupancy ($\sim .6$) of two of the iodine positions ($y = .28$; $y = .50$), and the "smearing out" of electron density in the y direction. The fractional occupancy ($\sim .8$) of the other positions may be explained by additional configurations. However, the very significant electron density between maxima suggests that there is exchange of iodine atoms between sites. Intermediate configurations such as Figure 4c and 4d may be involved in transitions from 4a to 4b. These would occasionally leave the higher occupancy sites empty. Species such as I_2 , I^- , I_5^- , and perhaps others would be present, although they would be much less likely than I_3^- , in agreement with the observation of I_3^- in TTT_2I_3 by Raman spectroscopy (8).

The presence of disorder in the iodide chains of TTT_2I_3 has been invoked to explain the retention of conductivity at low temperature, although the nature of the disorder and its behavior with cooling have not been understood (1-4,7). The iodide chains in (l.d.) TTT_2I_3 contain partially occupied sites which are statistically filled by several species in several possible configurations. Microscopically, these chains result in an aperiodic potential around the TTT stacks.

It is unlikely that there are important changes in the iodide chain at low temperature (3,7). This aperiodic potential may suppress a complete Peierls' transition by creating states within the gap and suppressing "locking" of charge density waves on adjacent stacks. This qualitatively explains the retention of moderate conductivity at low temperature in TTT_2I_3 , although a complete explanation must await completion of low-temperature structure refinement.

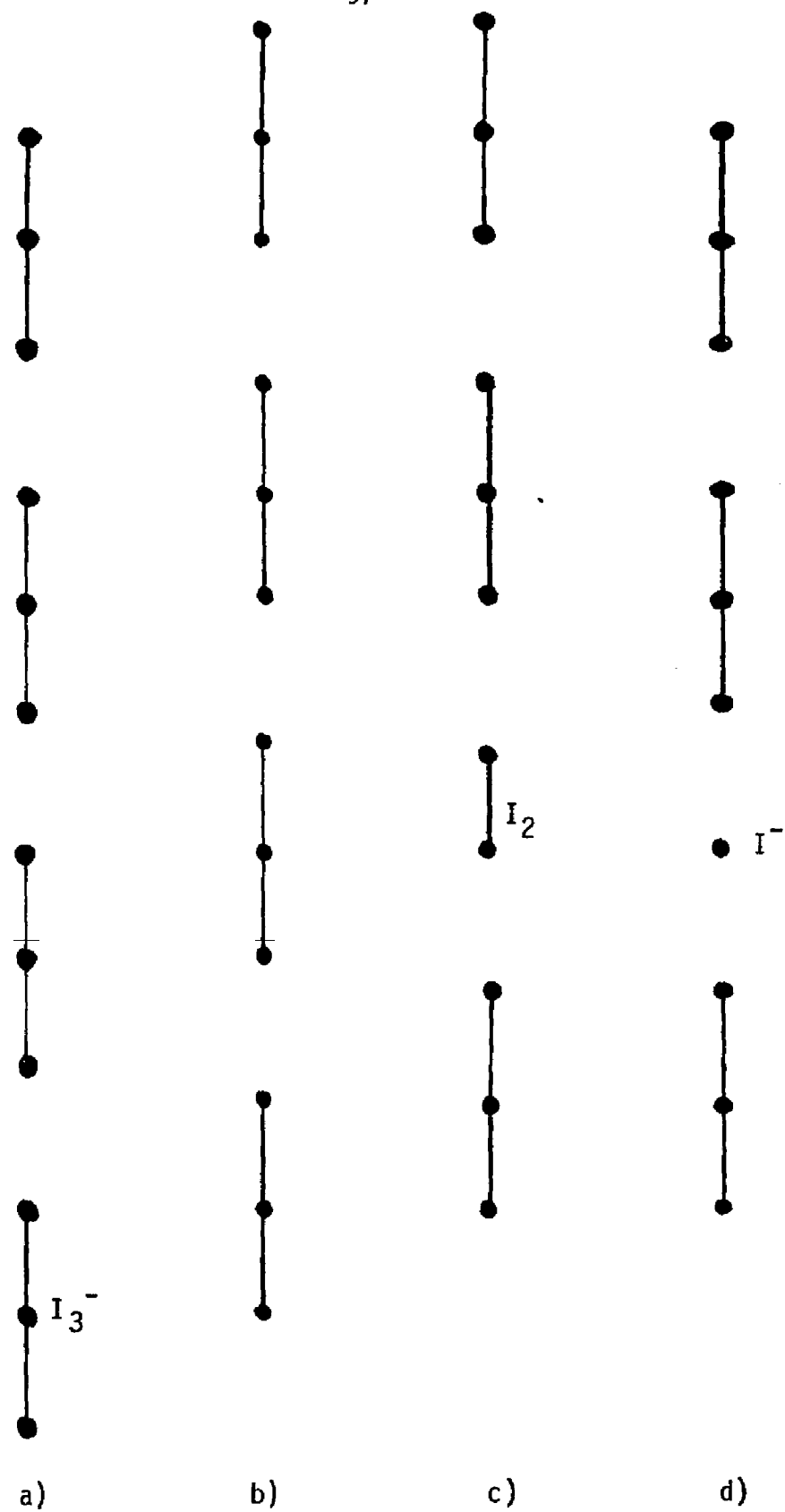


Figure 4. Possible configurations of iodine species contributing to chain in $(\text{TTT})_2\text{I}_3$

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CHAPTER 5

Tetramethyltetraselenofulvalene Bromide and Thiocyanate;
Electrochemical Preparation of
Conducting Organic Crystals

Tetrathiafulvalene (TTF) forms highly conducting stacked charge-transfer salts with both organic acceptors such as tetracyanoquinodimethane (TCNQ) and simpler anions such as halides and pseudohalides (1,2). Derivatives of TTF and its selenium analog, tetraselenofulvalene (TSF), also form highly conducting TCNQ complexes (3,4), but their halide and pseudohalide salts have been studied less thoroughly (5). I have prepared crystals of tetramethyltetraselenofulvalene bromide and thiocyanate $(\text{TMTSF})\text{Br}_{0.82}$ and $(\text{TMTSF})(\text{SCN})_{0.50}$ in order to explore further the relation between structure and physical properties in quasi-one-dimensional conductors.

Preparation of good single crystals of organic conductors is often the most serious obstacle to study of these materials. Electrochemical crystal growth techniques have successfully exploited the conducting properties of a variety of inorganic materials (6,7) and a few organic materials (8). The crystal formed by electrolysis must be quite insoluble in the solution or melt in which the precursor is dissolved. As the conducting crystal grows, it acts as an extension of the electrode. Defects inhibit

crystal growth, because of conductivity decrease. Large crystals are, therefore, well formed single crystals. Insulating phases, which are often formed in other kinds of crystallization experiments, are avoided by this technique. Electrochemical crystal growth experiments involving oxidation of TTF, TSF, TMTSF, and HMTSF (hexamethylene-tetraselenofulvalene) in the presence of Cl^- , Br^- , I^- , SCN^- , and SeCN^- were carried out at a variety of oxidation potentials and temperatures in several solvents. This investigation was not exhaustive, but some conclusions about the suitability of electrochemical crystal growth to these systems can be drawn.

Experimental

Synthesis

(TMTSF)(Br)_{0.82}: 0.01 g TMTSF was dissolved in 1.5 ml. benzonitrile containing 0.1 Molar tetraethylammonium bromide. This solution was placed in the working electrode cell (Figure 1) and oxidized, at 0.02 Volt vs. $\text{Ag}^{\circ}/\text{AgClO}_4$ (sat), with a Pt wire electrode, at 69-70° C. Well formed black crystals up to 1 mm long were recovered after eight minutes.

(TMTSF)(SCN)_{0.50}: 0.006 g TMTSF was dissolved in 1 ml benzonitrile containing 0.02 Molar tetraethylammonium thiocyanate. Oxidation, at 0.00 Volt vs. Ag/AgClO_4 (sat), at a Pt wire electrode, for fifteen minutes at 56-64° C, gave good crystals up to ~1 mm long.

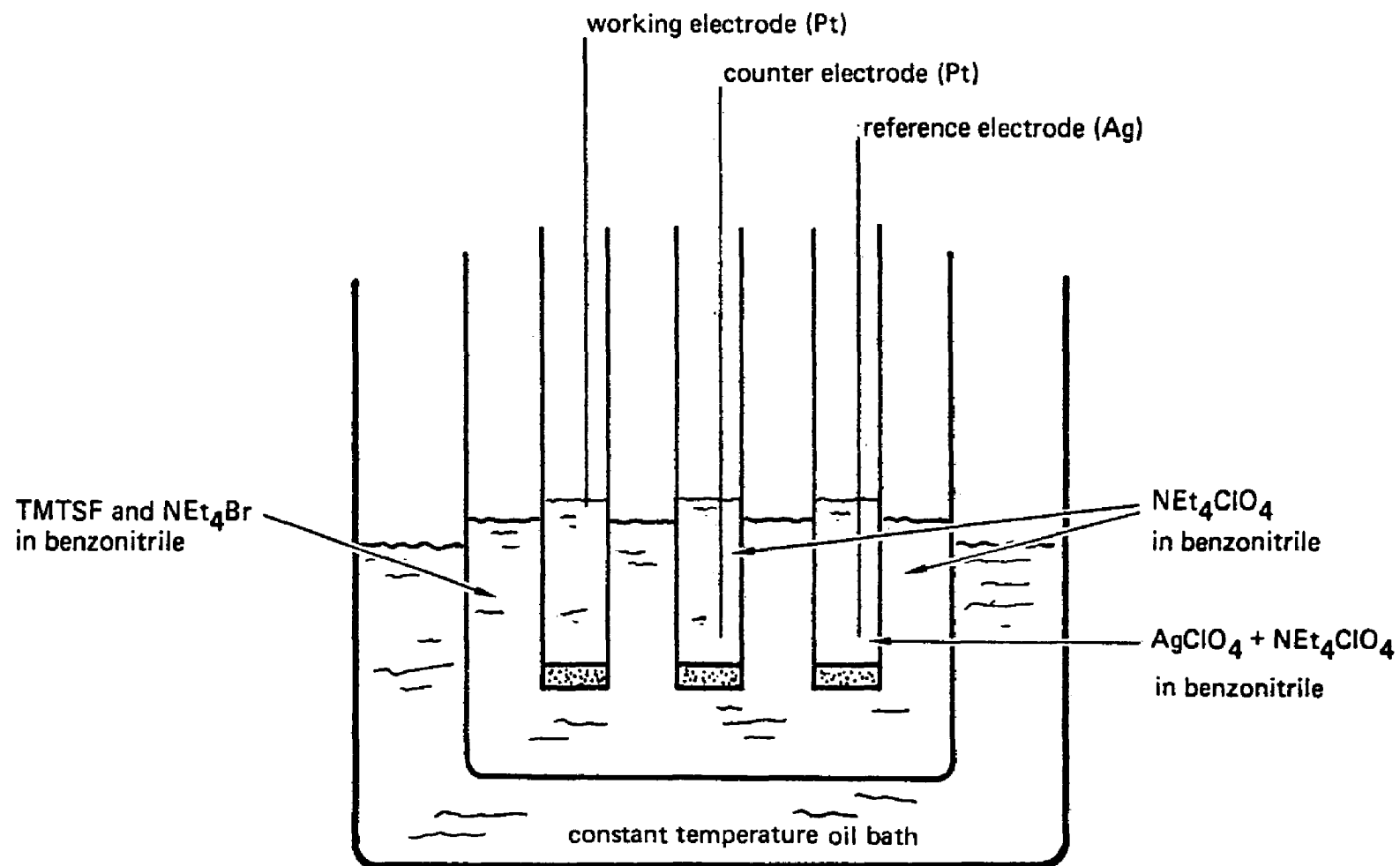


Figure 1. Electrochemical crystallization cell.

The conditions, and results, of these and other electrochemical crystallization experiments, are collected in Table 1.

The oxidizing potential was controlled using a Wenking 68 FR 0.5 potentiostat. Crystals of both of the TMTSF salts were flat needles exhibiting well formed {110} and {010} faces. Oxidation at temperatures greater than $\sim 80^\circ \text{C}$ resulted in little crystallization because of solubility of the products. Near room temperature, TMTSF is only slightly soluble in benzonitrile and oxidation gives crystals $\ll .01 \text{ mm}$ in thickness.

HMTSF is very insoluble in all common solvents at room temperature; a benzonitrile solution at 100°C is less than 0.02 Molar when saturated. Unfortunately, at this temperature, the oxidation product seems to be quite soluble also, even in the presence of fairly high concentrations of bromide or thiocyanate.

Oxidation of solutions containing I^- or SeCN^- resulted in oxidation of the anion, and formation of no solid product. Crystals of $(\text{TMTSF})\text{I}_x$ can be prepared by co-diffusion of $(\text{TMTSF})\text{ClO}_4$ and NBu_4I in acetonitrile.

Electrical measurements

Conductivity and thermoelectric power were measured for both kinds of crystals at room temperature and at lower temperatures. The results are shown in Appendix 2.

Table 1. Conditions, Results of Electrochemical Crystallization Experiments

Donor, Concentration	Electrolyte, Concentration, Working Electrode Cell	Solvent	Oxidation Potential, vs. $\text{Ag}^0/\text{AgClO}_4(\text{sat})$	Temperature	Results
TTF 1×10^{-3} - $6 \times 10^{-2} \text{M}$	$\text{NH}_4\text{SCN}, 4 \times 10^{-2} \text{M}$ $\text{NEt}_4\text{Br}, 1 \times 10^{-1} \text{M}$ $\text{NEt}_4\text{Cl}, 5-20 \times 10^{-2} \text{M}$	acetonitrile	0.0-0.5 Volt	Room Temperature	$(\text{TTF})_{12}(\text{SCN})_7^*$ good $(\text{TTF})(\text{Br})_{0.74}$ crystals, $(\text{TTF})(\text{Cl})_{0.67}$ $10 \times 1 \times 1 \text{ mm}^3$
TMTSF $1-2 \times 10^{-2} \text{M}$	$\text{NEt}_4\text{Br}, 1 \times 10^{-1} \text{M}$ $\text{NEt}_4\text{SCN}, 2 \times 10^{-1} \text{M}$ $\text{NBu}_4\text{I}, 1 \times 10^{-1} \text{M}$	benzonitrile	~ 0.0 Volt	60-80° C Room Temperature 100-120° C	$(\text{TMTSF})(\text{Br})_{0.82}$ good $(\text{TMTSF})(\text{SCN})_{0.50}$ crystals oxidation of I^- , no solid extremely thin crystals of bromide and thiocyanate no crystals, some coating of electrode
HMTSF $7 \times 10^{-3} \text{M}$	$\text{NEt}_4\text{Br}, 1 \times 10^{-1} \text{M}$ $\text{NEt}_4\text{SCN}, 2 \times 10^{-2} \text{M}$	benzonitrile	0.0-0.2 Volt	100-120° C	no crystals, some coating of electrode
TSF $3 \times 10^{-3} \text{M}$	$\text{NEt}_4\text{Cl}, 5 \times 10^{-2} \text{M}$ $\text{KSeCN}, 1.5 \times 10^{-2} \text{M}$	acetonitrile	0.1-0.4 Volt 0.0-0.4 Volt	Room Temperature	$(\text{TSF})\text{Cl}_x$, tiny hairlike crystals no solid produced in oxidation in presence of KSeCN

*

	%C	%H	%N
calc	33.19	1.69	3.43
found	33.01	1.94	3.39

X-ray diffraction photographs

Oscillation photographs of $(\text{TMTSF})(\text{Br})_{0.82}$ and $(\text{TMTSF})(\text{SCN})_{0.50}$ mounted with the rotation axis parallel to the C axis were taken. The C axis is $\sim 7 \text{ \AA}$ long for both compounds, but photographs of $\text{TMTSF}(\text{SCN})_{0.50}$ show additional weak layer lines. An oscillation photograph of $\text{TMTSF}(\text{SCN})_{0.50}$ is shown in Figure 2. Accurate measurement of this film gave the lattice spacings listed in Table 2. The satellite reflections are due to a superperiod $C' = 32.815 = 4.56 C$. The presence of strong satellites of the $hk0$ layer indicates that the superperiod is due to a structure factor modulation rather than a C axis length modulation, since the latter would give $hk0$ satellites of vanishingly small intensity (9). Systematic absences (hkl , $h + k = 2n + 1$; $h0l$, $l = 2n + 1$) were determined from Weissenberg photographs prior to data collection and indicated that the space group of both compounds was Cmcm .

Data Collection

Both data sets were obtained with Ni-filtered $\text{CuK}\alpha$ radiation ($\lambda = 1.54178 \text{ \AA}$) using 2θ - θ scans at rates of 0.5 or 1.0 degrees per minute on a locally assembled, Datex automated, General Electric quarter circle diffractometer. Background intensities were measured at both ends of the scans for a total of 60, 80, or 120 seconds. Two $\text{TMTSF}(\text{SCN})_{0.50}$ crystals were used for data collection. An entire hemisphere of data was collected, and merged, in order to

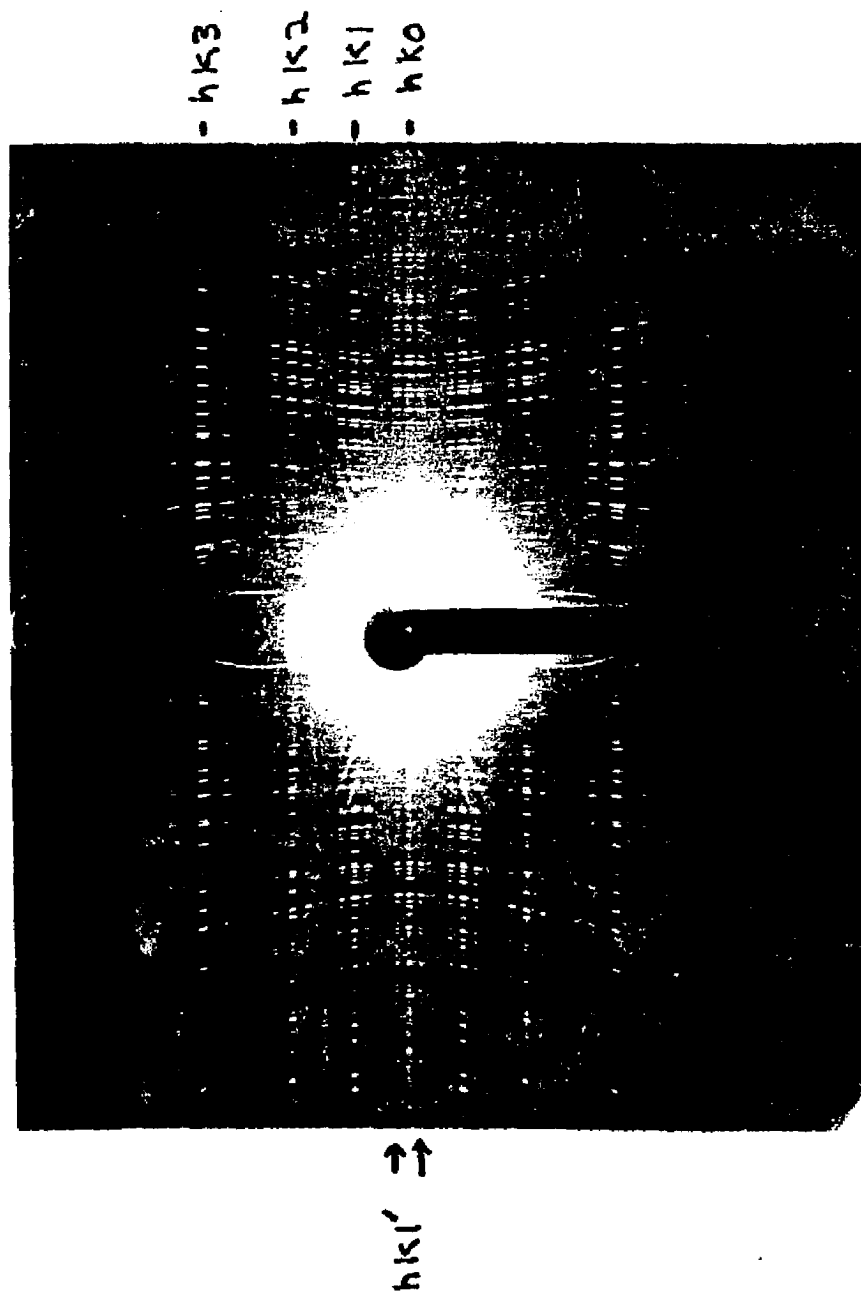


Figure 2. Oscillation photograph of (TMTSF)(SCN)_{0.5}.

Table 2. Layer Line Spacings of (TMTSF)(SCN)_{0.50}

$d \text{ (Å)}^*$	ℓ	ℓ'	$d/\ell = c_0 \text{ (Å)}$	$c_0'^\dagger$
32.8146	0	1'		32.8146
9.1088	1	-1'		34.2313
7.1893	1	0	7.189	
5.8934	1	+1'		32.5899
4.0242	2	-1'		33.9013
3.5983	2	0	7.197	
3.2363	2	+1'		32.2572
2.5774	3	-1'		34.4791
2.3970	3	0	7.191	
2.2268	3	+1'		31.1683
1.8000	4	0	7.200	

*

Very weak $\ell = n \pm 2'$ satellite reflections were observed but could not be accurately measured.

†

Calculated assuming that $d = (\ell/c_0 + \ell'/c_0')^{-1}$. $\overline{c_0} = 7.1944$;
 $\overline{c_0'} = 33.063$; $c_0'_{hk1'} = 32.815$; $\overline{c_0'}/\overline{c_0} = 4.60$.

average errors in data collection. A crystal measuring $0.070 \times 0.033 \times 0.533 \text{ mm}^3$ was used for intensity measurement of 1052 reflections due to the $9.919 \times 24.124 \times 7.220 \text{ \AA}^3$ cell; satellite reflections could not be well resolved on the diffractometer with this crystal which required a 1 mm collimator. A smaller crystal measuring $0.073 \times 0.027 \times 0.193 \text{ mm}^3$ was used for measurement of satellite reflection intensities. Using a 0.25 mm collimator, 197 unique $hk1'$ and $hk2'$ satellite reflections were measured, as well as several hundred combination reflections ($hkl \pm l'$).

The $(\text{TMTSF})(\text{Br})_{0.82}$ crystal used for data collection measured $0.05 \times 0.017 \times 0.40 \text{ mm}^3$. All 1029 reflections of $\text{TMTSF}(\text{Br})_{0.82}$ in one octant, having $2\theta \leq 155^\circ$ were collected. The data from both crystals were corrected for Lorentz, polarization, absorption, and decay effects (10). The scattering factors of selenium, sulfur, and bromide were corrected for the real part of anomalous dispersion (11). Crystal data for both crystals are given in Table 3.

Structure determination and refinement - $\text{TMTSF}(\text{Br})_{0.82}$

Symmetry, stacking, and steric considerations indicated that the TMTSF molecules were oriented perpendicularly to \bar{c} , lay on mirror planes at $Z = 1/4$ and $3/4$, and that successive molecules were slipped along b by $\sim 1.0\text{--}1.5 \text{ \AA}$. This initial model provided adequate phasing for refinement and location of other atoms in Fourier maps. The structure was refined to $R = 0.071$, $wR = 0.023$ for 1029 data; for

Table 3. Crystal Data, (TMTSF)Br_{0.82} and (TMTSF)(SCN)_{0.50}

(TMTSF)(Br)_{0.82}

$$a = 9.798 \text{ (2)}$$

$$b = 23.837 \text{ (5)}$$

$$c = 7.095 \text{ (1)}$$

$$v = 1657.1 \text{ \AA}^3$$

Space Group = Cmcn

$$\text{F.W. (C}_{10}\text{Se}_4\text{H}_{12}\text{Br}_{0.82}) = 513.6$$

z = 4 formula units per cell

$$d_{\text{calc}} = 2.058 \text{ g/cm}^3$$

$$\mu \text{ (absorption coefficient)} = 145.2 \text{ cm}^{-1} \text{ (CuK}\alpha\text{)}$$

(TMTSF)(SCN)_{0.50}

$$a = 9.919 \text{ (3)}$$

$$b = 24.124 \text{ (14)}$$

$$c = 7.220 \text{ (6)}$$

$$v = 1729.3 \text{ \AA}^3$$

Space Group = Cmcn

$$\text{F.W. (C}_{10.5}\text{Se}_4\text{H}_{12}\text{N}_{0.5}\text{S}_{0.5}) = 494.5$$

z = 4 formula units per cell

$$d_{\text{calc}} = 1.899 \text{ g/cm}^3$$

$$\mu \text{ (absorption coefficient)} = 122.3 \text{ cm}^{-1} \text{ (CuK}\alpha\text{)}$$

Space Group of modulation Cmc2₁

755 reflections having $F^2 > 3\sigma$, $R = .061$, $wR = .023$. During refinement, four peaks on the Fourier map of the $x=0$ plane indicated the incorporation of a disordered small molecule in the structure. These were refined as partially occupied carbons and nitrogens. Figure 3 shows the structure in the $z = 1/4$ plane. Figure 4 shows a projection showing overlap of consecutive TMTSF molecules view down z ; and Table 4 shows bond angles and lengths in the molecule. Table 5 contains atom coordinates and temperature factors. Most shifts were less than $\sigma/2$, all other shifts were less than σ at refinement, except for the coordinates of Br^- ion, which has very high thermal motion in the z direction, and can be fit equally well by constraining it to the $z = 1/4$ plane or placing it in two-fold positions above and below this plane. Hydrogen positions refined satisfactorily but their isotropic temperature factors could not be refined. The deviation of C-H bond lengths and angles from expected values may be due to torsional motion of the methyl groups. A difference map, based on refined coordinates of non-hydrogen atoms, through the plane of the hydrogen atoms, is shown in Figure 5.

Structure determination and refinement - $\text{TMTSF}(\text{SCN})_{0.50}$

The $\text{TMTSF}(\text{SCN})_{0.50}$ data were phased by placing TMTSF molecules at positions equivalent to those in the refined $\text{TMTSF}(\text{Br})_{0.82}$ structure. After several least squares refinement cycles, the thiocyanates could be seen in difference Fourier maps. They lie

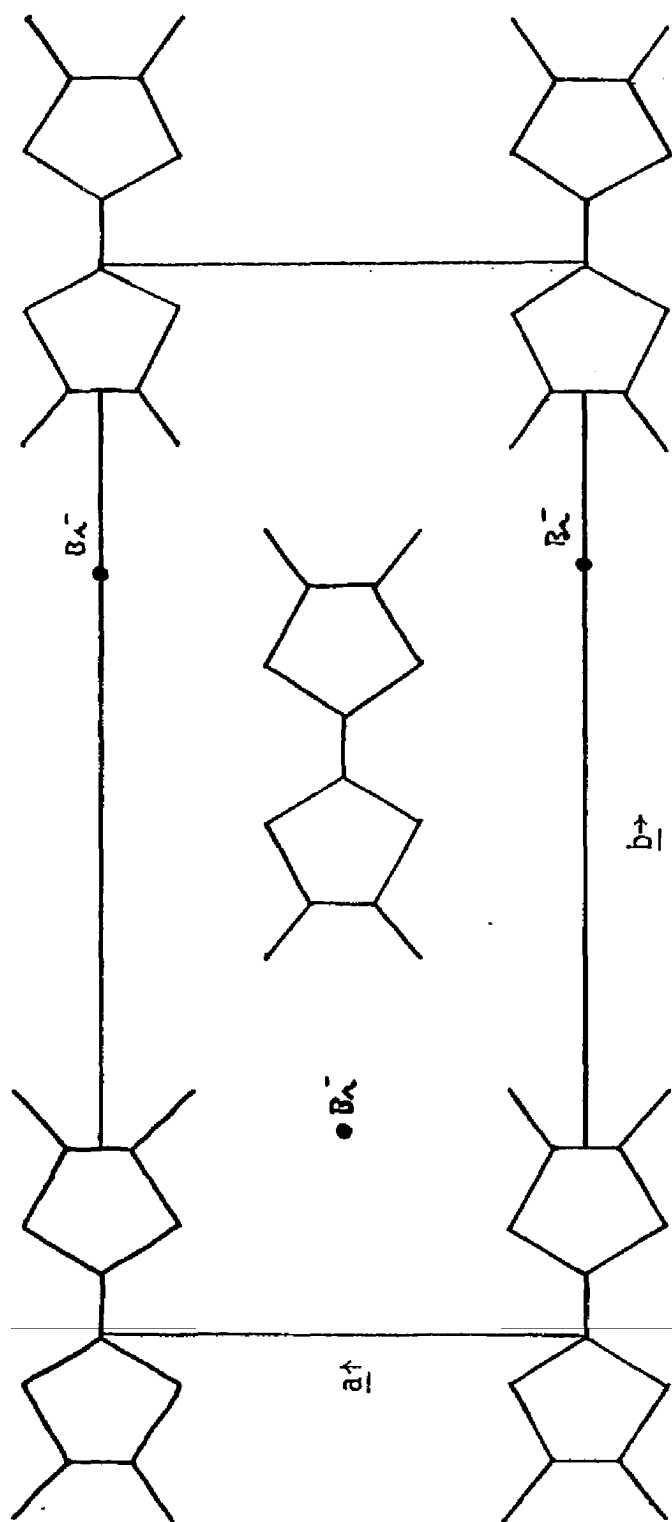


Figure 3. $z = 1/4$ plane of $(\text{TMTSF})\text{Br}_{0.8}$ structure. $(\text{TMTSF})(\text{SCN})_{0.5}$ is isostructural.

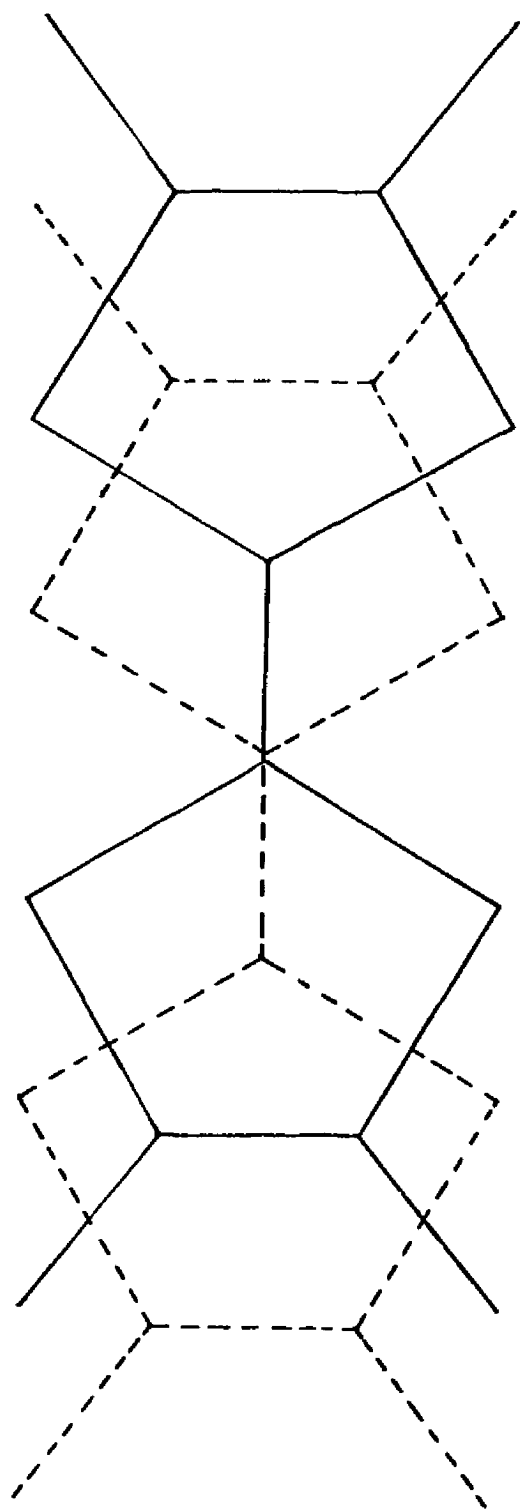


Figure 4. Overlap of consecutive TMTSF cations in $(\text{TMTSF})\text{Br}_{0.8}$. View is down the z axis.

Table 4. Bond Lengths and Angles in (TMTSF)Br_{0.8}

C1 - C2	1.406 (22)
C1 - Se1	1.861 (13)
C2 - Se2	1.845 (17)
Se1 - C3	1.871 (16)
Se2 - C4	1.889 (16)
C3 - C5	1.519 (24)
C4 - C6	1.542 (24)
C3 - C3'	1.340 (36)
C4 - C4'	1.320 (36)
C5 - H1	1.24 (18)
C5 - H2	1.12 (18)
C6 - H3'	1.11 (18)
C6 - H4	1.11 (18)

<C1 - C2 - Se2	121.9 (.1)
<C2 - C1 - Se1	122.2 (.1)
<Se1 - C1 - Se1	115.6 (.2)
<Se2 - C2 - Se2	116.0 (.2)
<C1 - Se1 - C5	93.3 (1.0)
<C2 - Se2 - C4	93.2 (1.0)
<Se1 - C3 - C3'	118.9 (1.0)
<Se2 - C4 - C4'	118.7 (1.0)
<Se1 - C3 - C5	115.5 (1.0)
<Se2 - C4 - C6	114.2 (1.0)
<C3 - C5 - H1	94 (10)
<C 3 - C5 - H2	122 (10)
<C4 - C6 - H4	109 (10)

Table 5. Refined Atomic Coordinates and Temperature Factors of (TMTSF)(Br)_{0.8}

	x	y	z	U11	U22	U33	U12	U13	U23
Se 1	16068 (14)	9935 (5)	25000	313 (7)	359 (6)	538 (9)	-20 (6)	0	0
Se 2	15994 (13)	-4216 (5)	25000	324 (7)	359 (6)	540 (9)	36 (5)	0	0
C 1	0	5763 (56)	25000	286 (83)	248 (64)	369 (102)	0	0	0
C 2	0	-130 (73)	25000	214 (76)	391 (78)	460 (112)	0	0	0
C 3	6759 (126)	16826 (44)	25000	435 (72)	376 (55)	527 (84)	-52 (51)	0	0
C 4	6689 (131)	-11167 (45)	25000	494 (78)	376 (55)	444 (80)	9 (55)	0	0
C 5	15622 (157)	22005 (45)	25000	673 (92)	238 (49)	1076 (131)	-178 (64)	0	0
C 6	16145 (150)	-16310 (49)	25000	523 (78)	380 (56)	628 (91)	68 (60)	0	0
H 1	26450 (1000)	19750 (560)	25000	B = 6.00					
H 2	14380 (910)	24880 (370)	13610 (1000)	B = 6.00					
H 3	2535 (910)	34320 (350)	11820 (1000)	B = 6.00					
H 4	1485 (1000)	-18640 (560)	25000	B = 6.00					
Br ⁻	50000	17573 (20)	23600	589 (23)	755 (24)	5269 (148)	0	0	3996 (382)
<u>Solvent</u>									
C7	0	41050 (80)	17800 (290)	B = 4.50					
C8	0	45730 (140)	5240 (550)	B = 4.50					
C9	0	50670 (170)	10530 (590)	B = 4.50					
C10	0	51960 (170)	25000	B = 4.50					

All coordinates have been multiplied by 10⁵; all Uij's by 10⁴.

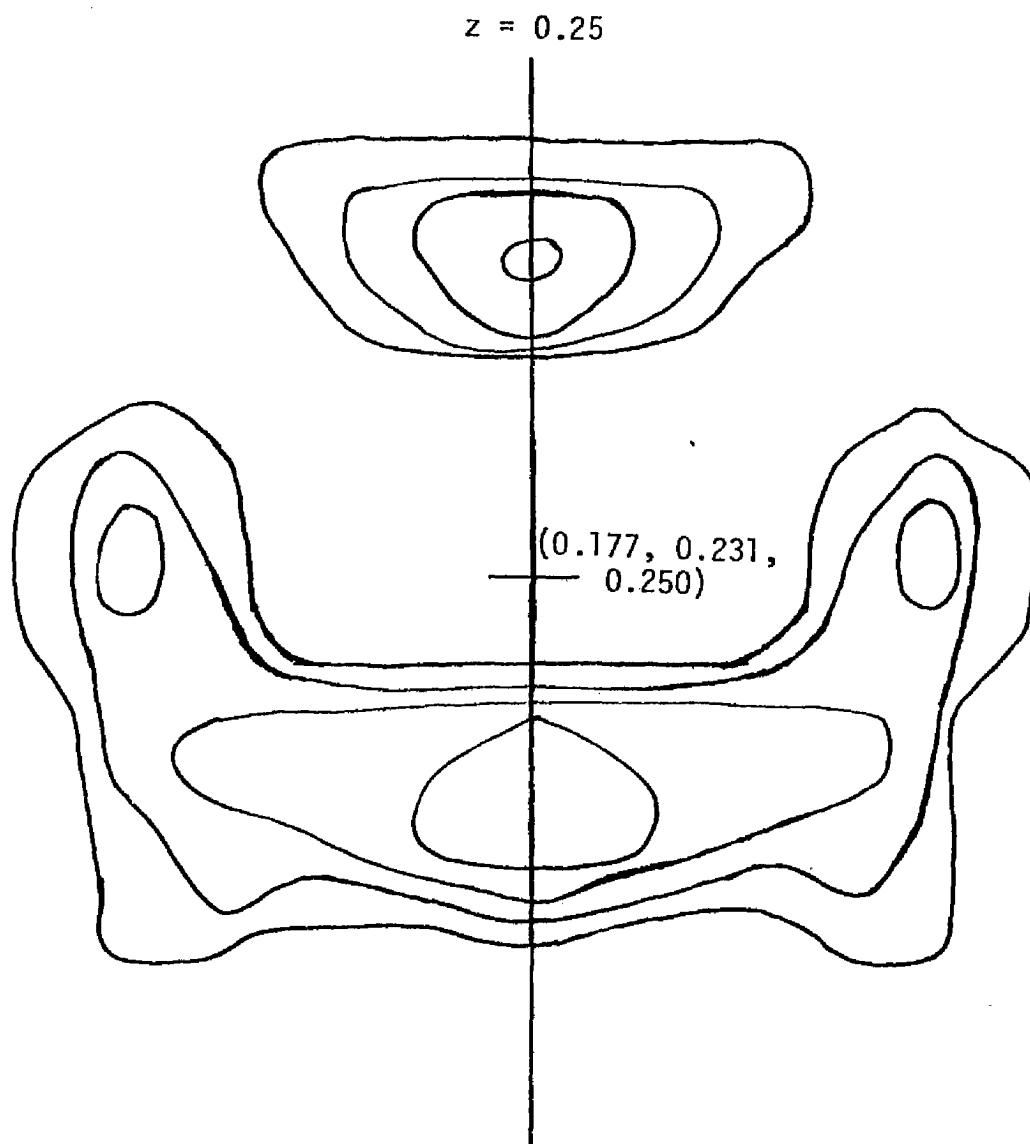


Figure 5. Difference map through hydrogens of methyl carbon (5).

$F_{\text{(calc)}}$ based on refined positions of all non-hydrogen atoms. Contours are in steps of $.1 \text{ e}/\text{\AA}^3$, with outermost contour at $.3 \text{ e}/\text{\AA}^3$. Scale is $4.9 \text{ cm}/\text{\AA}$

across mirror planes at 1/4 and 3/4, and there is two-fold disorder between sulfur and nitrogen. Electron density in the difference map also shows the presence of a disordered small molecule similar to that seen in $\text{TMTSF}(\text{Br})_{0.82}$. Hydrogen atom positions in $\text{TMTSF}(\text{SCN})_{0.50}$ are even less satisfactory than in the bromide. The structural refinement proceeded to $R = 0.054$, $wR = 0.0124$ for all data; $R = 0.042$, $wR = 0.0118$ for reflections with $F^2 \geq 3\sigma$. The temperature factors of the thiocyanate are fairly large, reflecting disorder and perhaps the modulation of the thiocyanate position. At the level of refinement achieved, all non-hydrogen coordinate and temperature factor shifts are less than $\sigma/4$; hydrogen coordinate shifts are less than σ and hydrogen isotropic temperature factors could not be refined.

Atom coordinates and temperature factors of the refined structure of $(\text{TMTSF})(\text{SCN})_{0.50}$ are given in Table 5. Bond angles and lengths of the TMTSF cation are shown on an ORTEP plot (Figure 6).

Analysis of the satellite reflections

Only the $hk1'$ and $hk2'$ ($C' = 33.815 \text{ \AA}$) reflections were used in the analysis of the structure modulation. The pattern of intensity, with respect to indices h and k , is about the same for the $hk1'$ and $hk2'$ and the combination reflections resulting from both lattices. A Patterson map based on $hk1'$ and $hk2'$ data shows a number of peaks with interatomic vectors having x -component $= 0$. Only one (independent) relatively weak peak lies off of the $x = 0$

Table 6. Refined Atomic Coordinates and Temperature Factors for TMTSF(SCN)_{0.50}

	x	y	z	U11	U22	U33	U12	U13	U23
Se 1	15849(13)	9724(4)	25000	470(6)	521(6)	644(8)	-38(5)	0	0
Se 2	15855(12)	-4293(4)	25000	413(5)	519(6)	628(7)	33(4)	0	0
C 1	0	5617(55)	25000	468(73)	529(70)	434(79)	0	0	0
C 2	0	-139(57)	25000	384(58)	506(64)	509(83)	0	0	0
C 3	6744(107)	16531(40)	25000	626(62)	510(46)	631(68)	-21(45)	0	0
C 4	6595(100)	-11165(37)	25000	528(56)	470(43)	609(63)	51(41)	0	0
C 5	15512(233)	21669(63)	25000	1097(106)	651(69)	858(98)	-271(82)	0	0
C 6	15495(134)	-16238(45)	25000	556(55)	545(51)	748(76)	117(55)	0	0
H 1	2829	1794(47)	2500	B = 6.00					
H 2	1609(92)	2204(34)	1531	B = 6.00					
H 3	2205(70)	-1508(28)	1567	B = 6.00					
H 4	748(98)	-1902(46)	2500	B = 6.00					
<u>SCN⁻</u>									
C 7	0	32400(131)	75000	310(141)	526(150)	1159(334)	0	0	0
S/N	0	33333(88)	58084(299)	484(57)	2421(207)	1803(170)	0	0	-711(169)
<u>Solvent</u>									
C 8	0	4623(19)	4607(69)	B = 10.15(1.75)					
C 9	0	4798(22)	7500	B = 10.68(2.17)					
C 10	0	4119(11)	1775(40)	B = 9.28(0.97)					
C 11	0	5111(19)	1000(88)	B = 8.91(1.94)					

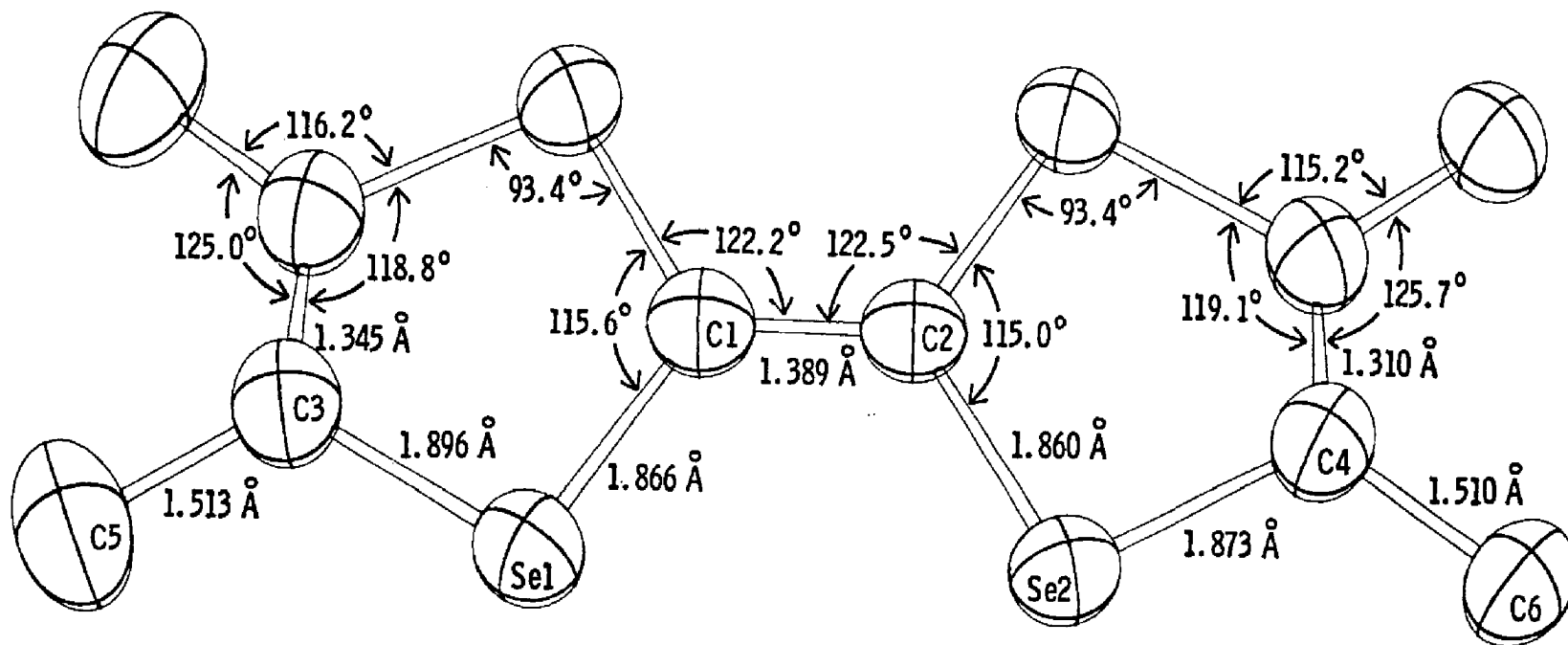


Figure 6. Bond angles and distances of the tetramethyltetraselenofulvalene cation in $(\text{TMTSF})(\text{SCN})_{0.5}$. Thermal ellipsoids are drawn at the 50% probability level.

plane of the Patterson. The strongest vector is (0.0, 0.025, 0.5) which was interpreted as a vector between the extreme positions of a modulated thiocyanate chain. It appears that a slight displacement in the y coordinates of the thiocyanate anions occurs with a period of 32.815 Å. This model of the modulation, in Space Group $Cmc2_1$, along with fitting of several smaller peaks, results in $R = .397$ for 54 reflections with $F^2 > 3\sigma$. A Fourier map based on this unrefined model is shown in Figure 7. Larger peaks ($.5 \text{ e}/\text{\AA}^3$) at (0.0, 0.333, 0.0), (0.0, 0.307, 0.5), and (0.0, 0.353, 0.5) are associated with the thiocyanate ions. A smaller peak ($.2 \text{ e}/\text{\AA}^3$) at (0.0, 0.463, 0.0) is associated with an atom in the disordered solvent molecule. Small peaks are associated with the TMTSF cations, in the region of the C=C bonds and methyl groups, but these cannot be identified with particular atoms. The atomic coordinates and temperature factors used to model the satellite data are given in Table 7.

Discussion

The TMTSF cations in $(\text{TMTSF})(\text{Br})_{0.82}$ and $(\text{TMTSF})(\text{SCN})_{0.50}$ are slip-stacked and have their molecular planes perpendicular to \vec{c}_0 , as shown in Figure 8a. This method of stacking is quite different than the stacking of TTF cations in TTF halides (8b) and $(\text{TTF})(\text{TCNQ})$ (8c). The stacking is rather similar to that observed in tetra-

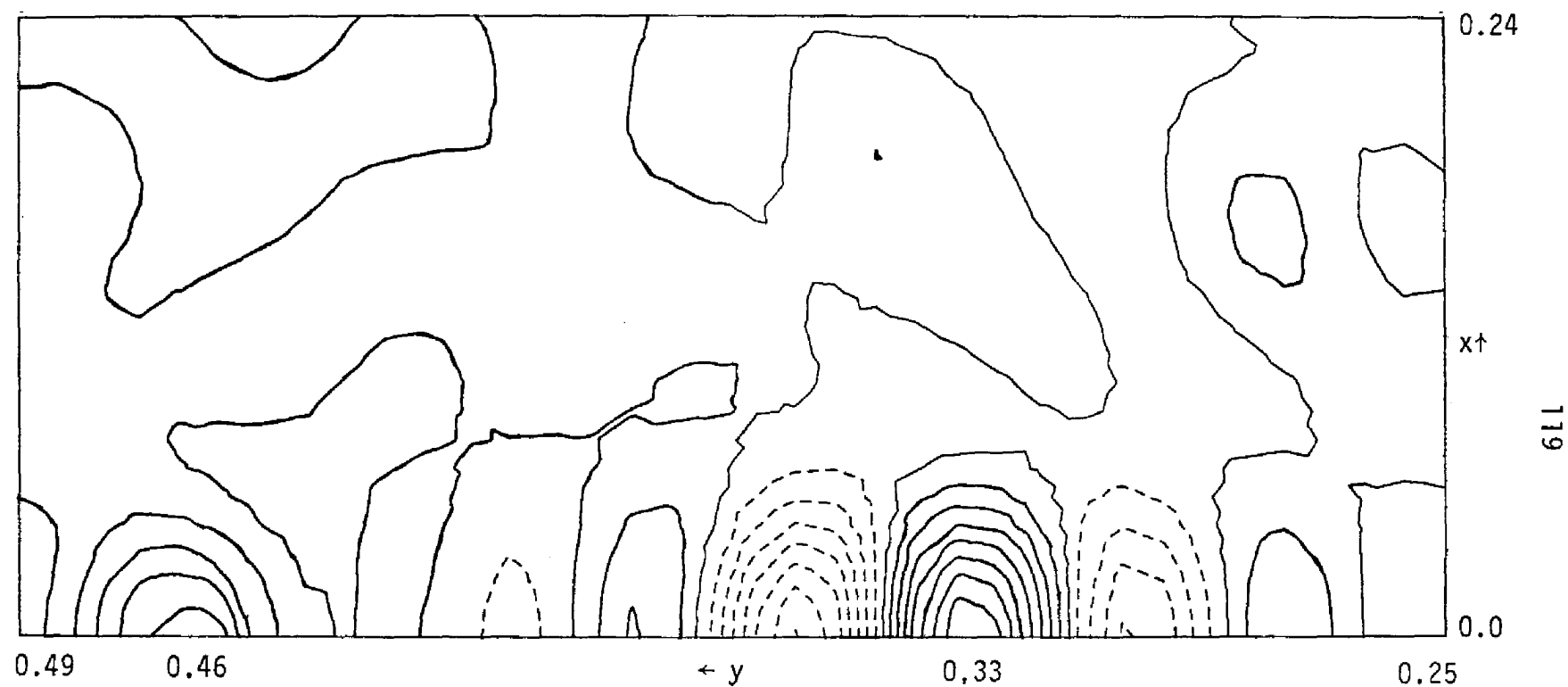


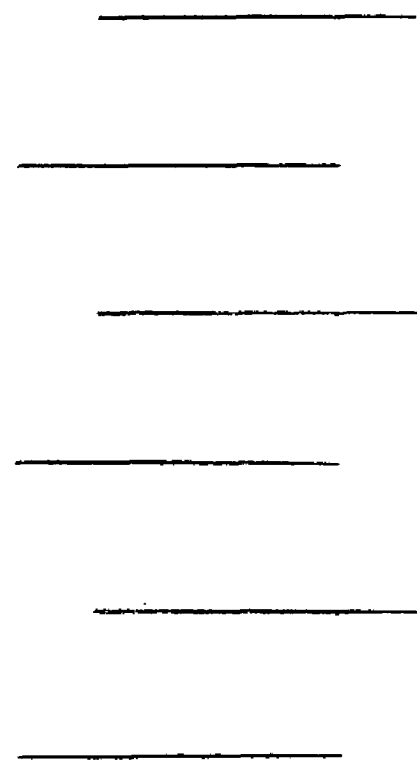
Figure 7. Electron density map of satellite reflection data ($z=0$). Contour intervals are at $\sim .05 \text{ e}/\text{\AA}^3$. Scale is $\sim 3.44 \text{ cm}/\text{\AA}$. Dotted contours represent positive electron density at $z = 0.5$; solid contours at $z = 0.0$. Strong peaks at and near $y = 0.33$ are due to the thiocyanate ion; peak at $y = 0.46$ is an atom in the solvent molecule.

Table 7. Atomic Coordinates and Temperature Factors used to model (TMTSF)(SCN)_{0.5} Structure Modulation

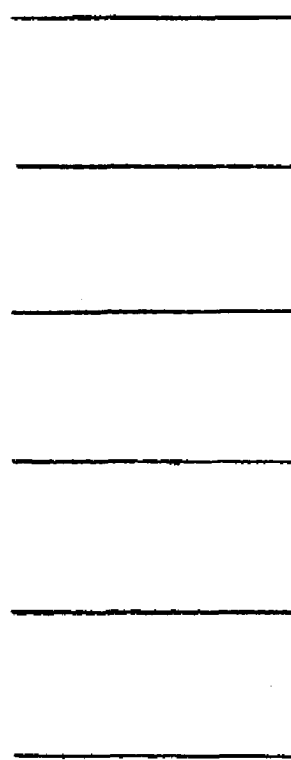
	x	y	z	U11	U22	U33	Population
S 1	0	3333	0	498	443	321948	1.9
S 2	0	3066	5000	498	443	321948	0.9
S 3	0	3528	5000	498	443	321948	1.5
C 1	0	4628	0	498	443	321948	2.0
C 2	0	1580	5000	498	443	321948	1.4
C 3	0	150	5000	498	443	321948	1.5
C 4	1850	1600	5000	498	443	321948	0.7

$U_{ij} = 0$ for $i \neq j$.

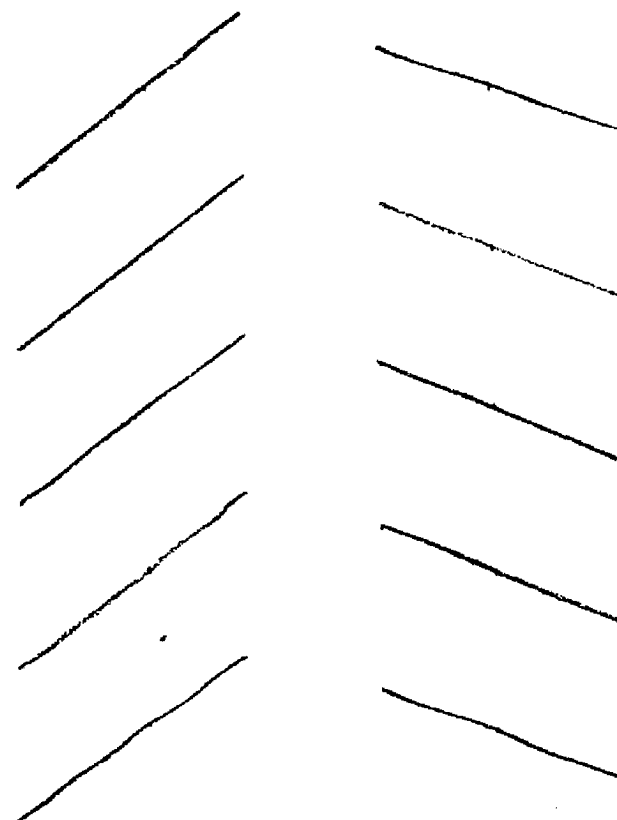
All parameters have been multiplied by 10^4 .



a) $(\text{TMTSF})\text{Br}_{0.8}$ and
 $(\text{TMTSF})(\text{SCN})_{0.5}$



b) $(\text{TTF})\text{Cl}_{0.67}$



c) $(\text{TTF})(\text{TCNQ})$

Figure 8. Stacking in some organic conductors.

methyltetrathiafulvalene bromide, $(\text{TMTTF})(\text{Br})_{0.5}$; where only a slight tilt away from perpendicularity to the stacking axis is observed (5).

Interchain coupling should be quite low in $(\text{TMTSF})(\text{Br})_{0.82}$ and $(\text{TMTSF})(\text{SCN})_{0.50}$. There are no short contacts between stacks, and van der Waals contacts (3.77 \AA in the bromide) only occur between methyl groups (see Figure 3). TMTSF salts, including those with TCNQ, generally have longer interchain contacts than are observed for TTF and HMTSF, for example (4).

High disorder exists in both $(\text{TMTSF})(\text{Br})_{0.82}$ and $(\text{TMTSF})(\text{SCN})_{0.50}$. In the former, the bromide ion exhibits very high thermal parameters in the z direction. In the latter, the orientation of the thiocyanates is two-fold disordered. In both compounds, anion sites are not fully occupied, and a partially occupied small molecule (solvent) site exists in the $x=0$ plane.

The period of the modulation of the $(\text{TMTSF})(\text{SCN})_{0.50}$ structure is similar to the periods of incommensurate charge density waves observed in $\text{K}_2\text{Pt}(\text{CN})_4\text{Br}_{0.30} \cdot x\text{H}_2\text{O}$, $(\text{TTF})(\text{TCNQ})$, $(\text{TSeF})(\text{TCNQ})$, $(\text{HMTSF})(\text{TCNQ})$, and $(\text{HMTTF})(\text{TCNQ})$ (12). It is not unreasonable to interpret the modulation of the SCN^- chain as a response to a charge density wave on the cation column.

While the electronic transport properties of $(\text{TMTSF})(\text{Br})_{0.82}$ and $(\text{TMTSF})(\text{SCN})_{0.50}$ are still being investigated, several predictions may be made on the basis of the structural properties. Exchange

of anions between sites cannot occur, so that some disorder will be retained at low temperature. Thiocyanate orientational disorder will also be retained, while the high thermal motion of the bromide ions should decrease. Disorder in $(\text{TMTSF})(\text{SCN})_{0.50}$ may be expected to result in retention of moderate conductivity to lower temperatures than in the bromide.

The 32.8 Å superperiod in $(\text{TMTSF})(\text{SCN})_{0.50}$ may be tentatively identified as a distortion resulting from an incommensurate charge density wave. If this is true, then $2k_F = .215 c^*$, which implies that the average charge on each TMTSF is +0.43. This is reasonable in view of the uncertainty in the stoichiometry of this material, and the possibility of back-charge transfer ($\text{SCN}^- + \text{TMTSF}^+ \rightarrow \text{SCN}^0 + \text{TMTSF}^0$).

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APPENDIX 1

Structure Factors Tables

The following tables contain the observed and calculated structure factors for refined and partly refined structures reported in this thesis. The heading above each group gives the indices for that group of reflections, with one index indicated by a letter. That index varies and is given in the first column. F_{obs} is given in the second column, F_{calc} in the third, and the standard deviation in the fourth. A negative sign on F_{obs} means that the observed intensity, $I \sim F_{\text{obs}}^2$, was negative.

A. $[\text{Rh}(\text{CNCHCH}_2)_4]\text{ClO}_4$ - Structure factors, room temperature data
obtained with Ni-filtered $\text{CuK}\alpha$ radiation. Space group Imm.

				26	139	126	6	23	193	205	8	22	147	127	7	4	285	244	7	
	0	K	0	28	84	91	8	25	130	131	8	24	46	91	15	6	281	259	7	
								27	55	84	11					8	275	248	7	
2	2492	3018	16		0	K	3						0	K	9	10	216	206	8	
4	3069	2963	33						0	K	6					12	189	174	8	
6	1479	1253	11	1	1657	1520	12					1	749	727	7	14	147	142	8	
8	2505	2591	20	3	1303	1246	9	0	1717	1600	13	3	725	716	7	16	107	97	9	
10	1525	1422	11	5	2131	2009	15	2	1300	1323	9	5	674	667	7	18	51	80	12	
12	1534	1517	11	7	2173	2187	15	4	1153	1054	9	7	541	493	6					
14	985	1024	8	9	1502	1509	11	6	930	991	8	9	380	352	7		0	K	13	
16	876	826	8	11	914	896	7	8	1166	1115	9	11	347	380	7					
18	494	520	6	13	951	864	8	10	1096	1101	9	13	369	395	7	1	202	183	8	
20	453	461	6	15	806	825	7	12	803	787	7	15	333	301	7	3	226	151	7	
22	310	330	7	17	613	524	6	14	487	498	6	17	234	218	7	5	230	202	7	
24	195	215	8	19	288	365	7	16	353	427	7	19	169	169	8	7	223	194	7	
26	129	127	9	21	320	305	7	18	312	323	7	21	114	121	9	9	172	168	8	
28	65	96	11	23	205	237	8	20	271	277	7	23	56	90	12	11	157	144	8	
				25	163	157	8	22	207	204	7					13	130	110	8	
	0	K	1	27	86	102	10	24	123	140	9		0	K	10	15	80	78	9	
								26	82	89	8									
1	1710	2542	13		0	K	4					0	536	563	6		0	K	14	
3	1716	1695	11						0	K	7		2	538	533	6				
5	2009	1520	16	0	2345	2363	16					4	466	467	6	0	163	150	8	
7	1223	1251	9	2	1029	1082	8	1	1091	1163	9	6	368	359	7	2	142	152	9	
9	1625	1515	12	4	1828	1843	13	3	979	994	7	8	277	312	8	4	171	157	7	
11	1446	1383	11	6	2024	1965	16	5	866	890	7	10	329	362	7	6	156	149	8	
13	1212	1122	9	8	1661	1590	12	7	941	914	8	12	344	326	7	8	142	131	8	
15	903	910	8	10	1131	1112	8	9	843	835	7	14	285	236	7	10	105	112	9	
17	736	627	7	12	1026	1110	8	11	649	604	6	16	188	201	8	12	56	87	13	
19	426	490	6	14	798	787	7	13	487	467	6	18	156	156	8					
21	395	385	6	16	559	566	6	15	394	429	7	20	109	109	9		0	K	15	
23	214	267	8	18	302	331	7	17	363	350	7	22	53	86	11					
25	164	165	8	20	287	323	7	19	282	263	7					1	105	68	9	
27	89	108	10	22	253	259	7	21	180	196	9		0	K	11	3	99	89	9	
				24	178	188	8	23	135	136	8					5	96	87	9	
	0	K	2	26	117	115	8	25	88	91	8	1	445	465	6	7	79	62	9	
												3	396	408	6	9	56	75	11	
0	2977	3388	17		0	K	5					5	314	343	7					
2	1593	1432	12						0	K	8		7	314	336	7		1	K	0
4	4008	3495	25	1	1594	1723	12	0	1066	1079	9	9	316	334	7					
6	1315	1291	10	3	1403	1447	10	2	973	952	8	11	277	263	7	1	459	448	4	
8	1761	1770	14	5	1355	1312	10	4	886	879	7	13	195	200	9	3	75	135	8	
10	1320	1192	10	7	1285	1244	9	6	835	853	7	15	151	175	10	5	482	470	4	
12	1154	1101	5	9	1234	1366	10	8	700	649	7	17	144	127	8	7	279	280	5	
14	872	825	7	11	1167	1201	9	10	524	460	6	19	91	90	9	9	211	219	5	
16	817	793	7	13	836	800	7	12	392	428	7					11	62	1	15	
18	498	472	6	15	503	503	6	14	408	440	6		0	K	12	13	53	7	19	
20	373	388	6	17	349	362	7	16	366	367	7					15	-29	42	32	
22	274	295	7	19	287	322	7	18	271	245	7	0	345	271	7	17	112	67	14	
24	165	203	9	21	295	278	7	20	181	181	8	2	313	256	7	19	69	49	21	

21	+1	2	29		1	K	4		8	-31	31	32	4	13	15	46	17	25	89	37
23	49	7	24						10	42	65	28	6	-18	2	41	19	140	72	12
				1	181	190	7	12	38	8	31	8	-42	7	28	21	17	12	42	
	1	K	1		3	203	198	6	14	-39	15	32	10	46	22	25	23	47	0	24
				5	222	210	5	16	-18	46	42	12	72	10	17					
0	+34	+33	4	7	176	129	6	18	-46	37	27	14	-34	13	28		2	K	2	
2	1347	1292	10	9	39	13	22	20	-50	21	23									
4	647	656	5	11	77	60	15						1	K	12		0	2159	2128	15
6	926	627	7	13	28	61	33		1	K	8					2	781	727	6	
8	120	64	8	15	49	25	26					1	27	21	34	4	556	460	5	
10	303	335	5	17	54	51	26	1	44	30	26	3	64	21	21	6	381	362	+	
12	101	172	8	19	-38	37	30	3	-55	1	21	5	-59	26	21	8	162	156	7	
14	100	140	13	21	38	15	29	5	-25	12	35	7	46	32	25	10	143	188	8	
16	124	97	12	23	41	15	25	7	-66	6	20	9	-10	27	44	12	92	83	13	
18	81	70	19					9	28	39	37	11	39	17	26	14	97	80	14	
20	23	45	39		1	K	5	11	-16	49	46					16	-34	32	33	
22	-42	6	28					13	-44	11	29		1	K	13		18	134	134	13
24	35	12	27	0	347	335	5	15	49	54	26					20	-30	16	34	
				2	358	159	5	17	86	44	17	0	28	3	32	22	-54	9	23	
	1	K	2	4	-15	60	34	19	50	25	23	2	-56	11	21	24	31	13	28	
				6	39	131	22					4	-31	17	30					
1	228	211	4	8	76	49	14		1	K	9	6	-54	29	20		2	K	3	
3	1054	1069	8	10	97	54	13					8	82	22	14					
5	694	677	6	12	93	94	15	0	50	34	25					1	334	251	5	
7	263	265	5	14	-70	61	20	2	80	56	18		2	K	0	3	278	159	5	
9	-14	46	33	16	55	6	25	4	33	61	33					5	353	382	5	
11	93	102	11	18	51	7	25	6	51	63	26	0	575	579	5	7	-79	419	5	
13	18	68	35	20	-49	28	25	8	-57	30	24	2	440	422	5	9	41	45	22	
15	36	42	29	22	11	10	43	10	74	12	20	4	2355	2094	16	11	-59	68	19	
17	143	114	11					12	-7	37	51	6	84	207	12	13	56	46	21	
19	47	65	28		1	K	6	14	80	45	18	8	528	467	5	15	71	96	21	
21	-71	7	20					16	59	33	21	10	-50	2	19	17	52	58	26	
23	75	4	17	1	29	23	27	18	-31	25	30	12	110	142	11	19	49	42	26	
				3	102	50	11					14	89	30	15	21	-64	7	21	
	1	K	3	5	38	3	24		1	K	10	16	-38	73	30	23	-19	13	37	
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2	744	777	6	11	88	61	16	3	92	76	17	22	-26	10	35					
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6	238	215	5	15	78	12	20	7	45	7	29					2	100	77	11	
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10	49	98	19	19	69	38	20	11	42	36	28					6	405	379	5	
12	94	99	13	21	65	17	19	13	59	25	22	1	458	454	4	8	93	51	12	
14	56	70	22					15	38	19	28	3	915	773	7	10	26	7	30	
16	97	107	16		1	K	7	17	25	25	31	5	161	166	7	12	54	57	22	
18	85	53	18									7	196	220	6	14	102	111	15	
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22	46	5	25	2	56	82	19					11	103	55	11	18	-61	43	23	
24	-74	24	16	4	50	50	22	0	48	11	28	13	39	21	26	20	-17	7	42	
				6	65	105	19	2	67	3	21	15	45	15	26	22	-34	17	30	

21	134	131	8	9	213	200	7	3	166	123	7	16	-58	0	24	10	-35	15	33
23	81	93	8	11	178	161	7	5	89	3	12	18	73	52	19	12	53	6	25
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	3	K	9	15	97	102	9	9	149	93	9				16	-45	33	26	
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6	473	492	7	0	172	157	9	19	60	35	23	5	78	47	17	3	-23	41	41
8	379	344	7	2	194	159	8	21	44	0	26	7	69	85	20	5	-28	20	37
10	282	297	7	4	190	168	8					9	+3	26	29	7	-40	14	30
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15	195	181	8									16	40	2	29				
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19	112	113	8	2	83	81	8	3	174	163	8	20	-17	16	37	3	-27	12	35
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2	343	362	7	0	876	932	7	13	61	29	23	3	38	60	31				
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18	89	91	9	16	-29	17	36	0	344	378	5						5	K	0
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1	287	230	7					8	72	58	19	2	56	41	25	5	296	302	6
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17	178	181	7									10	53	4	23	5	116	120	15
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	6	K	8	4	144	139	8	4	86	73	20					11	9	24	49
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16	152	182	8					1	71	38	25	4	47	4	24				
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	6	K	9	1	160	124	12	7	39	30	12	8	K	0		1	-10	9	52
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15	171	163	6					2	67	38	24					2	65	3	23
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3	197	196	7	5	-59	28	27					8	-28	10	38	9	282	297	7
5	158	176	8	7	67	28	24		7	K	7	10	16	14	43	11	264	278	7
7	155	162	8	9	-42	25	31					12	-40	1	29	13	213	250	7
9	150	155	8	11	49	33	27	0	-39	4	32					15	175	209	8
11	140	139	7	13	-77	26	19	2	67	1	22		8	K	3				
13	104	110	7	15	-59	16	22	4	45	14	29						9	K	1
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	6	K	12		7	K	3	8	34	11	32	3	-8	58	55	0	429	465	7

2	463	476	7	11	226	235	8		10	224	235	7	4	194	210	7
4	356	370	7	13	195	205	8	1	12	173	185	8	6	186	192	7
6	332	341	7	15	152	170	8	3					8	162	174	7
8	296	294	7					5		9	K	6	10	140	157	7
10	279	282	7		9	K	3	7								
12	228	244	7					9	1	242	268	7		9	K	8
14	188	222	8	0	290	301	8	11	3	228	251	8				
16	135	172	8	2	334	321	7	13	5	229	243	7	1	181	200	7
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	9	K	2		6	312	329	7		9	184	214	8	5	156	161
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1	376	386	7	10	245	250	8	0								
3	359	353	7	12	202	199	8	2	9	K	7			9	K	9
5	319	314	7	14	150	166	9	4								
7	313	315	7					6				7	0	145	165	6
9	279	274	7		9	K	4	8	2	232	230	6	2	139	156	7

B. $[\text{Rh}(\text{CNCHCH}_2)_4]\text{ClO}_4$ - Structure factors 22° K data obtained with graphite-monochromatized $\text{MoK}\alpha$ radiation. Space group Immm.

[illegible]

12	245	198	53	19	-252	59	86					2	-231	53	82	15	252	7	145
14	116	32	109	21	149	76	116	1	K	8		4	150	61	115	17	262	4	143
16	135	245	115	23	341	59	67					6	41	35	170				
18	99	92	143					1	178	109	81	8	129	23	119		2	K	0
20	-73	95	152		1	K	5	3	-169	77	88	10	116	11	128				
22	402	31	52					5	125	83	109	12	-358	21	74	0	702	524	16
24	-295	76	78		0	273	233	37	7	-66	31	149	14	158	5	115	2	500	371
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	1	K	2		4	150	55	60	11	110	47	139	18	126	1	204	6	-231	178
					6	-133	24	63	13	-137	4	127	20	306	0	130	8	337	382
1	236	275	31	8	163	144	70	15	-271	34	76	22	-343	7	127	10	225	204	51
3	600	993	17	10	-244	67	57	17	-353	34	63	24	388	7	111	12	234	140	56
5	492	503	24	12	86	99	138	19	29	14	198					14	-124	173	112
7	65	577	98	14	99	31	130	21	242	20	92		1	K	12	16	-60	79	168
9	-150	16	65	16	250	84	87	23	216	20	154					18	189	127	107
11	-105	148	89	18	191	43	104					1	-150	67	111	20	276	51	71
13	129	51	93	20	-210	50	96		1	K	9	3	164	61	106	22	87	88	145
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17	195	153	94	24	-328	45	74	0	-269	147	68	7	-363	41	62				
19	-244	79	90					2	-126	91	122	9	102	39	139		2	K	1
21	-168	76	103		1	K	6	4	182	90	93	11	112	26	133				
23	-163	74	112					6	225	68	85	13	181	18	105	1	482	344	24
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4	352	372	33	9	172	29	79	16	266	4	77					11	-242	19	55
6	293	200	34	11	274	15	63	18	304	9	70		1	K	13	13	-43	71	148
8	-49	350	107	13	-130	54	120	20	278	16	78					15	198	167	85
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12	-144	108	91	17	52	58	154	24	504	13	90	2	-114	53	181	19	250	51	79
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16	141	166	113	21	170	47	111		1	K	10	6	300	36	105	23	81	76	153
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					0	251	62	52	7	168	54	100	16	246	8	143	2	644	734
	1	K	4		2	120	44	90	9	-281	43	75	18	187	0	178	4	520	450
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					6	-71	86	129	13	-179	16	102					8	-110	43
1	169	165	51	8	232	29	70	15	102	5	141		1	K	14	10	-157	230	76
3	190	121	50	10	-246	15	75	17	-96	8	148					12	-161	40	85
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15	93	92	136	22	304	31	77					11	-107	18	207	24	259	61	85
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1	197	380	44	8	-123	95	103	15	171	30	105	22	134	6	214	10	1128	1230	21		
3	-5	160	140	10	194	28	82	17	-246	13	85					12	825	930	29		
5	207	83	49	12	-205	2	89	19	-104	4	147		2	K	13		14	771	831	37	
7	315	340	39	14	210	1	96	21	-107	14	210					16	573	515	48		
9	8	121	163	16	-178	52	108	23	-203	17	162		1	-283	56	113	18	395	467	66	
11	-170	59	71	18	-257	29	80						3	-172	50	158	20	407	353	57	
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6	301	280	39	13	-157	18	107	20	-81	15	230	2	-207	40	146	15	762	580	37		
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16	254	93	86	23	-268	25	137					12	83	15	224						
18	-210	64	95					1	185	43	89	14	-127	12	214		3	K	3		
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22	161	63	113					5	165	60	100	18	-224	0	171	0	1086	941	17		
24	160	52	115		0	246	128	69	7	-72	37	154				2	1245	1360	17		
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	2	K	5		4	-231	49	74	11	214	6	91				6	1367	1680	20		
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				16	16	-58	16	235	1	149	10	173	4	440	38	83				
1	161	17	108						3	-205	13	146	6	124	36	181	1	467	105	63
3	-119	28	137		10	K	5		5	140	11	172	8	-160	19	172	3	324	81	80
5	-248	35	87						7	-192	0	164	10	237	6	133	5	356	65	106
7	365	15	85	1	1	-239	5	118					12	-203	11	158				
9	-415	14	84	3	3	-170	25	151									12	K	2	
11	252	21	121	5	5	330	34	99	10	K	10		11	K	4		0	322	89	85
13	286	33	114	7	7	134	13	181	0	-365	18	108	1	199	18	149	2	235	73	109
15	240	39	132	9	9	-359	7	99	2	-81	14	224	3	-489	5	80	4	352	62	77
17	-339	22	110	11	11	79	16	231					5	264	18	126				
				13	13	-646	0	72	11	K	0		7	-411	17	91	12	K	3	
	10	K	2		15	-254	16	138					9	-216	19	151	1	545	69	71
									1	53	42	227	11	-212	18	148				
0	244	16	93		10	K	6		3	-274	21	116								

C. (TTF)Cl_{0.67} - Structure factors, room temperature P4₂/mmm phase.Ni-filtered CuK α radiation.

				3	710	629	5	5	68	57	2	3	46	25	2	6	36	31	2
	H	O	0		4	40	25	2	6	129	116	2	4	140	137	1	7	12	14
				5	737	693	5	7	80	61	2	5	173	163	1	8	497	455	3
2	515	372		3	6	237	232	1	8	103	91	2	6	78	77	1	9	76	74
4	1334	1256		10	7	304	308	2	9	61	54	1	7	87	79	2	10	66	62
6	880	887		5	8	170	178	1	10	58	59	1	8	362	341	3	11	24	0
8	25	1		4	9	19	38	9					9	65	62	2	12	122	120
10	282	273		2	10	105	103	1		H	1	4	10	55	51	2	13	27	24
12	24	19		3	11	86	77	1					11	30	27	1			
14	6	3		3	12	42	38	2	1	103	90	2	12	89	88	1		H	3
					13	101	104	1	2	41	42	2							
	H	O	1		14	29	34	1	3	33	51	2		H	2	3		3	625
									4	21	7	5					4	29	14
1	1606	1551		8		H	1	1	5	148	98	1	2	67	63	2	5	43	44
3	40	35		2					6	49	33	2	3	277	263	2	6	79	68
5	1135	1108		12	1	709	690	6					4	24	26	6	7	303	293
7	54	59		1	2	272	281	2		H	2	0	5	52	49	3	8	55	51
9	115	102		1	3	-8	5	7					6	104	85	2	9	136	131
11	136	132		2	4	506	490	3	2	528	596	4	7	207	186	2	10	24	21
13	48	49		1	5	143	147	1	3	111	105	1	8	42	32	3	11	17	11
					6	319	312	2	4	31	10	3	9	65	63	1	12	17	1
	H	O	2		7	165	170	2	5	347	331	2	10	30	27	1			
					8	227	228	2	6	31	18	2						H	3
0	1441	1437		5	9	145	143	1	7	126	122	1		H	2	4			
2	44	44		2	10	176	163	2	8	508	482	4					2	281	263
4	498	512		4	11	44	38	2	9	56	101	1	2	78	87	1	3	25	21
6	422	415		3	12	50	47	1	10	121	127	1	3	29	2	4	4	163	141
8	59	52		2	13	39	42	1	11	44	42	2	4	42	13	5	5	52	43
10	151	141		2					12	131	122	1	5	52	39	3	6	32	16
12	14	2		3		H	1	2	13	29	31	1	6	32	5	2	7	15	2
									14	43	39	1					8	191	150
	H	O	3		1	462	484	3						H	3	0		9	29
					2	229	221	2		H	2	1					10	21	20
1	421	453		6	3	152	158	1					3	1446	1339	9			
3	49	12		2	4	22	8	4	2	219	201	1	4	84	64	1		H	3
5	403	384		4	5	482	475	3	3	763	756	6	5	76	74	1			
7	34	17		4	6	138	132	1	4	92	88	1	6	139	123	1	3	136	147
9	38	35		2	7	124	117	1	5	138	128	1	7	556	572	4	4	25	5
					8	113	106	1	6	249	229	2	8	87	91	2	5	16	1
	H	O	4		9	28	7	3	7	515	499	4	9	249	262	2	6	-16	20
					10	71	63	2	8	70	74	2	10	24	36	7			
0	296	333		3	11	65	62	1	9	156	154	2	11	23	0	3		H	4
2	30	50		2	12	24	24	2	10	71	72	3	12	15	1	4			
4	131	138		1					11	26	21	3	13	144	149	1	4	152	156
6	131	198		1		H	1	3	12	34	31	2					5	93	100
									13	137	134	1		H	3	1		6	488
	H	1	0		1	126	132	2									7	37	31
					2	65	70	2		H	2	2		3	88	82	1	8	231
1	671	745		5	3	39	14	3						4	464	449	3	9	136
2	642	610		4	4	153	150	2	2	482	476	3	5	151	146	1	10	185	182

D. (1.d.) $(\text{TTT})_2\text{I}_3$ - Structure factors Cmca subcell. Ni-filtered
CuK α radiation.

				10	1058	1063	12				6	148	183	6	9	695	788	9	
	H	0	0	12	924	865	11	0	1408	1335	15				11	278	319	9	
				14	711	665	10	2	1008	950	17	H	1	0	13	-34	68	29	
2	1105	867	8	16	644	609	10	4	588	560	10				15	225	260	13	
4	1849	1670	19	18	534	494	8	6	192	90	16	1	23	0	20	17	152	117	15
6	1346	1147	14	20	447	432	8	8	318	279	12	3	46	0	25	19	87	80	21
8	317	390	7	22	291	292	7	10	678	642	10	5	41	0	21	21	201	167	9
10	2187	2183	22					12	867	772	11	7	28	0	24				
12	1808	1976	20		H	0	8	14	699	657	9	9	33	0	22		H	1	4
14	1607	1631	18					16	292	277	8	11	27	0	26				
16	545	433	11	0	1057	931	12	18	57	90	11	13	-62	0	18	1	1761	1574	18
18	264	242	12	2	1460	1394	16					15	42	0	25	3	320	415	7
20	180	138	12	4	1418	1373	15		H	0	16		-26	0	30	5	432	454	7
22	262	254	8	6	1736	1660	18					19	65	0	24	7	158	130	10
				8	1426	1377	15	0	331	295	12	21	18	0	38	9	1046	1122	12
	H	0	2	10	853	793	11	2	389	359	10					11	278	319	9
				12	673	626	10	4	667	625	10		H	1	1	13	26	50	49
0	261	422	9	14	500	480	10	6	857	800	11					15	457	450	9
6	2840	2836	28	16	623	575	10	8	671	629	10	1	274	326	7	17	198	193	15
8	2600	2648	27	18	614	582	9	10	361	355	8	3	937	1095	10	19	-19	68	46
10	1059	966	12	20	436	447	8	12	195	181	11	5	608	689	7	21	279	267	8
12	351	215	8					14	127	131	13	7	265	275	7				
14	413	335	9		H	0	10	16	245	253	6	9	568	617	8		H	1	5
16	815	775	11									11	415	455	8				
18	947	915	12	0	1949	1902	21		H	0	18		87	72	23	1	947	1070	11
20	886	845	10	2	1750	1708	18					15	383	384	10	3	112	73	15
22	346	331	7	4	560	424	9	0	579	593	10	17	294	310	11	5	144	167	11
				6	864	750	11	2	462	447	10	19	-63	36	26	7	200	208	9
	H	0	4	8	506	390	10	4	355	333	8	21	221	201	8	9	372	374	8
				10	796	746	11	6	374	354	8					11	100	74	14
0	2396	2349	25	12	1522	1535	17	8	256	227	9		H	1	2	13	29	65	48
2	2882	2713	28	14	554	502	11	10	309	314	8					15	233	258	13
4	1694	1558	18	16	590	566	9	12	359	360	6	1	283	346	10	17	-33	27	27
6	858	762	10	18	149	153	16	14	268	266	6	3	654	733	8	19	76	71	24
8	1194	1109	13	20	70	7	21					5	288	332	6	21	26	5	38
10	1341	1297	14						H	0	20		7	100	95	13			
12	1543	1538	16		H	0	12					9	415	479	7		H	1	6
14	1227	1201	14					0	382	372	7	11	259	282	9				
16	692	670	10	0	407	271	10	2	325	296	7	13	125	108	17	1	733	802	9
18	234	217	11	2	419	286	9	4	254	279	8	15	254	257	12	3	1121	1301	12
20	191	158	11	4	1666	1601	18	6	267	287	8	17	182	176	15	5	179	251	10
22	282	276	8	6	1102	1095	13	8	222	218	8	19	36	23	37	7	78	40	20
				8	1272	1206	15	10	248	259	7	21	161	148	11	9	757	826	9
	H	0	6	10	831	795	12	12	213	238	6					11	644	650	9
				12	374	359	13						H	1	3	13	408	426	9
0	1358	1218	15	14	563	523	9		H	0	22					15	330	329	11
2	1966	1917	20	16	289	302	10					1	1255	1450	14	17	377	358	8
4	1710	1621	17	18	552	522	8	0	251	175	7	3	357	417	7	19	105	85	18
6	1623	1593	12					2	411	426	6	5	63	150	18	21	354	311	7
8	1518	1500	16		H	0	14	4	197	195	7	7	176	173	9				

	H	I	7		13	80	34	31	11	116	140	26					0	167	208	9
					15	257	268	13	13	99	145	22	1	197	180	10	4	459	510	7
1	108	143	17	17	158	182		12	15	54	41	27	3	46	63	32	6	261	286	8
3	404	446	8	19	60	44		25	17	47	32	23	5	79	65	20	8	394	413	7
5	302	326	8										7	89	87	17	10	187	139	12
7	250	245	9		H	1	11			H	1	15	9	94	79	16	12	215	223	12
9	34	18	38										11	75	10	14	14	87	121	28
11	242	293	12	1	160	153		13	1	83	110	29					16	-113	35	10
13	391	399	10	3	379	382		9	3	355	341	10		H	1	20	18	167	135	12
15	65	76	33	5	386	372		10	5	-36	34	27					20	213	171	9
17	106	79	20	7	125	146		24	7	54	80	27	1	185	137	10				
19	98	72	20	9	246	286		16	9	103	126	23	3	147	118	12		H	2	2
21	111	80	14	11	39	125		57	11	251	264	10	5	104	91	15				
				13	83	9		32	13	258	213	10	7	0	26	72	0	388	312	7
				15	333	261		10	15	37	15	30	9	187	156	7	4	981	1006	8
				17	214	187		11	17	161	122	7	11	36	61	21	6	592	579	8
				19	32	14		32									8	684	646	9
										H	1	16					10	475	444	8
1	170	165	12														12	134	207	19
3	1063	1189	12		H	1	12										14	371	348	10
5	147	184	15														16	181	166	14
7	146	144	14														18	324	253	6
9	509	548	9	1	-35	9		22	3	349	301	10	3	95	54	14	20	324	287	8
11	686	714	10	3	162	205		16	5	-72	49	17	5	-57	32	24				
13	330	341	11	5	-26	58		28	7	101	44	24	7	-34	25	32				
15	211	209	14	7	95	47		27	9	162	138	14	9	-38	26	24				
17	313	309	10	9	127	102		25	11	266	235	9						H	2	3
19	-32	29	33	11	151	146		22	13	253	215	8								
21	312	265	7	13	67	1		37	15	56	13	19					0	1013	1053	11
				15	-26	53		44					1	-56	16	22	4	439	518	7
				17	26	53		40					3	141	114	8	6	135	185	12
				19	-26	11		27					5	52	37	17	8	319	354	8
																	10	110	65	19
1	203	211	11						1	142	142	17					12	886	509	11
3	384	429	5		H	1	13		3	187	207	15		H	1	23	14	184	141	14
5	51	37	36						5	-35	24	20					16	104	133	23
7	67	71	30	1	201	195		13	7	90	85	21	1	-20	10	32	18	180	159	12
9	298	310	11	3	341	347		10	9	38	16	35					20	280	238	8
11	358	322	11	5	104	116		25	11	230	163	8								
13	123	129	23	7	-21	33		35	13	67	56	18						H	2	4
15	62	68	37	9	361	323		12	15	53	19	17	0	1141	1189	12				
17	59	83	33	11	319	263		12					4	561	519	7				
19	56	6	29	13	-51	74		36					6	147	21	11	0	933	534	11
21	160	139	8	15	77	87		24					8	220	163	9	2	765	740	10
				17	51	132		19	1	376	359	8	10	563	541	9	4	133	13	14
									3	108	68	17	12	555	530	9	6	614	532	8
					H	1	14		5	102	115	18	14	489	447	9	8	230	159	10
1	361	392	8						7	91	20	18	16	164	157	17	10	301	270	9
3	378	367	9	1	43	11		42	9	315	259	8	18	31	35	40	12	746	746	10
5	335	374	10	3	142	156		19	11	103	53	15	20	-55	1	23	14	141	119	18
7	92	106	28	5	-58	56		20	13	18	5	34					16	294	254	11
9	344	393	11	7	-42	6		26									18	127	59	17
11	209	206	17	9	119	56		22		H	1	19					20	-26	11	33

				10	173	106	17	2	185	177	15	12	24	29	38	0	78	74	10
	H	2	5		12	271	263	13	4	400	382	11	14	82	54	13	2	80	56
					14	183	170	17	6	440	408	11							
0	615	730	9		16	274	268	10	8	353	349	12	H	2	17		H	3	0
2	420	503	9		18	75	77	25	10	220	161	15							
4	674	634	9		20	150	150	9	12	78	51	30	0	326	292	8	1	43	0
6	955	975	11						14	144	116	16	2	222	207	11	3	-57	0
8	694	686	9		H	2	4		16	172	141	11	4	101	39	19	5	77	0
10	100	27	21						18	177	175	7	6	158	116	13	7	-35	0
12	600	613	10		0	969	1039	12					8	136	87	13	9	-66	0
14	255	244	11		2	188	155	13	H	2	13		10	121	96	13	11	79	0
16	82	67	25		4	373	353	10					12	255	218	7	13	-66	0
18	409	353	8		6	459	451	11	2	-28	98	31	14	109	102	9	15	66	0
20	409	362	7		8	681	684	11	4	60	102	40					17	-57	0
					10	113	109	26	6	153	175	19	H	2	18		19	-35	0
					12	300	302	12	8	50	52	44							
					14	230	214	12	10	152	177	21	0	118	154	17	H	3	1
0	482	419	9		16	85	58	26	12	107	72	21	2	228	210	9			
2	486	417	9		18	227	181	10	14	77	43	24	4	172	169	12	1	346	362
4	424	405	9		20	281	249	7	16	17	17	42	6	81	2	19	3	985	1088
6	609	599	9										8	143	132	12	5	93	109
8	403	377	9		H	2	10		H	2	14		10	129	146	12	7	74	15
10	330	297	10										12	75	66	13	9	566	573
12	285	268	12		0	673	613	10	2	111	33	22					11	737	728
14	178	154	17		2	377	350	9	4	259	243	9	H	2	19		13	485	464
16	208	188	12		4	341	303	11	6	162	176	18					15	227	150
18	215	209	11		6	235	221	15	8	108	33	23	0	149	99	12	17	372	321
20	136	130	13		8	159	172	23	10	320	326	10	2	141	95	12	19	75	66
					10	357	350	13	12	240	218	10	4	209	214	9			
					12	312	295	13	14	240	223	9	6	182	135	9	H	3	2
					14	289	240	11	16	47	35	22	8	194	175	8			
0	1463	1559	16		16	55	106	38					10	58	70	17	1	141	149
2	310	301	10		18	58	82	18	H	2	15						3	121	117
4	55	86	36										H	2	20		5	339	362
6	523	556	9		H	2	11		2	129	156	20					7	177	164
8	521	567	9						4	134	118	20	0	-75	30	22	9	161	158
10	395	375	10		0	431	379	9	6	218	196	12	2	183	177	9	11	99	153
12	601	581	10		2	262	286	12	8	272	256	10	4	-11	28	49	13	273	259
14	561	505	10		4	-33	9	27	10	85	67	22	6	202	181	7	15	232	202
16	53	19	33		6	174	128	17	12	57	30	28	8	107	114	10	17	80	38
18	221	192	11		8	-44	77	26	14	84	68	17					19	72	27
20	277	261	8		10	177	110	19	16	53	94	10	H	2	21				
					12	267	226	15									H	3	3
					14	222	185	14	H	2	16		0	188	174	8			
					16	73	60	26					2	120	113	10	1	313	290
0	332	401	10		18	-49	52	21	2	197	184	15	4	79	36	13	3	593	621
2	909	928	11						4	249	255	13	6	44	22	18	5	288	316
4	385	341	10		H	2	12		6	225	230	10					7	147	141
6	281	224	8						8	294	256	9	H	2	22		9	492	461
8	628	601	10		0	188	191	16	10	120	113	15					11	400	377

13	140	139	18	9	-64	43	22	11	145	119	20	3	-48	28	34	2	70	47	31
15	262	235	10	11	-33	8	33	13	52	50	35	5	54	41	28	4	126	8	18
17	266	228	9	13	52	39	36	15	332	263	8	7	73	25	21	6	130	140	20
19	68	25	21	15	23	40	50	17	100	59	11	9	106	91	15	8	101	23	21
				17	-54	21	24					11	61	37	19	10	-25	30	36
	H	3	4		15	40	3	25		H	3	12				12	71	4	25
													H	3	17	14	62	47	27
1	49	2	45		H	3	8	1	183	222	17					16	-42	22	26
3	268	290	13					3	110	166	27	1	95	53	17				
5	102	75	23	1	81	85	33	5	101	45	25	3	103	126	18		H	4	2
7	57	23	32	3	189	171	17	7	98	114	30	5	73	71	21				
9	122	129	21	5	158	157	25	9	51	60	37	7	76	30	17	0	456	454	10
11	138	160	18	7	123	127	27	11	119	54	19	9	154	94	8	2	149	119	21
13	97	11	24	9	96	26	29	13	47	39	32	11	101	85	10	4	160	117	17
15	71	68	27	11	118	106	24	15	-32	14	30					6	372	326	10
17	131	102	15	13	-96	51	21						H	3	18	8	298	264	11
19	-27	5	29	15	55	61	23		H	3	13					10	89	75	24
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	H	3	5					1	429	406	10	3	116	82	13	1+	220	165	10
								3	320	254	11	5	34	24	30	16	57	3	24
1	116	142	31					5	192	159	15	7	-52	16	26				
3	265	261	15	1	33+	358	12	7	-56	1	33	9	30	34	24		H	4	3
5	119	165	24	3	242	162	15	9	469	388	8								
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9	220	20+	14	7	-36	10	29	13	-71	33	22		H	3	19	2	111	143	37
11	156	135	17	9	363	346	13	15	229	201	7	1	31	24	29	4	165	118	19
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15	181	147	12	13	87	43	27		H	3	14	5	61	27	17	8	124	121	22
17	134	109	15	15	203	164	13					7	20	5	28	10	46	54	40
19	83	19	17	17	102	52	16	1	162	121	17					12	228	205	11
								3	154	146	19		H	3	20	14	28	34	41
								5	122	75	17					16	-21	11	35
								7	118	103	18	1	43	21	19				
1	218	249	17	1	297	329	13	9	45	6	34	3	-80	9	15		H	4	4
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9	183	186	18	9	-38	106	30		H	3	15	0	420	393	10	4	1+9	193	25
11	154	136	18	11	134	106	22					2	328	314	12	6	133	114	21
13	30	23	50	13	-14	20	54	1	125	92	17	4	193	185	14	8	89	55	28
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17	81	85	22	17	72	72	18	5	206	150	11	8	256	258	12	12	318	269	9
19	52	46	25					7	62	64	27	10	-29	45	34	14	33	8	38
								9	210	197	10	12	346	255	8	16	94	56	17
								11	136	130	13	14	240	166	10				
	H	3	7					13	60	34	17	16	-54	17	23		H	4	5
				1	684	687	11												
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3	60	61	42	5	162	154	21		H	3	16		H	4	1	2	82	14	46
5	93	43	33	7	143	94	21									4	245	269	19
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12	452	350	8		H	4	10					13	49	1	27					
14	-75	53	22						H	4	15									
16	-7	16	44	0	220	212	15						H	5	3		1	175	123	16
				2	205	127	15	0	162	136	12						3	307	248	10
	H	4	6	4	139	109	17	2	108	93	15	1	177	168	15		5	103	45	21
				6	246	200	12	4	56	26	24	3	68	51	58		7	-34	9	41
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4	192	196	23	12	212	161	10					9	41	84	39		H	5	9	
6	109	23	28	14	125	83	12		H	4	16		11	55	36	28				
8	125	103	22									13	43	3	30		1	37	62	43
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6	156	114	25					4	84	62	13						7	30	18	36
8	311	248	11		H	4	12						H	5	5		9	102	68	13
10	73	112	32						H	5	0									
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14	-24	80	39	2	123	109	18	1	-82	0	29	3	15	52	71					
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	H	4	8	8	180	123	11	9	-51	0	35	9	37	21	41		5	107	73	14
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0	240	215	17	12	173	141	9	13	55	0	20									
2	-70	59	19										H	5	6		H	5	12	
4	83	81	36		H	4	13		H	5	1									
6	148	82	24									1	124	105	35		1	54	47	24
8	166	113	18	C	131	104	16	1	216	204	12	3	-27	257	49		3	20	22	38
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14	91	73	23	6	57	12	27	7	47	75	35	9	269	214	11					
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	H	4	9	10	56	6	22	11	-44	49	30									
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0	213	159	17														3	59	58	19
2	151	122	22		H	4	14		H	5	2		1	91	28	32	5	55	39	17
4	145	81	19									3	55	20	37					
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8	114	110	26	2	114	82	15	3	80	1	35	7	27	25	56					
10	-45	43	40	4	119	77	15	5	28	45	43	9	63	32	35		0	167	159	54

2 117 28 87 2 66 113 109 H 6 2 H 6 4 0 259 181 23
H 6 1 0 309 289 38

E. (1.d.) $(\text{TTT})_2\text{I}_3$ - Structure factors $\text{Pmc}2_1$, full cell. Ni-filtered
 $\text{CuK}\alpha$ radiation.

				10	2191	2267	25					6	376	362	13	9	828	576	14
H	J	0		12	1844	1861	23	0	2809	2786	31					10	70	50	45
				14	1420	1461	21	2	2012	2034	24	H	2	0		11	518	557	19
2	2212	1785	16	16	1286	1331	20	+	1174	1247	20					12	77	44	47
4	3690	3222	37	18	1066	1072	17	6	384	416	33	1	47	1	41	13	250	189	34
6	2685	2369	29	20	892	937	16	8	635	657	23	3	92	3	50	14	-27	39	82
8	634	944	15	22	581	643	15	10	1353	1375	21	5	82	4	43	15	507	487	25
10	4365	4412	44					12	1730	1562	22	6	46	25	56	16	-111	35	40
12	3727	4021	40	H	0	8		14	1396	1352	18	7	56	4	49	17	364	355	30
14	3208	3339	35					16	584	581	17	8	-99	26	36	19	73	8	74
16	1097	1053	22	0	2109	2116	24	18	193	183	22	9	65	3	45	21	321	301	22
18	567	611	23	2	2914	2577	32					10	-106	27	35				
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22	523	558	16	6	3465	3488	37					12	98	27	39				
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H	0	2		10	1702	1749	22	2	777	829	21	14	119	28	38	3	713	904	14
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6	5666	5410	57	16	1243	1266	20	8	1340	1296	21	17	-53	3	60	7	351	444	19
8	5187	5169	53	18	1225	1234	18	10	722	755	17	19	130	4	48	8	44	80	57
10	2112	2077	24	20	671	925	16	12	390	407	23	21	35	4	77	9	1387	1513	18
12	701	641	17					14	253	285	26					10	10	63	83
14	825	855	18	H	0	10		16	490	514	13	H	2	1		11	555	570	16
16	1627	1669	22													12	23	48	78
18	1841	1965	24	0	3889	3949	41	H	0	18		1	546	636	15	13	-68	241	57
20	1769	1774	21	2	3493	3566	37					3	1871	2171	21	14	80	36	52
22	691	741	15	4	1118	1164	18	0	1157	1199	20	5	1214	1358	15	15	449	560	27
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H	0	4		8	1010	1023	20	+	710	719	17	7	528	617	14	17	304	253	31
				10	1589	1629	23	6	746	719	17	8	-64	81	46	19	173	205	42
0	4761	4466	51	12	3037	3121	35	8	511	500	18	9	1134	1261	16	21	401	322	19
2	5750	5091	56	14	1107	1127	22	10	618	635	16	10	-33	64	63				
4	3380	3165	36	16	1178	1165	19	12	716	718	13	11	828	939	16	H	2	4	
6	1712	1659	20	18	297	361	32	14	535	526	12	12	115	48	34				
8	2383	2394	25	20	139	126	42					13	174	106	46	1	3515	3604	37
10	2676	2708	29					H	0	20		14	80	35	49	3	639	686	15
12	3080	3164	33	H	0	12						15	764	766	20	5	863	1020	14
14	2448	2548	28					0	763	737	15	16	-83	26	49	6	46	112	47
16	1381	1458	20	0	813	804	20	2	656	605	15	17	586	618	23	7	315	311	20
18	467	542	23	2	836	857	18	+	508	550	17	19	-126	41	52	8	22	58	72
20	381	446	22	4	3325	3306	36	6	533	566	16	21	441	401	17	9	2088	2210	24
22	564	625	16	6	2199	2359	26	8	444	437	16					10	81	84	45
				8	2538	2527	30	10	495	502	14	H	2	2		11	555	573	19
H	0	6		10	1659	1668	25	12	425	464	13					12	38	71	71
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2	3922	3879	41	16	578	649	21					5	576	659	13	15	913	1032	19
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6	3239	3301	24					2	821	779	13	7	201	231	26	17	395	438	30
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21	556	498	17	9	68	91	76	H	2	10	15	-52	139	68	17	321	196	15	
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3	223	203	30	14	24	37	89	6	58	177	49					3	696	552	21
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6	-75	93	35	16	-73	29	52	8	95	157	39	3	680	720	21	5	-143	93	35
7	400	519	19	17	212	84	41	9	686	853	23	4	27	66	74	6	-89	138	48
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14	-109	37	45	3	2121	2376	25	17	395	315	24	11	637	465	24				
15	466	415	26	5	294	432	31	19	120	33	50	12	-37	39	81		H	2	17
16	-70	28	55	6	43	170	51					13	-103	193	72				
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	H	2	6	11	1370	1361	20	5	774	685	20					7	180	143	42
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1	1463	1598	19	13	660	744	22	7	250	374	48	3	285	332	39	11	459	279	17
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6	125	148	24	17	624	671	20	10	-11	52	98	6	67	158	50				
7	157	53	40	19	-64	102	66	11	78	319	113	7	-84	23	51		H	2	18
8	-90	130	37	21	624	493	14	12	-86	43	56	8	157	142	29				
9	1510	1723	19					13	166	91	64	9	237	76	45	1	751	704	16
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11	1286	1451	19					15	665	475	20	11	232	240	52	5	205	225	37
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5	603	741	17	17	117	242	66	10	77	134	57	9	205	297	47				
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7	0	58	144	4	549	693	17	12	404	387	30					11	192	134	61
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1	123	14	41	10	534	477	25		H	3	4					18	41	106	115
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				17	165	171	56	4	302	305	27	13	247	146	45	1	215	20	45
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				2	298	331	29	11	231	162	42	0	507	559	22	10	341	321	43
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				4	270	312	28	13	176	157	60	2	479	549	26	13	127	145	84
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10	326	225	28	19	191	118	45	5	142	117	55	17	58	145	106	8	246	208	59
11	151	173	46					6	555	547	19	18	247	173	43	10	100	187	105
12	206	154	51		H	3	3	7	194	149	42					11	46	117	136
14	220	164	51					8	509	480	21		H	3	8	13	258	116	49
15	184	155	56	0	521	716	20	9	145	165	58					14	127	142	86
16	237	136	44	1	166	28	70	10	476	422	25	0	329	283	32	15	54	109	123
17	0	138	143	2	513	701	22	11	0	178	160	1	137	20	65	17	278	59	38
18	188	111	51	3	141	62	51	12	448	358	30	2	257	279	42				
19	53	120	98	4	546	659	17	13	145	178	71	3	175	59	60		H	3	11
				5	165	127	42	14	339	298	37	4	253	267	59				
				6	560	599	18	15	154	171	53	5	171	91	62	0	367	380	34
				7	157	160	47	16	255	242	47	6	103	249	90	1	-139	17	72
0	458	756	18	8	519	530	20	17	113	159	77	7	191	115	55	2	327	375	39
1	218	30	70	9	120	181	62	18	195	192	53	8	243	227	47	3	147	49	69

12	623	617	27	14	479	434	18	6	364	290	19	H	6	2	19	167	20	35
14	577	511	22	16	54	97	45	8	388	316	16							
16	110	242	76					10	116	125	35	1	282	279	32	H	6	6
18	196	170	36	H	4	15						3	242	200	34			
	H	4	11		2	259	316	41				5	677	720	18	1	436	450
					4	269	241	40	0	-151	80	7	353	329	28	3	272	365
0	861	829	19	6	435	394	25	2	366	300	45	9	323	304	32	5	-65	161
2	523	612	24	8	544	523	20	4	-22	81	19	11	198	303	53	7	100	85
4	-67	81	54	10	170	126	44	6	434	303	99	13	546	519	23	9	365	387
6	347	219	34	12	114	55	56	8	214	195	15	15	463	423	21	11	308	282
8	-88	106	53	14	168	125	35				20	17	159	85	43	13	61	67
10	353	267	38	16	187	193	21	H	4	21		19	144	66	40	15	100	160
12	533	498	30													17	161	166
14	444	410	28	H	4	16		0	375	319	16	H	6	3		19	104	104
16	147	146	52					2	241	209	21	1	625	583	22			
18	-98	72	43	2	394	350	30	4	158	90	27	3	1184	1276	20	H	6	7
				4	458	474	26	6	89	60	36	5	576	672	20	1	49	28
	H	4	12	6	450	424	21					7	293	293	31	3	120	129
				8	587	498	18	H	4	22		9	981	963	19	5	187	71
0	377	372	32	10	240	207	31					11	796	729	20	7	-53	76
2	369	382	31	12	49	77	77	0	157	112	21	13	281	263	36	9	-128	108
4	795	768	22	14	165	123	27	2	161	154	22	15	523	520	20	11	-67	51
6	878	822	23									17	532	431	18	13	104	52
8	705	698	24	H	4	17		H	6	0		19	137	44	42	15	40	72
10	440	350	30													17	-108	37
12	157	133	60	0	654	504	17	1	86	4	38	H	6	4		19	30	35
14	287	236	32	2	443	436	22	3	-114	13	33							
16	344	291	23	4	201	84	38	5	155	20	31	1	98	32	90	H	6	8
18	353	361	15	6	316	266	27	7	-71	26	67	3	536	569	27			
				8	272	160	27	9	-133	28	44	5	204	174	47	1	162	162
	H	4	13	10	241	181	26	11	157	24	33	7	115	79	65	3	376	362
				12	509	427	15	13	-132	28	39	9	244	261	43	5	316	314
				14	219	230	19	15	133	25	51	11	276	338	37	7	245	255
2	-57	175	62					17	-114	22	53	13	194	36	47	9	193	72
4	120	224	80					19	-71	19	55	15	143	149	53	11	236	207
6	305	380	39	H	4	18						17	263	198	29	13	-191	107
8	100	97	89									19	-55	28	57	15	191	142
10	305	371	42	0	235	264	34	H	6	1						17	221	151
12	214	177	43	2	455	377	19											
14	155	97	48	4	344	292	24	1	692	729	16	H	6	5				
16	34	6	84	6	162	83	39	3	1965	2172	23							
				8	286	229	24	5	187	301	42	1	231	297	63	H	6	9
	H	4	14	10	259	250	23	7	148	98	51	3	530	517	31	1	667	720
				12	150	137	27	9	1129	1175	19	5	237	351	48	3	482	403
2	222	173	45					11	1471	1450	21	7	152	141	58	5	283	252
4	516	461	19					13	969	881	19	9	440	442	28	7	-73	27
6	323	351	37	H	4	19		15	454	426	21	11	311	289	34	9	725	683
8	216	151	47	0	298	201	24	17	742	638	16	13	-88	21	52	11	353	219
10	639	639	20	2	283	217	24	19	150	128	42	15	362	284	24	13	174	107
12	475	432	21	4	418	392	19					17	267	212	31	15	405	252

17	235	149	32	1	324	235	34					12	456	404	22	16	140	78	39	
	H	6	10		3	308	284	38		H	6	20		14	56	82				
					5	244	139	35						16	-43	4	71	H	8	
					7	237	194	36	1	86	43	39								
1	593	651	27	5	90	7	67	3	-161	14	31		H	8	4		0	479	453	34
3	426	418	33	11	94	146	62										2	-140	138	38
5	165	136	64	13	152	23	33		H	8	0		0	267	168	47	4	165	132	72
7	275	239	52										2	521	463	45	6	296	173	48
9	-76	206	60		H	6	15		0	839	783	20	4	298	415	50	8	332	204	37
11	268	213	45						2	655	553	25	6	266	326	42	10	193	105	52
13	-28	41	109	1	250	180	35	4	386	435	29	8	178	230	57	12	294	241	32	
15	134	37	56	3	593	520	19	6	775	819	20	10	52	196	95	14	183	156	47	
17	145	154	37	5	412	301	22	8	512	577	24	12	635	498	18					
				7	125	153	54	10	-59	64	69	14	66	12	76		H	8	9	
	H	6	11	9	420	395	21	12	691	575	17	16	187	80	35					
				11	272	282	26	14	479	389	20						0	425	346	35
				13	120	54	34	16	-108	6	46		H	8	5		2	302	266	44
1	1366	1313	22														4	291	156	39
3	476	389	26										0	1139	1051	23	6	513	526	27
5	323	318	42		H	6	16		H	8	1		2	164	56	92	8	229	215	52
7	285	148	43										4	489	511	38	10	-91	100	80
9	1038	999	23	1	277	272	30	0	122	9	69	6	54	117	114	12	12	348	232	30
11	265	266	39	3	-97	73	68	2	140	94	62	8	641	630	22	14	268	234	28	
13	105	84	71	5	107	76	56	4	251	11	37	10	38	87	107					
15	663	503	16	7	147	40	42	6	260	302	40	12	903	687	17		H	8	10	
17	200	139	22	9	213	180	31	8	203	44	43	14	-150	72	45					
				11	122	76	37	10	-49	53	72	16	-14	39	86		0	439	406	31
	H	6	12					12	141	13	49						2	410	274	30
					H	6	17	14	124	113	55						4	275	233	35
								16	-85	51	52		H	3	6		6	491	437	24
1	306	379	35														8	291	289	37
3	221	285	53	1	190	105	35						0	708	695	31	10	126	38	63
5	202	72	51	3	205	259	36		H	8	2		2	633	691	30	12	423	323	21
7	196	200	61	5	147	140	42						4	384	322	47	14	250	173	24
9	102	110	75	7	153	53	35	0	910	987	20	6	6	218	40	57				
11	238	103	38	9	307	177	17	2	297	343	41	8	250	231	45					
13	94	73	64	11	202	168	20	4	319	187	35	10	404	358	26		H	8	11	
15	-64	23	61					6	744	647	21	12	470	408	21					
					H	6	18	8	595	492	22	14	359	254	25		0	182	268	53
	H	6	13					10	178	195	49	16	266	229	26		2	152	177	55
				1	112	18	48	12	524	479	20						4	144	86	57
1	857	764	20	3	233	154	26	14	439	429	20						6	320	225	30
3	635	546	22	5	69	50	61	16	114	11	49		H	3	7		8	277	135	30
5	383	357	30	7	-105	28	52										10	-109	66	66
7	-112	37	65	9	61	75	48		H	8	3		0	1128	1108	28	12	241	193	27
9	937	750	17										2	-82	34	52	14	114	139	35
11	432	310	22		H	6	19		0	348	407	36	4	370	258	41				
13	-143	81	45						2	223	307	74	6	312	267	50				
15	457	419	14	1	63	35	59	4	330	219	39	8	622	512	23		H	8	12	
				3	63	37	58	6	416	303	29	10	146	234	65					
	H	6	14	5	122	51	34	8	248	221	45	12	612	508	20		0	575	510	21
				7	41	15	55	10	93	106	80	14	-48	173	78		2	247	245	37

4	213	127	40	4	148	103	34	13	98	15	54	9	538	438	22	11	11	11
6	497	399	21	6	199	133	25					11	401	279	24			
8	376	252	23	8	175	194	23		H	10	3					1	58	19
10	56	67	78										H	10	7	3	48	51
12	346	362	16		H	8	17		1	353	291	31				5	214	134
									3	136	83	117	1	183	52	64	7	66
	H	8	13		0	431	325	15	5	176	19	59	3	111	48	74		
					2	-153	34	38	7	129	41	58	5	86	76	87	H	11
0	261	204	32	4	169	131	26		9	82	156	77	7	54	34	112		12
2	133	41	54						11	109	67	56	9	126	66	71	1	108
4	-87	69	73		H	10	0		13	85	5	59	11	101	15	62	3	41
6	114	11	55														5	163
8	123	136	50		1	-165	0	56		H	10	+		H	10	8	7	-126
10	113	11	44		3	-56	0	89										
12	93	72	38		5	160	0	59	1	352	316	37	1	350	252	32	H	10
					9	-101	1	70	3	599	496	30	3	614	523	21		
	H	8	14		11	-108	1	55	5	300	266	41	5	205	139	43	1	91
					13	109	1	40	7	228	108	38	7	-68	25	63	3	118
0	385	304	23						9	542	444	20	9	440	356	24	5	110
2	228	177	30		H	10	1		11	392	276	22	11	449	365	17		
4	238	154	30						13	151	69	41					H	12
6	352	285	20		1	435	372	23						H	10	9		
8	290	220	22		3	40	113	129		H	10	5					0	333
10	-123	10	41		5	77	49	79					1	75	120	86	2	234
					7	93	125	71		1	-227	5	42	3	116	70	64	
	H	8	15		9	217	216	36	3	29	104	142	5	175	166	45	H	12
					11	-88	50	60	5	-178	16	50	7	181	57	40		1
0	324	245	24		13	-104	43	42	7	118	35	68	9	83	91	62	2	133
2	216	131	30						9	75	41	83						
4	112	47	47		H	10	2		11	-93	63	56		H	10	10	H	12
6	146	55	35															2
8	217	138	23		1	805	703	17		H	10	6		1	-126	57	55	0
					3	159	59	70					3	520	406	19		
	H	8	16		5	57	103	86	1	248	205	71	5	148	19	44	H	12

F. (TMTSF)Br_{0.8} - Structure factors, Ni-filtered CuK α radiation.

Space group Cmcm.

				22	311	231	8	15	314	251	9	16	363	373	8	2	1368	1372	10
	0	K	0	24	-39	53	37	17	423	404	8	18	39	29	33	4	1889	1761	13
				26	53	57	24	19	426	414	7	20	134	128	11	6	2271	2168	16
2	2374	2290	24	28	33	58	27	21	441	353	7	22	-64	58	16	8	3369	3030	25
4	1300	1119	10					23	83	49	20					10	1405	1318	11
6	550	640	6		3	K	0	25	64	36	21		9	K	0	12	1528	1515	12
8	1168	1164	9					27	40	12	18					14	1742	1628	13
10	310	474	7	1	4382	4052	32					1	1608	1589	12	16	194	228	10
12	273	351	7	3	421	271	5		6	K	0	3	-66	96	28	18	347	365	7
14	2828	2734	19	5	645	656	5					5	454	401	8	20	622	672	7
16	1666	1603	13	7	1171	1236	9	0	3485	3519	24	7	391	419	9	22	915	883	9
18	940	934	9	9	390	298	6	2	1537	1482	11	9	71	88	27	24	692	676	8
20	1192	1166	10	11	135	233	10	4	248	237	7	11	104	118	19	26	648	686	7
22	1071	1132	9	13	1577	1584	12	6	648	635	7	13	750	731	8	28	811	855	7
24	256	263	10	15	2522	2377	19	8	608	657	7	15	1112	1086	9				
26	101	110	17	17	556	522	7	10	119	212	16	17	257	228	7		1	K	1
28	329	341	6	19	861	806	8	12	398	364	8	19	401	385	5				
30	252	302	5	21	1166	1107	9	14	1550	1487	12					1	36	35	19
				23	665	717	7	16	1090	1004	9		10	K	0	3	320	351	5
	1	K	0	25	132	122	14	18	330	316	9					5	200	305	7
				27	143	146	10	20	945	866	8	0	869	876	8	7	2584	2387	19
1	1994	2161	14	29	264	289	5	22	855	852	8	2	532	568	7	9	1153	1202	9
3	652	581	5					24	135	133	11	4	201	217	11	11	119	92	11
5	1318	1396	10		4	K	0	26	44	79	16	6	563	570	7	13	546	458	6
7	709	569	6									8	53	18	30	15	927	913	8
9	427	535	6	0	1111	1322	8		7	K	0	10	51	12	29	17	50	59	29
11	675	762	6	2	904	1074	7					12	377	379	8	19	77	8	22
13	793	714	7	4	263	407	5	1	1131	1235	9	14	636	574	6	21	547	527	7
15	1127	1074	9	6	1506	1418	11	3	722	781	7	16	528	532	6	23	439	418	7
17	339	320	7	8	207	195	7	5	707	740	7					25	57	9	28
19	995	1006	9	10	35	137	28	7	620	570	7		11	K	0	27	415	417	6
21	647	552	7	12	924	516	8	9	284	252	9					29	399	441	5
23	192	153	11	14	997	892	9	11	454	499	7	1	39	49	34				
25	94	73	18	16	1130	1138	10	13	613	538	11	3	119	154	15		2	K	1
27	77	52	19	18	178	203	12	15	748	679	9	5	60	99	26				
29	189	173	6	20	875	794	8	17	362	350	8	7	42	1	31	0	-32	0	22
				22	430	345	9	19	644	620	8	9	66	95	21	2	180	137	6
	2	K	0	24	76	78	21	21	721	644	7	11	25	50	32	4	755	735	6
0	464	215	5	26	46	58	23	23	225	195	7	13	-40	28	19	6	1755	1792	12
2	684	824	6	28	98	92	10									8	2486	2289	18
4	-30	85	24						8	K	0					10	465	376	6
6	1572	1519	11		5	K	0					12	K	0		12	490	430	6
8	322	336	5	1	361	489	5	2	186	192	30	0	1096	1027	9	14	536	578	6
10	109	209	13	3	843	823	7	4	207	208	11	2	498	451	6	16	490	452	7
12	969	939	8	5	605	628	6	6	344	357	10	4	117	127	11	18	244	257	9
14	446	409	6	7	333	316	6	8	157	180	13	6	209	194	7	20	282	237	8
16	1093	1030	9	9	381	356	6	10	99	6	20		0	K	1	22	533	485	7
18	167	121	12	11	433	486	6	12	196	208	14					24	56	66	27
20	751	680	8	13	267	200	9	14	255	194	10	0	-69	0	13	28	227	229	6

				23	263	242	8					12	546	481	6	5	874	935	7
	3	K	1	25	43	17	26		9	K	1	14	2316	2191	17	7	1102	1063	9
				27	147	170	6					16	1469	1475	11	9	93	50	14
1	1068	1109	8					1	382	400	9	18	736	754	8	11	54	9	24
3	165	102	6		6	K	1	3	-30	26	46	20	1160	1094	9	13	1444	1322	11
5	1815	1903	13					5	803	784	8	22	921	874	8	15	2087	1999	15
7	3066	2960	21	0	55	0	21	7	1176	1116	10	24	182	193	12	17	620	595	7
9	2275	2223	16	2	795	816	7	9	957	939	9	26	66	89	22	19	761	754	8
11	756	793	7	4	496	517	6	11	172	192	15	28	278	264	5	21	1051	949	9
13	1902	1791	14	6	1693	1668	13	13	823	803	8				23	507	521	7	
15	786	781	8	8	2154	2160	16	15	396	378	7	1	K	2	25	70	92	20	
17	91	133	20	10	544	529	7	17	22	43	39				27	121	105	9	
19	433	464	8	12	1149	1091	10	19	164	178	8	1	2410	2423	18				
21	700	676	7	14	936	848	8					3	368	440	5	4	K	2	
23	882	852	11	16	116	159	22	10	K	1		5	783	778	7				
25	153	160	11	18	151	183	17					7	402	456	5	0	1795	1739	13
27	773	808	7	20	483	466	7	0	62	0	26	9	217	196	7	2	942	1012	7
29	385	431	4	22	764	724	7	2	123	95	17	11	396	389	6	4	329	305	5
				24	365	370	6	4	113	59	18	13	680	736	7	6	902	947	7
	4	K	1	26	491	529	4	6	825	825	8	15	1126	1058	9	8	-14	51	41
								8	975	956	9	17	188	134	11	10	107	76	14
0	10	0	36		7	K	1	10	95	81	21	19	891	844	8	12	662	676	7
2	-24	17	27					12	360	355	8	21	551	544	7	14	987	973	9
4	185	167	6	1	56	13	25	14	476	478	6	23	228	250	10	16	1020	944	9
6	1780	1799	13	3	343	372	7	16	163	188	8	25	90	53	19	18	225	222	11
8	2214	2213	16	5	362	377	7					27	53	69	20	20	729	672	7
10	102	54	13	7	1901	1941	14	11	K	1		29	160	176	6	22	392	408	8
12	556	543	7	9	812	816	8									24	110	104	15
14	924	916	8	11	475	446	8	1	43	52	32	2	K	2		26	37	45	24
16	464	458	8	13	501	444	8	3	119	108	13								
18	144	199	16	15	537	517	7	5	76	78	21	0	870	793	7	5	K	2	
20	352	309	8	17	-14	12	55	7	124	171	13	2	820	863	6				
22	562	513	8	19	87	48	21	9	101	68	14	4	36	1	22	1	774	762	7
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26	334	329	6	23	383	369	5	13	-12	65	33	8	-22	13	34	5	342	317	6
28	344	374	4									10	190	121	8	7	209	260	8
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9	553	513	6	10	131	60	17					26	-22	43	37	23	136	125	13
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13	91	24	20	14	157	213	13	2	1949	1943	14								
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19	72	94	28	20	65	34	18	8	762	681	7	1	2865	3085	21	0	2563	2726	19
21	389	372	7	22	74	82	12	10	403	344	6	3	-31	47	24	2	1264	1308	10

11	K	3		14	497	489	7	17	161	125	13	5	366	366	7	23	231	241	6
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7	45	83	21	24	46	61	23		6	K	4	15	679	668	6	2	139	111	13
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6	882	818	8	7	704	683	7	12	429	390	8	8	74	79	17	18	49	20	38
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	6	K	5	4	105	107	12	7	362	319	9	3	185	168	12	4	53	68	38
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2	32	20	39	21	201	222	6	9	81	6	27	12	336	317	8	12	196	158	10
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	9	K	5	8	63	96	35	0	952	975	9	3	64	9	35	5	110	75	18

7	246	242	10									4	100	50	21		4	K	8
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	6	K	7					5	118	140	44					6	118	150	18
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10	109	67	14	8	96	50	45					9	-36	16	37	5	-42	51	37
				10	124	36	40	0	170	197	15	11	-15	35	41				
	7	K	7	12	148	118	24	2	113	116	19								

G. (TMSF)(SCN)_{0.5} - Structure factors, Ni-filtered CuK α radiation.

Space group Cmc \bar{m} .

[illegible]

6	K	2		5	395	389	6	22	445	462	6	19	55	54	24	22	341	342	5	
				7	370	336	6	24	358	337	5	21	328	343	6	24	176	175	5	
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4	84	84	16	13	398	401	6					27	380	374	4					
6	850	826	7	15	620	598	6		1	K	3					1	206	165	10	
8	426	407	7	17	228	227	6						4	K	3	3	164	172	12	
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18	68	83	17	0	904	874	7	9	1044	952	10	8	1398	1406	11	13	466	464	6	
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5	367	376	7					27	239	237	5	26	176	177	5	2	42	4	33	
7	490	511	7		11	K	2									4	120	103	13	
9	55	76	29						2	K	3			5	K	3	6	117	66	13
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3	122	97	19	2	592	609	6	11	58	80	24					3	119	116	15	
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7	376	354	11	6	532	518	6	15	407	394	6					7	1390	1424	11	
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11	23	2	45	10	34	2	39	19	256	244	5	4	330	264	9	11	271	275	10	
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25	36	28	20	24	83	78	10	4	83	101	18	18	-42	10	27					
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					5	K	4	8	-37	12	30	22	274	285	6					
	2	K	4					10	-21	13	17	24	209	204	5	2	95	126	19	
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							11	18	-16	29	27	1	120	168	14	10	122	152	19	

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16	195	184	9	8	46	37	23				9	9	48	12	26	4	152	120	15	
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22	159	150	6	14	76	80	12	0	308	361	9	15	54	65	21	10	241	228	8	
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					9	K	5	6	243	223	11					16	126	145	10	
1	33	56	38					8	90	106	20		6	K	6	18	-6	1	35	
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11	41	37	32	9	344	343	5	18	65	59	18	6	278	289	7	3	42	27	31	
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	8	K	5	13	209	211	9					7	108	121	7					
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2	-7	58	50	4	14	21	41	6	195	182	11	10	-35	0	28	1	89	111	14	
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	6	K	7	0	688	666	10	4	-58	20	25									
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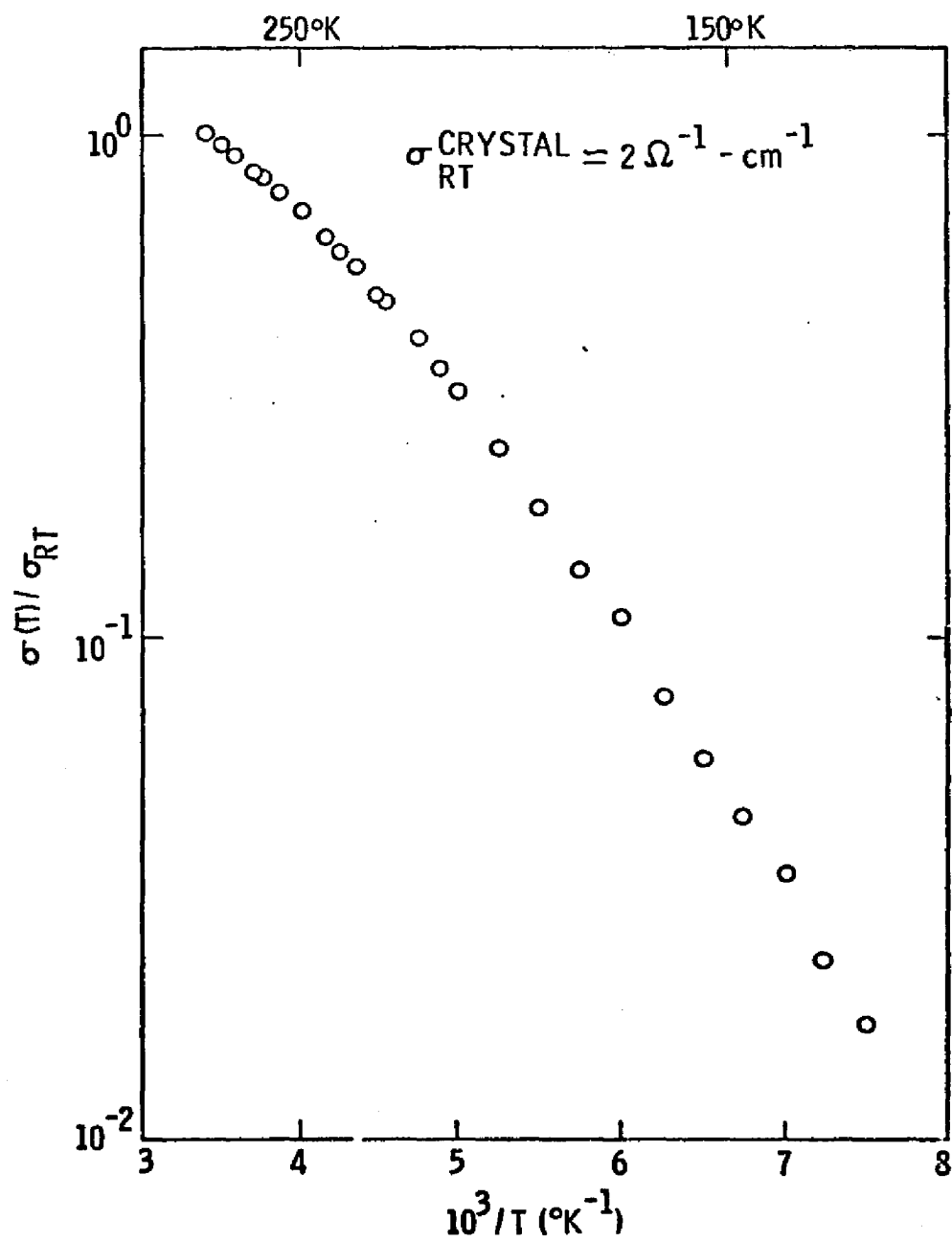
H. (TMTSF)(SCN)_{0.5} - Structure factors, Ni-filtered CuK α radiation, satellite reflections, Space group Pmc2₁.

	0	K	1'	26	42	163	29	25	-30	157	31	5	28	39	47	2	K	2'	
					3	K	1'		6	K	1'								
2	124	260	8									11	43	64	36	0	-44	305	16
4	91	79	12	1	137	128	7	0	-39	0	42	13	107	117	21	4	49	68	16
6	155	284	8	3	58	32	13	2	105	57	13	15	227	80	10	6	46	186	19
8	203	550	7	5	119	147	9	4	-26	30	33	17	91	128	17	8	38	38	23
10	-22	230	33	7	244	179	6	6	129	106	13					10	-48	66	22
12	84	170	16	9	43	73	23	8	75	219	21	10	K	1'		12	44	43	24
14	533	443	7	11	32	184	31	10	66	127	22					14	-51	3	27
16	537	450	8	13	287	309	9	12	58	93	30	0	69	0	27	16	60	37	25
18	203	74	11	15	562	207	9	14	271	247	11	2	21	16	47				
20	519	304	9	17	267	332	11	16	311	274	11	4	71	13	25	3	K	2'	
22	405	342	9	19	256	292	14	18	22	50	50	6	45	42	37				
24	161	38	14	21	436	171	9	20	316	189	10	8	61	87	27	1	42	56	18
26	-68	184	24	23	263	232	11	22	310	216	9	10	-36	56	36	3	-21	162	27
28	-50	124	22	25	39	191	31	24	67	25	19	12	72	41	22	5	39	58	21
			27	58	48	20						14	140	108	13	7	87	45	12
	1	K	1'		4	K	1'		7	K	1'		11	K	1'	9	10	56	41
1	61	271	10					1	48	66	27					11	-11	4	42
3	29	36	21	0	-25	0	25	3	74	7	20	1	49	33	28	13	91	11	17
5	149	294	7	2	45	76	18	5	91	67	18	3	-27	1	37	15	82	25	23
7	129	172	9	4	60	78	15	7	95	76	19	5	28	33	35				
9	57	112	19	6	193	175	7	9	35	43	38	7	-36	24	31	4	K	2'	
11	14	161	39	8	21	305	33	11	-31	92	44	9	26	20	35	0	-13	212	33
13	169	332	9	10	29	154	30	13	132	175	18					2	21	43	29
15	351	260	8	12	122	137	14	15	185	122	16	0	K	2'		4	37	54	22
17	91	346	19	14	285	312	9	17	85	189	26					6	40	138	24
19	290	317	10	16	312	350	10	19	179	174	13	0	71	386	10	8	35	30	26
21	256	206	11	18	-24	73	48	21	255	107	10	2	-36	69	22	10	-56	53	21
23	168	243	13	20	273	234	11					4	40	91	23	12	-4	37	54
25	30	212	38	22	218	277	12		6	K	1'	6	-26	231	30				
27	-47	62	24	24	37	39	34					8	63	45	18	5	K	2'	
			26	52	144	21		0	-331	0	15	10	-47	83	22				
	2	K	1'		5	K	1'	2	-82	5	29	12	68	56	17	1	-13	58	36
0	40	0	16					4	38	40	36	14	146	3	11	3	33	101	26
2	-38	95	17	1	45	138	20	6	107	74	25	16	100	46	16	5	46	38	22
4	-6	153	37	3	46	9	20	8	28	118	44					7	25	28	32
6	194	267	7	5	-16	146	36	10	54	76	31					9	65	63	19
8	34	398	25	7	50	89	21	12	-59	74	31	1	K	2'		11	-28	2	37
10	-26	172	32	9	39	72	28	14	-2	153	68	1	30	116	18				
12	128	183	12	11	60	100	24	16	104	185	18	3	18	192	27	6	K	2'	
14	134	365	13	13	35	227	35	18	-46	45	30	5	45	69	19				
16	259	409	13	13	35	227	35	20	28	121	31	7	-19	45	34	0	55	140	23
18	71	92	25	15	89	181	22					9	59	107	18	2	-14	33	40
20	185	264	25	17	77	241	25	9	K	1'		11	-36	6	27	4	31	40	30
22	174	322	13	19	67	233	29					13	-37	5	29	6	55	97	24
24	57	51	14	21	137	154	16	1	81	39	26	15	99	24	16	8	28	22	38
			28	23	42	175	33	3	-24	4	46								

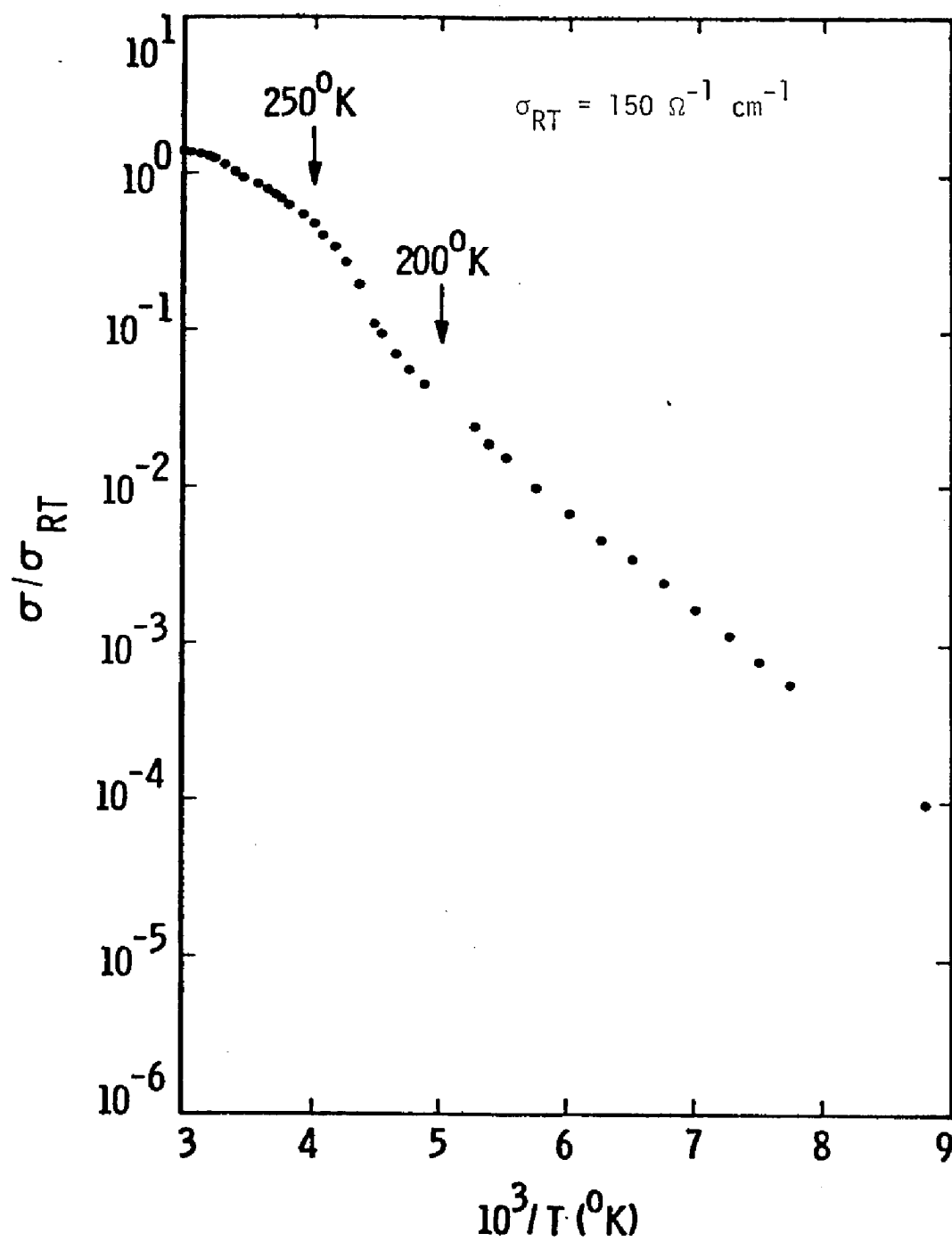
11 K 2' 1 67 15 30 5 88 10 21 9 45 17 33
3 -62 27 30 7 28 8 47

APPENDIX 2

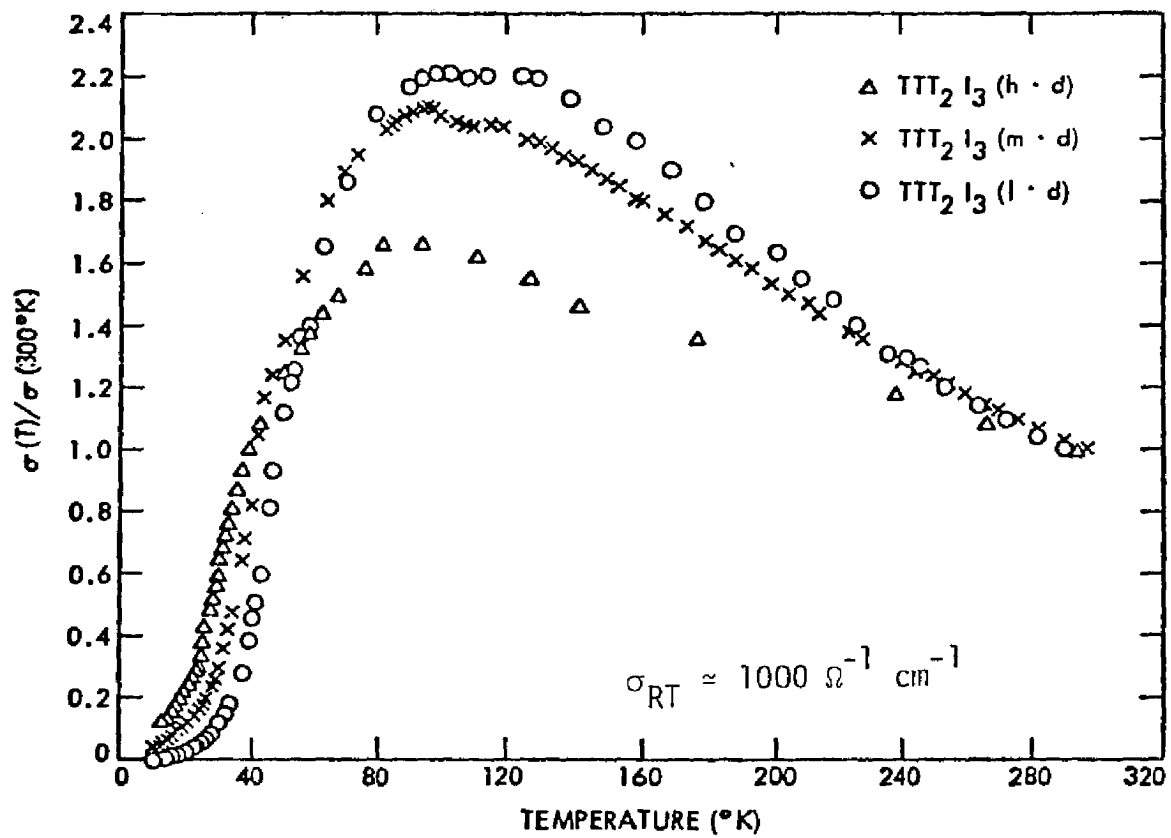
Electrical properties of crystals. Conductivities were measured using the standard 4 probe technique. Silver paint contacts were used with $[\text{Rh}(\text{CNCHCH}_2)_4]\text{ClO}_4$; aquadag contacts were used with all other crystals.



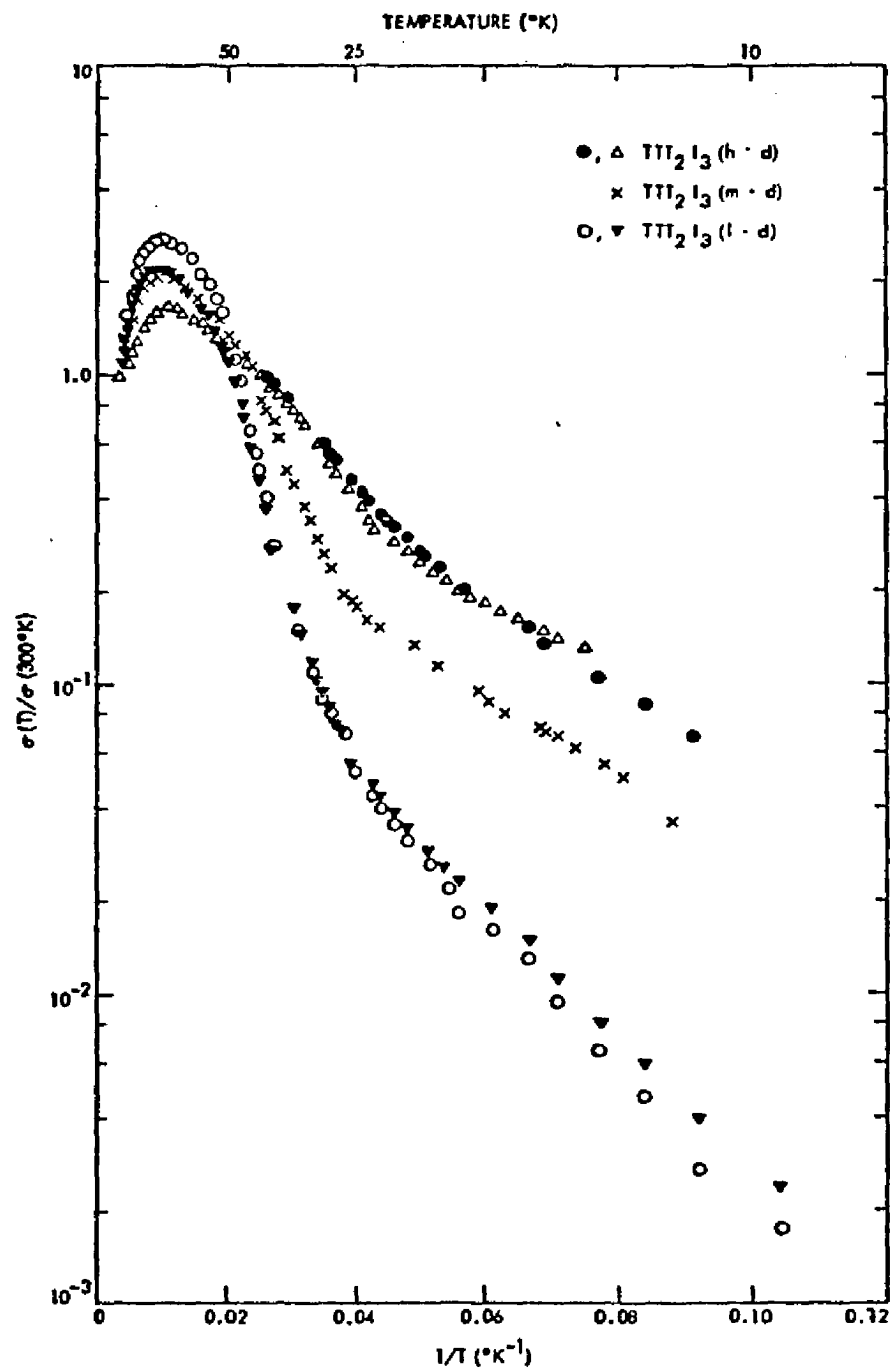
A. Conductivity of $[\text{Rh}(\text{CNCHCH}_2)_4]\text{ClO}_4$



B. Conductivity of $(\text{TTF})\text{Cl}_{0.67}$.

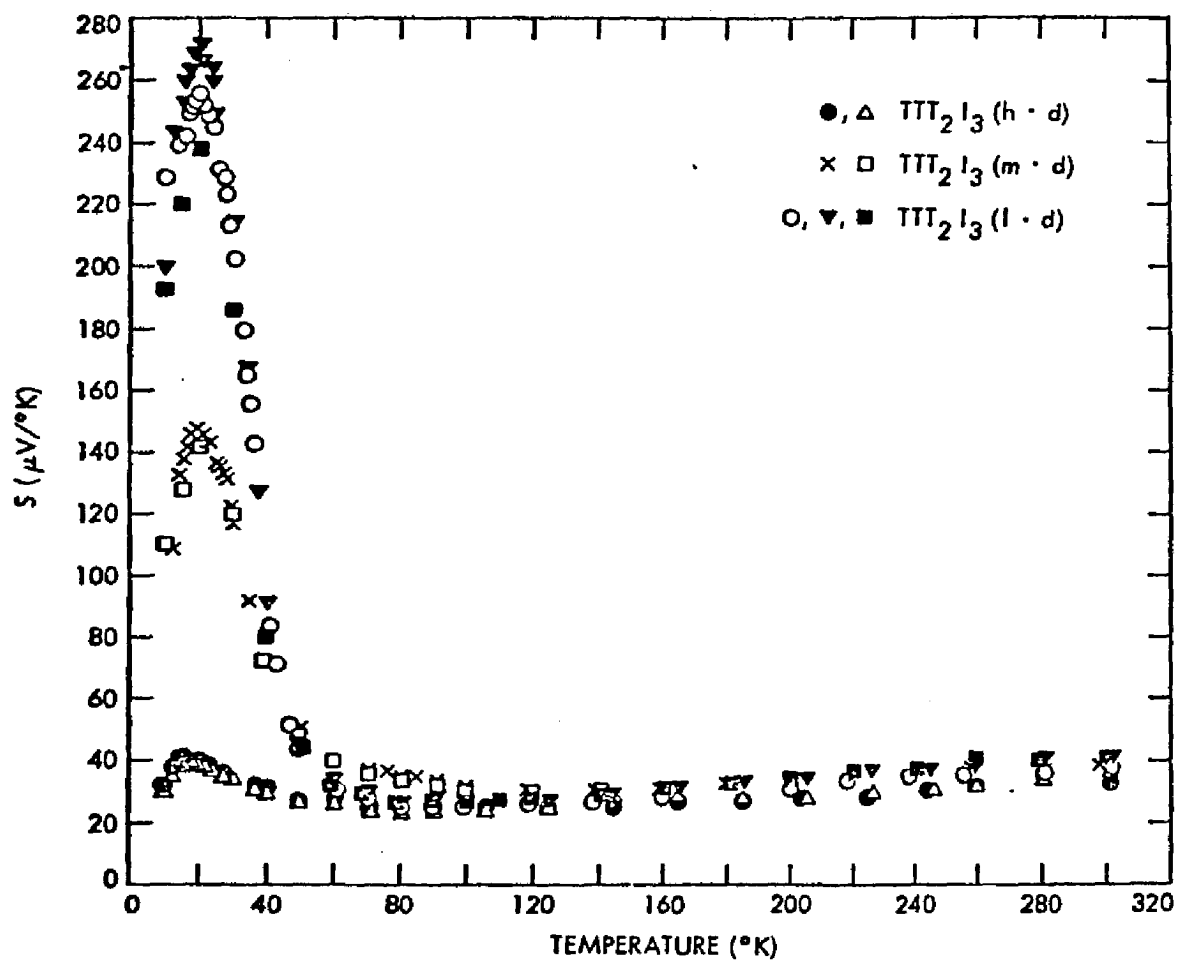


C. Conductivity of high, medium, and low disorder $(\text{TTT})_2\text{I}_3$ crystals.

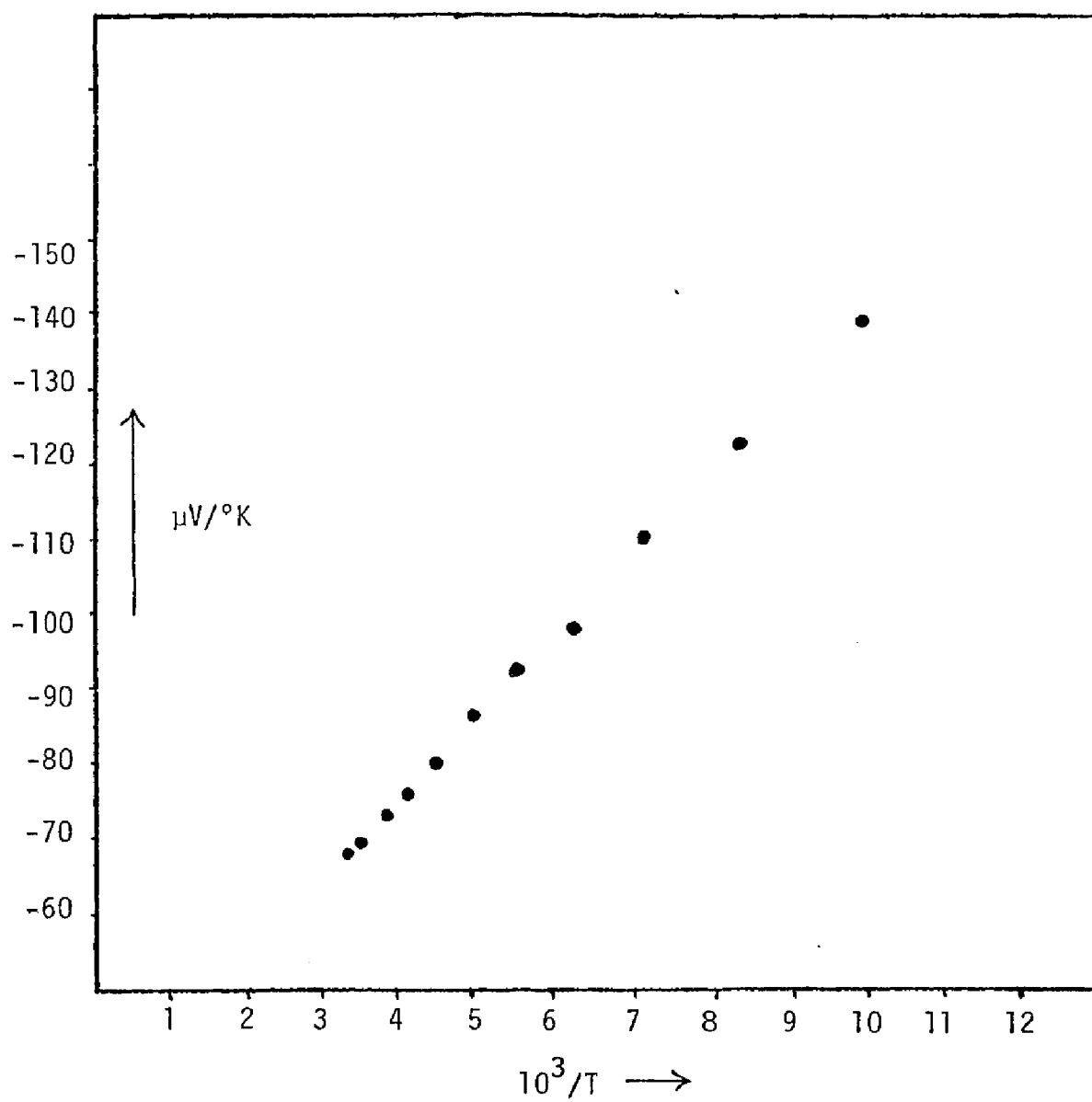


D. Conductivity of high, medium, and low disorder $(\text{TTT})_2\text{I}_3$ crystals.

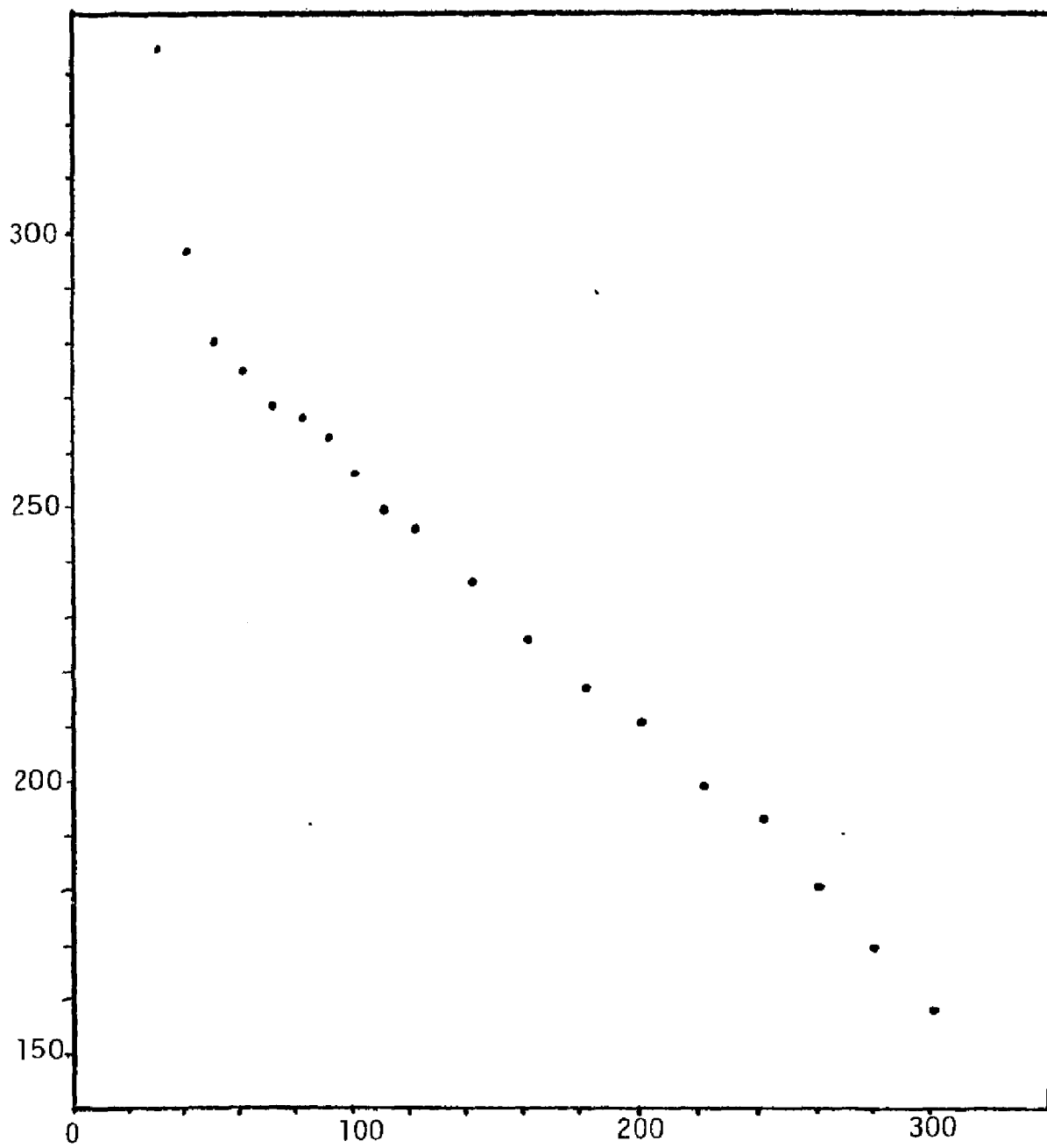
This $1/T$ plot emphasizes low temperature region.



E. Thermoelectric power of high, medium, and low disorder TlTl_2I_3 crystals. The peak at $\sim 20^\circ \text{K}$ may represent a phase transition.



F. Thermoelectric power of $(\text{TMTSF})\text{Br}_{0.8}$.



G. Thermoelectric power of $(\text{TMTSF})(\text{SCN})_{0.5}$.

APPENDIX 3

Notes on Structure Refinement

All structure refinement was carried out using CRYM system programs and the Institute's IBM 370/3032 computer. Least-squares refinement proceeds by minimization of the quantity $\sum w(F_{\text{obs}}^2 - F_{\text{calc}}^2)$. The weighted residual, wR , is $\sum w^2(F_{\text{obs}}^2 - F_{\text{calc}}^2)^2 / \sum w^2 |F_{\text{obs}}|^4$. The residual, R , is $\sum ||F_{\text{obs}}| - |F_{\text{calc}}|| / \sum |F_{\text{obs}}|$. The real goodness of fit is $[\sum w(F_{\text{obs}}^2 - F_{\text{calc}}^2)^2 / (n-p)]^{1/2}$. Weights are determined from counting statistics:

$$w = 1/(\sigma^2 F_{\text{obs}}^2 + t)$$

where t is a term which accounts for errors other than counting statistics.

In the refinement of structures with considerable disorder, refinement of some parameters is difficult or impossible. There are three important cases.

1) Atoms are located close to, but not on, symmetry elements such as mirror planes or rotation axes. Terms in matrices blow up if refinement places the atom very close to the symmetry elements.

2) The fitting function is incorrect. If thermal motion is high, approximation with thermal ellipsoids may be far from correct.

Compare, for example, the electron density map and ORTEP of carbon 6 in the room temperature structure of $[\text{Rh}(\text{CNCHCH}_2)_4]\text{ClO}_4$.

3) The fitting function is not unique. Fitting very "smeared out" electron density such as that observed for halide chains in $(\text{TTT})_2\text{I}_3$ and $(\text{TTF})\text{Cl}_{0.67}$ or the disordered perchlorates in $[\text{Rh}(\text{CNCHCH}_2)_4]\text{ClO}_4$, can be done in many ways which are equally valid.

In all cases, the observed electron density is far more informative than the parameters used to fit it. Some parameters were not refined by least squares. A fit which provided a flat difference map, low residuals, and convergence of refinable parameters was considered satisfactory.

The kinds of structure disorder and distortion which give rise to diffuse spots in the diffraction patterns of $(\text{TTF})\text{Cl}_{0.67}$ and $(\text{TTT})_2\text{I}_3$, and satellite reflections in the diffraction pattern of $(\text{TMTSF})(\text{SCN})_{0.5}$, are fairly well understood.

Disorder which appears as "mistakes" with respect to an ideal lattice gives rise to broadening of diffraction peaks (1,2,3). This phenomenon arises for the same reason that peak broadening in powder diffraction occurs. Diffraction peaks are infinitely sharp only for infinite lattices. As the number of repeating units in a diffracting domain (a single particle in powder diffraction or a mosaic block in single crystal diffraction) decreases, the resulting diffraction spots will increase in width. According to Wilson's treatment, the probability of mistakes (α) will determine the

average size of the diffracting domains and the intensity (I) as a function of a normalized coordinate (w) in reciprocal space will be given by:

$$I(w) \sim \frac{\alpha}{\alpha^2 + \pi^2 w^2}$$

This expression was used to determine the size of diffracting domains in $(\text{TTF})\text{Cl}_{0.67}$ after fast cooling. It also might be used to determine the range of order of the chloride sublattice in $(\text{TTF})\text{Cl}_{0.67}$ at room temperature, and the iodide lattice in $(\text{TTT})_2\text{I}_3$. This was not done quantitatively for the following reasons.

- a) Accurate profiles must be obtained using a very small or narrow aperture for the x-ray beam.
- b) Chloride sublattice reflections were extremely weak and very diffuse at room temperature.
- c) The diffuse third layer reflections were only ~50% wider than sublattice reflections in (l.d.) $(\text{TTT})_2\text{I}_3$, and w scans could not be obtained on the quarter-circle diffractometer.

Qualitatively, the range of order of chlorides in $(\text{TTF})\text{Cl}_{0.67}$ might be $\leq 100 \text{ \AA}$. The range of order of iodide ions in (l.d.) $(\text{TTT})_2\text{I}_3$ is hard to estimate, as part of the diffracted intensity indicates only one-dimensional order, while the spots on the diffuse third layer indicate quite long three-dimensional order ($\geq 1000 \text{ \AA}$).

Modulation of structures results in superperiods and leads to the observation of satellite reflections on diffraction patterns.

Early treatments of this effect distinguished between modulation of unit cell parameters and structure factor amplitude (4). In the first case, for modulation in, say, the x direction of the crystal, satellites with index $h = 0$ will have very small intensity. If the structure factor amplitude is modulated, these satellites will have significant intensity. Later treatments describe satellite intensity for specific kinds of structure modulation (5). A number of structures have been solved in which refinement of satellite, or study of its significance, played an important role (6,7).

The modulation of the $(\text{TMTSF})(\text{SCN})_{0.5}$ structure clearly involved structure factor amplitude modulation rather than unit cell parameter modulation. Patterson maps showed that most of the modulation was associated with the $x = 0$ plane. Further details of the model are given in Chapter 5. In general, satellites of order greater than one are not observed or are very weak. No attempt was made to fit the great decrease in intensity on going from first- to second-order satellites, which could be observed for $(\text{TMTSF})(\text{SCN})_{0.5}$.

The space groups used to fit the full cell data of $(\text{l.d.})(\text{TTT})_2\text{I}_3$ and the satellite data of $(\text{TMTSF})(\text{SCN})_{0.5}$ were picked to be consistent with the basic structures of the crystals. Symmetry elements which the superperiod necessarily destroyed were removed, and lower symmetry space groups were then picked which contained the remaining symmetry of the structure. Likewise, the space group of the low temperature monoclinic phase of $(\text{TTF})\text{Cl}_{0.67}$ has the most symmetry that

can remain after the distortion of the tetragonal cell. A good treatment of derivative crystal structures exists (8).

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